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Prediction of supercritical carbon dioxide solubility in polymers based on hybrid artificial intelligence method integrated with the diffusion theory

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Solubility is one of important research hotspots of physical chemistry properties and is widely utilized in the modification, synthesis and preparation of a lot of materials. To avoid the defects of traditional thermodynamic dissolution forecasting methods, according to the mass transfer features of a two-phase system, the dissolution process is simulated. In this paper, the diffusion theory is integrated into the improvement of particle swarm optimization (PSO) so that the particles in the algorithm evolve along with the diffusion energy. In this way, the improved PSO of dual-population diffusion is obtained and used to train the parameters of the radial basis function artificial neural network. Then, a prediction model for supercritical carbon dioxide solubility in polymers is proposed. The solution experiments of 8 polymers indicate that the predicted values with the model are consistent with the experimental results. The prediction accuracy is higher and the correlation is significant. The average relative error, mean square error and square correlation coefficient are respectively 0.0043, 0.0161, and 0.9954. The prediction model has a high comprehensive performance and provides the basis for the prediction, analysis and optimization of other physical and chemical fields.

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1. Introduction

The solubility property of supercritical carbon dioxide (ScCO₂) in a polymer is widely utilized in the modification and synthesis of materials, the preparation of new materials, and other fields.^{1–6} Under the supercritical conditions of high temperature and high pressure, dissolution experiments are characterized by the relatively difficult operation procedure, high cost, long time, and high manpower requirements and it is not easy to get solubility data.^{7,8} Therefore, it is of great significance to establish the precise dissolution prediction model.^{9,10}

The dissolution of ScCO₂ in polymers is influenced by many factors, such as temperature, pressure, density and the polarity of polymer molecules. These factors represent the extremely complex nonlinear relationships with the dissolution and are correlated with each other. Therefore, traditional thermodynamic state equation and empirical equation cannot provide the satisfactory prediction accuracy of the solubility.^{5,11,12} Artificial neural network (ANN) has the better self-organization, tolerance and nonlinear processing abilities, which make it especially suitable for solving the problem of solubility prediction.^{13–16} Bakhbakhi¹⁷ and Lashkarbolooki¹⁸ *et al.* compared the

solubility prediction results obtained with ANN and equations of state and indicated that ANN method was superior to the equations of state in the prediction accuracy and correlation. Gharagheizi *et al.*¹⁹ predicted the solubility of various compounds in ScCO₂, indicating that the prediction accuracy and correlation of ANN were satisfactory. Eslamimanesh *et al.*²⁰ indicated that ANN had the superior prediction performance in the solubility experiment of ScCO₂. Pahlavanzadeh *et al.*²¹ predicted the solubility with ANN and Deshmukh–Mather method and indicated that ANN was superior to the traditional thermodynamic prediction methods.

Prediction reliability and accuracy of ANN depend on its training algorithm for the optimizing the model structure and parameters. Various optimization algorithms had been developed for ANN training. Evolution algorithms are the most widely used,²² including genetic algorithm,²³ simulated annealing algorithm, particle swarm optimization algorithm (PSO),^{24–26} and ant colony algorithm.²⁷ Liu *et al.*²⁸ used the PSO algorithm and online strategy to train the fuzzy neural network and successfully predicted the melt index. Lazzus *et al.*²⁹ precisely predicted the phase equilibrium data of ScCO₂ with PSO algorithm. The solubility prediction method of radial basis function artificial neural network and adaptive fuzzy nervous system proposed by Khajeh *et al.*³⁰ is superior to the traditional method in prediction accuracy and correlation. Hussain *et al.*³¹ proposed the mixed neural network solution calculation model by combining Kent–Eisenberg with ANN and realized the better

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prediction performance. Li *et al.*^{32–38} also proposed several dissolution prediction models by combining chaos theory and particle swarm algorithm with the clustering method, improved the ANN training algorithm, and obtained the better prediction accuracy and correlation.

The above research focuses on the improvement in the model method itself, but the essence of the dissolution process is not considered. In the static condition, the essence of dissolution is the diffusion process of the solute molecules adsorbed on the surface to the solvent. The diffusion process is a mass transfer phenomenon under the thermal motion of molecules. The mass transfer is caused by the difference in density or temperature of the two-phase medium. Molecules move from the medium phase with the larger density to the medium phase with the smaller density until the equilibrium state is reached. In the polymer/ScCO₂ two-phase system, due to the interface force, CO₂ molecules are adsorbed on the interfacial film layer and diffused to the lining of the interfacial membrane. CO₂ molecules into the membrane layer are dissolved in the polymer melt. At present, the PSO algorithm based on the migration idea is seldom used in practical problems, especially in the field of thermodynamics. In the study, we attempted to improve the PSO algorithm in two aspects. Firstly, particles in PSO algorithm are similar to the dissolved molecules in the dissolution process and the similarity is mainly reflected by endowing particles in algorithm with the related properties of the dissolved molecules. Secondly, the essence of dissolution is the mass transfer, which is closely correlated with the diffusion theory. Inspired by the diffusion theory, dissolution theory and particle evolution algorithm, we found that the trajectory of CO₂ molecules in the diffusion process could be simulated by the motion of particles in the evolution algorithm. Therefore, we integrated the diffusion theory with the improvement of PSO algorithm and effectively simulated the dissolution process of ScCO₂ in polymers. With the improved PSO algorithm based on diffusion theory, we performed ANN training and obtained the hybrid intelligent prediction model for the solubility of ScCO₂ in polymers.

2. Model theory

2.1 Particle swarm optimization

PSO algorithm is a group evolution algorithm put forward by Eberhart and Kennedy³⁹ based on the social behaviors of birds. In the iterative process of PSO algorithm, it is assumed that in an n -dimensional search space, the particle population $X = (x_1, x_2, \dots, x_n)^T$ is composed of m particles. The position of the i th particle is denoted as $x_i = (x_{i,1}, x_{i,2}, \dots, x_{i,n})^T$ and the velocity of the i th particle is expressed as $v_i = (v_{i,1}, v_{i,2}, \dots, v_{i,n})^T$. Particles use individual extremum and global extremum to save the particle's own best position and particle population's best position. The formula is expressed as follows:

$$v_{i,d}^{k+1} = \omega v_{i,d}^k + c_1(p_{i,d}^k - x_{i,d}^k) + c_2(p_{g,d}^k - x_{i,d}^k) \quad (1)$$

$$x_{i,d}^{k+1} = x_{i,d}^k + v_{i,d}^{k+1} \quad (2)$$

where $p_i = (p_{i,1}, p_{i,2}, \dots, p_{i,n})^T$ is the individual extremum of a particle; $p_g = (p_{g,1}, p_{g,2}, \dots, p_{g,n})^T$ is the global extremum of particle population; ω is called inertial weight factor reflecting the property that particles maintain their own movement inertia; c_1 and c_2 are learning factors $\text{rand}()$ is a random number within (0,1); $V_{i,d}^k$ and $X_{i,d}^k$ are velocity and position of particle i in d -dimensional k th iteration; $P_{i,d}^k$ is individual extremum of particle i in d -dimensional solution space; $P_{g,d}^k$ is global extremum of the whole population.

2.2 Double-population particle swarm optimization (PSO) algorithm based on the diffusion theory

In PSO, particle is designed as a potential solution in search space. A particle updates its speed and position through an iterative formula. In the ScCO₂/polymer solution system, there are two-phase molecular systems of ScCO₂ and polymer melt. According to the diffusion theory, the energy indicating the process that molecules overcome their barrier to diffuse from their original positions to other positions is called diffusion energy. In other words, the molecules migrate through the diffusion energy and achieve the mass transfer. The magnitude of the diffusion energy is proportional to the square of the movement velocity of the molecule. However, the speed mainly depends on the temperature of the system. For molecule systems, temperature is a statistic index representing the average kinetic energy of the molecules and it is related to the velocity of the molecules. The faster the molecules move, the higher the temperature is.

Inspired by the nature of thermodynamics diffusion and dissolution, in this paper, a double population particle swarm algorithm based on diffusion theory (hereafter referred to as DP-DT-PSO algorithm for short) is proposed. In the DP-DT-PSO algorithm, the velocity of the molecule is substituted for the velocity of the particle, thus simulating the effect of the molecular force field. The diffusion temperature is a statistic index collectively representing the thermal motion of a large number of molecules. In the study on dissolution behaviors, temperature is one of the key influencing factors. It is assumed that there is a population temperature in the particle system reflecting the average temperature of the molecular system during the diffusion process. Before the discussion on the algorithm, several concepts of particle diffusion energy, center of mass of a population, distance of center of mass, and diffusion probability are introduced.

2.2.1 Basic concepts and definitions in the algorithm

(1) *Particle diffusion energy.* Objects in motion have kinetic energy. The magnitude of the kinetic energy E is related to the mass and velocity of the object. The calculation formula is provided as follows:

$$E = \frac{1}{2}mv^2 \quad (3)$$

where m and v are respectively the mass and velocity of the object. According to the molecular kinetic theory, molecules generate kinetic energy due to high-speed motion. According to the diffusion theory, if the active energy of a molecule can



overcome its energy barrier, the molecule can be diffused from the original position to a new position, this energy is called diffusion energy. In the PSO algorithm, in the particle population of n -dimensional space, it is assumed that all of the particles have a mass of 1. Then, the particle diffusion energy Q_i is defined as:

$$Q_i = \frac{1}{2} \sum_{j=1}^n V_{ij}^2 \quad (4)$$

where V_{ij} is the velocity of the particle. In the execution of the algorithm, the velocity of the particle is dynamically adjusted based on the velocity calculation formula through each iteration.

(2) *Center of mass of a population.* The algorithm converges with the probability of 100% when particles are moving toward a center.⁴⁰ In order to ensure the convergence of the algorithm, it is assumed that there is a center in the whole population and that all the particles travel to the center in a certain way. Center of mass of a population is a hypothetical point of a population and all the particles in the population approximate the center of mass of a population. The center of mass of a population X_{cen} is defined as follows:

$$X_{\text{cen}} = \frac{\sum x_i m_i}{\sum m_i} \quad (5)$$

where m is population size; x_i is the position of particle i . The mass of all the particles is assumed to be 1, so the center of mass of a population is also the statistical index of all the particles in the population.

(3) *Distance of center of mass.* In order to describe the positional relationship between the particle and the center of mass of a population, the distance of center of mass d_i is defined as the distance between the particle and the center of mass of a population:

$$d_i = |x_i - X_{\text{cen}}| \quad (6)$$

(4) *Particle diffusion probability.* When a particle meets the diffusion conditions, it does not necessarily generate effective diffusion. The diffusion probability of a particle reflects its diffusion capability or efficiency. The diffusion probability of particles is related to the diffusion energy and the temperature of the population. The diffusion probability P_i is defined as:

$$P_i = \frac{1 - e^{-\frac{Q_i}{T}}}{\Delta d} \quad (7)$$

where T is the population temperature. From a microscopic point of view, temperature has a decisive effect on the motion of particles, but the temperature analysis of a single particle has no practical significance. In the population of particles, temperature reflects the average kinetic energy of particles and the population temperature is the actual temperature of the system. When the diffusion energy meets the diffusion requirement, in order to indicate the tendency that the particles near the interface are prone to diffuse, Δd is used to denote the

distance difference between particle and centers of mass of two populations:

$$\Delta d = ||x_i - X_{\text{cen}}^A| - |x_i - X_{\text{cen}}^B|| \quad (8)$$

In the same temperature system, the diffusion probability of particles is determined by the diffusion energy of particles and the distance difference among the populations. If the higher diffusion energy of particles and the smaller distance difference mean the large diffusion probability of particles.

2.2.2 Thought and flow of the DP-DT-PSO algorithm. The diffusion system involves at least two phases. There are gas phase and melt phase in the ScCO_2 /polymer solution system. To this end, in the DP-DT-PSO algorithm, two populations representing the gas phase and melt phase are selected to simulate the diffusion system. Populations A and B are used to indicate gas molecule population in the supercritical phase and the molecule population in the polymer melt phase. The two populations are basically the same in implementation and function in the algorithm.

In the iteration of DP-DT-PSO, the diffusion energy of particles can be firstly obtained by calculating the velocity of each particle in population A (or B). According to the formula of diffusion probability, the diffusion probability value of each particle is calculated. For population A (or B), if the diffusion probability of particles is larger, then the particle is copied to the candidate population of population A (or B). We select the particle with the maximum diffusion probability from the candidate population of population A (or B) to replace the particle with the worst adaptive value in population B (or A). In this way, the sharing of information is realized. Finally, the global extreme value is updated. The implementation process of DP-DT-PSO algorithm is described as follows:

Step 1: initialization of the algorithm.

Two populations (A and B) are initialized to set the number of particles in the two populations, population specification, the position and velocity of all the particles, *etc.*;

Step 2: calculation of adaptive value.

The adaptive values of all particles are calculated according to the adaptive value function and evaluate the current performance of each particle;

Step 3: extremum updating.

To update the individual extremum of all particles and the global extreme of the two populations;

Step 4: condition judgment.

To determine whether the convergence accuracy meets the requirements or the maximum number of iterations is achieved. If the requirements are met or the maximum number of iterations is achieved, then go to Step 11; otherwise, go to Step 5.

Step 5: calculation of the diffusion probability of particles.

To calculate the diffusion probability P_i of all particles;

Step 6: generation of the candidate population.

For all the particles in population A, the following operation is performed:

For ($i = 1$; $i < M$; $i++$){if ($P_i > \text{rand}()$) copy the particle i to the candidate population T_A };

For population A, the following operation is performed:



For ($i = 1; i < M; i++$){if ($P_i > \text{rand}()$) copy the particle i to the candidate population T_B };

Step 7: information sharing among populations for simulation particle diffusion.

For population A, the particle with maximum diffusion probability is selected from the candidate population T_A to replace the particle with the worst adaptation value in population B. For population B, the particle with maximum diffusion probability is selected from the candidate population T_B to replace the particle with the worst adaptation value in population A. If the number of particles in the candidate population is insufficient, return the actual number;

Step 8: global extreme updating.

If the adaptive value of the particles returned from the candidate population is better than the global extremum of the population, the returned particles are used as global extremum, otherwise it is not updated;

Step 9: to save the results obtained in the process.

Return parameters such as the global extremum and local extremum of populations A and B in this iteration, and save them;

Step 10: iterative updating.

Update the formula according to velocity and location, update the velocity and position of the particles in population A/B;

Step 11: to save the results.

To determine whether the results are met. If the results are met, save the results, otherwise jump to Step 2.

2.3 Artificial neural network model based on DP-DT-PSO algorithm

Radial basis function artificial neural network (RBF ANN) is one of the most popular network models and consists of an input layer, an implicit layer, and an output layer. The activation function is a Gaussian function and defined as follows:

$$g_i(x_k) = \exp\left(-\frac{\|x_k - c_i\|^2}{\sigma_i^2}\right) \quad (9)$$

where x_k ($1 \leq k \leq n$) is the k th output vector; c_i ($1 \leq i \leq c$) is the basis function center; σ_i is the extension constant; n is the number of samples; c is the number of implied nodes, and the network output is:

$$O(x_k) = \sum_{i=1}^c w_i g_i(x_k) \quad (10)$$

where w_i is the connection weight of the i th implicit node.

The training process of RBF ANN is mainly to optimize the basis function center, the expansion constant and the connection weights. Through iteration of the three parameters c_i, σ_i, w_i , the network is trained. The output of RBF ANN network can be defined as:

$$y = f(w_{h,o}, \sigma_{h,o}, c_i) \quad (11)$$

where $w_{h,o}$ and $\sigma_{h,o}$ ($1 \leq h \leq c$), ($1 \leq o \leq p$) respectively represent the connection weights and extension constants between the

implicit node and the output node; c_i is the basis function center; p is the number of output nodes. When the DP-DT-PSO algorithm is used to train RBF ANN, the particle structure is defined as follows:

$$p(i) = [w_{h,o}, \sigma_{h,o}, c_i] \quad (12)$$

After RBF ANN is trained by DP-DT-PSO algorithm RBF ANN, a prediction model (DP-DT-PSO RBF ANN) is obtained for the subsequent solution prediction experiment.

3. Model building

3.1 Experimental data

In this paper, 8 kinds of polymers commonly used in industrial production are explored and 327 sets of related data constitute the model database. The source statistics are provided in Table 1.

According to the types of polymer, the database is subdivided three sets: the training set, validation set and test set. The data of training set are used to train the model. The purpose of the training is to learn all the data, find out the rules among the sample data, and save the rules with parameters such as weights and bias. The data of the validation set is used to verify and subtly correct the trained model so that the model is more accurate. The data of the test set is used to test the predicted performance of the model. The test results directly reflect the advantages and disadvantages of model. For the purpose of more effectively utilizing the data, the data proportions of training set, validation set and test set are respectively 70%, 15%, and 15%. Table 2 shows the data distribution statistics of various polymers.

3.2 Model evaluation indexes

In this paper, measures the prediction performance of the model is analyzed from the three aspects of prediction accuracy, efficiency and stability. In order to evaluate the prediction accuracy and stability, we adopted the three indexes: average relative deviation (ARD), root mean square error of prediction (RMSEP) and squared correlation coefficient (R^2) evaluation, which are respectively defined as:

$$\text{ARD} = \frac{1}{N} \sum_{i=1}^N \frac{|\bar{y}_i - y_i|}{y_i} \quad (13)$$

$$\text{RMSEP} = \sqrt{\frac{1}{N} \sum_{i=1}^N (y_i - \bar{y}_i)^2} \quad (14)$$

$$R^2 = \frac{\left[\sum_{i=1}^N (y_i - y_{\text{ave}})(\bar{y}_i - \bar{y}_{\text{ave}}) \right]^2}{\sum_{i=1}^N (y_i - y_{\text{ave}})^2 \sum_{i=1}^N (\bar{y}_i - \bar{y}_{\text{ave}})^2} \quad (15)$$

where N is the number of data sample; \bar{y}_i and \bar{y}_{ave} respectively are the predicted value and predicted mean; y_i and y_{ave} are



Table 1 Experimental data in this work

Polymer	T (K)	P (MPa)	Solubility (g g^{-1})	Data points	References
PP ^a	313.20–483.70	7.400–24.910	0.03950–0.26170	67	41–44
PLLA ^b	308.00–323.00	9.620–31.460	0.16520–0.43010	27	45
HDPE ^c	433.15–473.20	10.731–18.123	0.00551–0.12296	20	41 and 42
CPEs ^d	306.00–344.00	10.150–31.020	0.09840–0.63660	56	46
PS ^e	338.22–473.15	7.540–44.410	0.02641–0.16056	70	41 and 47–49
PBS ^f	323.15–453.15	8.008–20.144	0.04534–0.17610	31	41 and 50
PBSA ^g	323.15–453.16	7.870–20.128	0.04763–0.17411	29	41 and 50
PLGA ^h	308.00–323.00	10.140–31.470	0.09030–0.29630	27	45
Total	306.00–483.70	7.400–44.410	0.00551–0.63660	327	

^a Polypropylene. ^b Poly(L-lactide). ^c High-density polyethylene. ^d Carboxylated polyesters. ^e Polystyrene. ^f Poly(butylene succinate). ^g Poly(butylene succinate-co-adipate). ^h Poly(D,L-lactide-co-glycolide).

Table 2 Experimental data point distribution

Polymer	Training set	Validation set	Testing set	Total
PP	47	10	10	67
PLLA	19	4	4	27
HDPE	14	3	3	20
CPEs	38	9	9	56
PS	50	10	10	70
PBS	21	5	5	31
PBSA	21	4	4	29
PLGA	19	4	4	27

respectively the experimental value and the mean experimental value.

3.3 Model structure

In the three-layer network structure of DP-DT-PSO RBF ANN model, the input layer consists of two nodes respectively indicating the temperature and pressure of the solution system. The number of output layer nodes is 1, representing the solubility of

ScCO₂ in the polymer. In the hidden layer, this paper adopts the probe method to optimize the number of hidden layer nodes. When the number of nodes is increased from 3 to 13, a total of 11 DP-DT-PSO RBF ANN models are obtained. Fig. 1 illustrates the relationship diagram of the number of hidden layer nodes and the training error (MSE).

In general, the structure with the relatively small error and high correlation coefficient is used as the optimal network structure. As shown in Fig. 1, the hidden layer containing seven neurons is the optimal network structure model.

4. Results and discussion

4.1 Prediction results of the DP-DT-PSO RBF ANN model

Through Matlab 2010a software programming, a 2-7-1 DP-DT-PSO RBF ANN model was established to predict the solubility of supercritical carbon dioxide in 8 polymers. Simulation experiments are performed in Windows 7 SP1 64-bit operating system with 4.00 GB of memory and Intel (R) Core™ i5-4460 processor. Fig. 2 shows the correlation between the predicted values and the experimental values in the training set.

The vertical distance between the predicted data points and the line indicates the size of the prediction error. The predicted data points are basically distributed in the vicinity of the line. In the data prediction with the training set, DP-DT-PSO RBF ANN model shows the good performance and the correlation between the predicted value and the experimental value is high. Fig. 3 shows the relationship between model prediction data points and experimental data. The vertical distance indicates that DP-DT-PSO RBF ANN model has high prediction accuracy and good correlation.

The results of the training set and validation set show that, the performance of model DP-DT-PSO RBF ANN is excellent. The prediction results of the training set indicate the model is well trained. The prediction results of the validation set confirm the reliability of the model. The trained and validated model is applied in the test set. Fig. 4 shows the relationship between the prediction data points and the experimental data points obtained with DP-DT-PSO RBF ANN model in the test set.

In the test set, the prediction data points of the DP-DT-PSO RBF ANN model are basically distributed in the vicinity of the

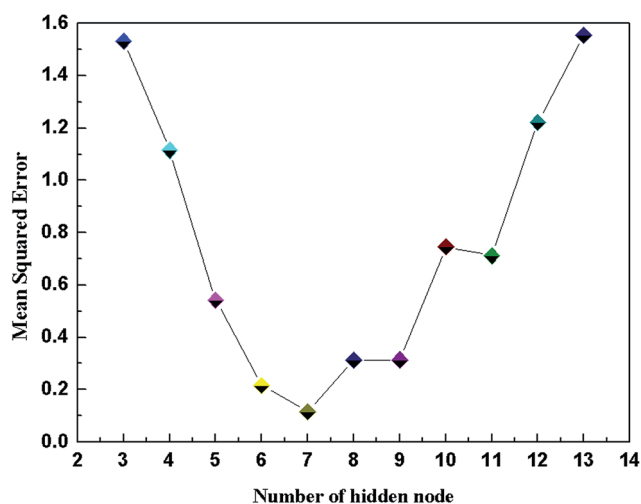


Fig. 1 Corrections between the number of hidden node and mean squared error.



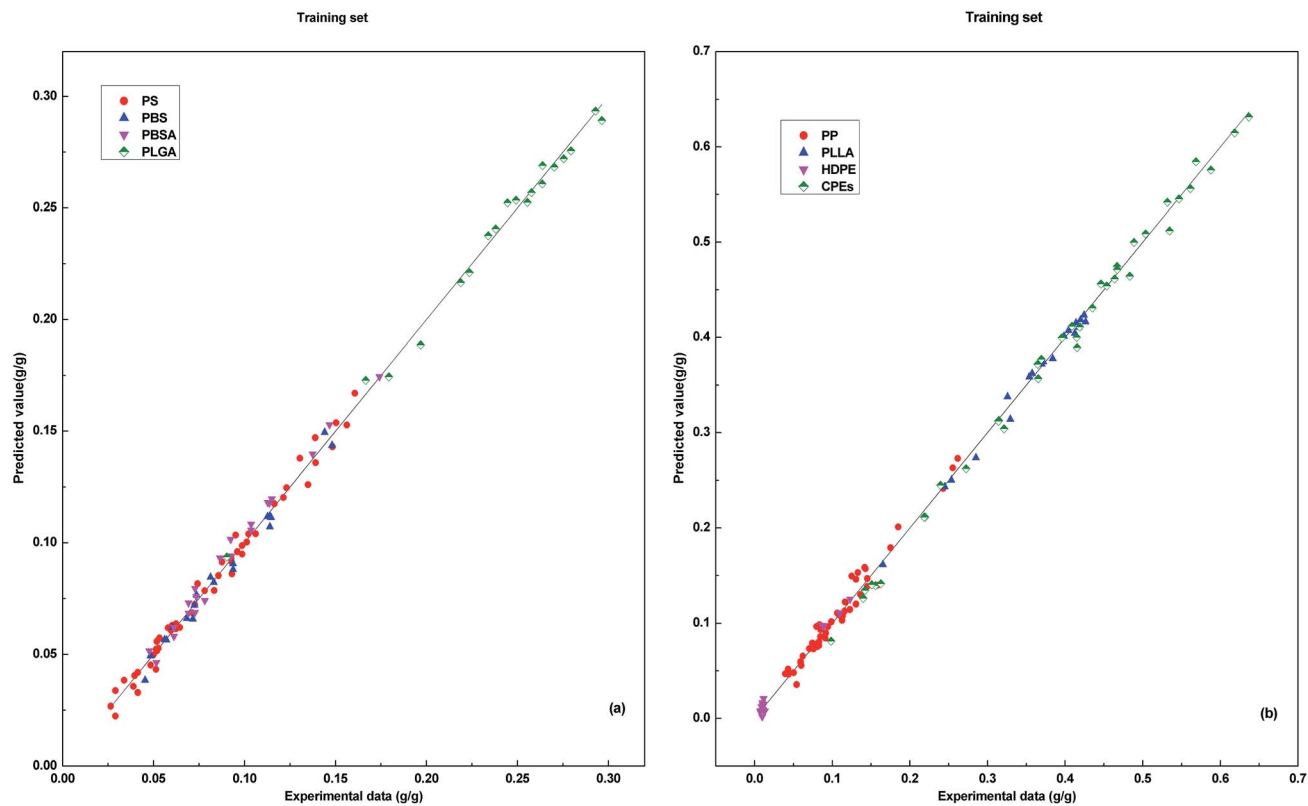


Fig. 2 Correlations between the prediction results and experimental data for training set.

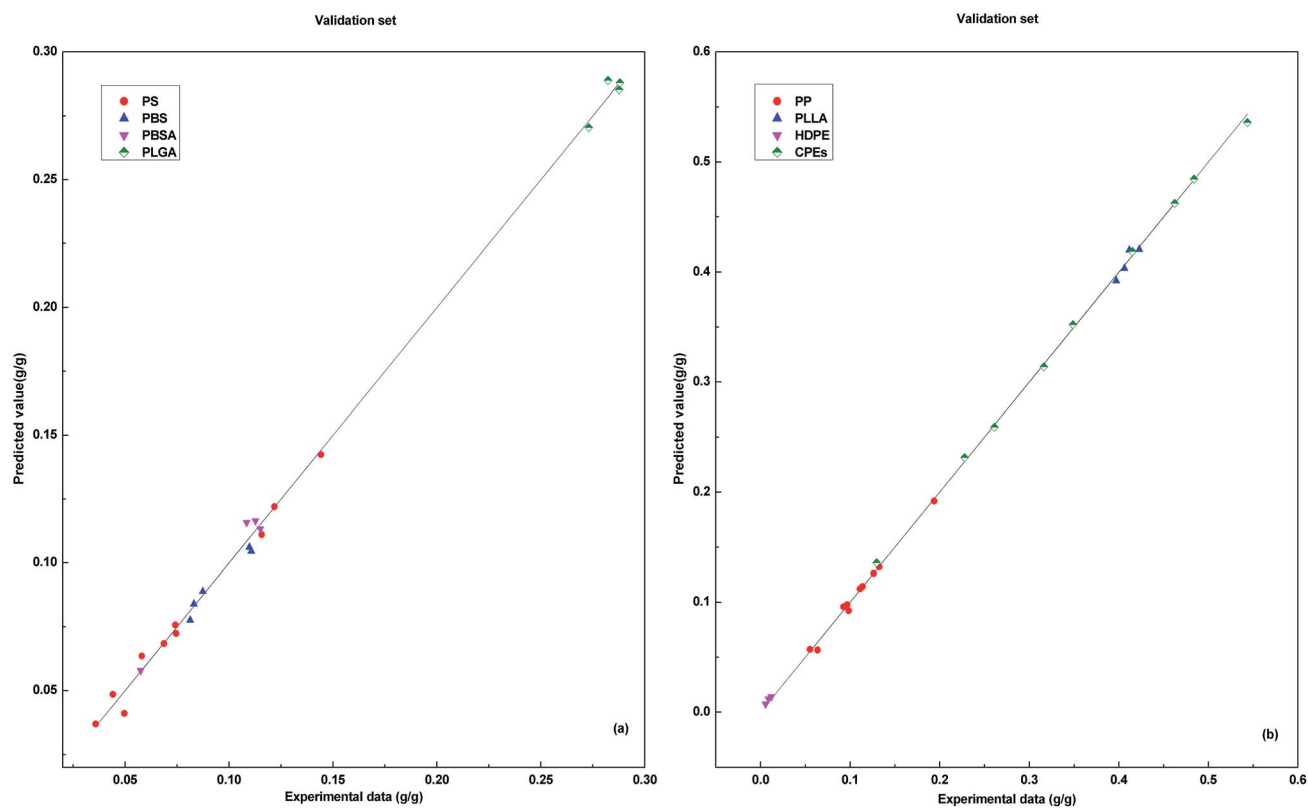


Fig. 3 Correlations between the prediction results and experimental data for validation set.

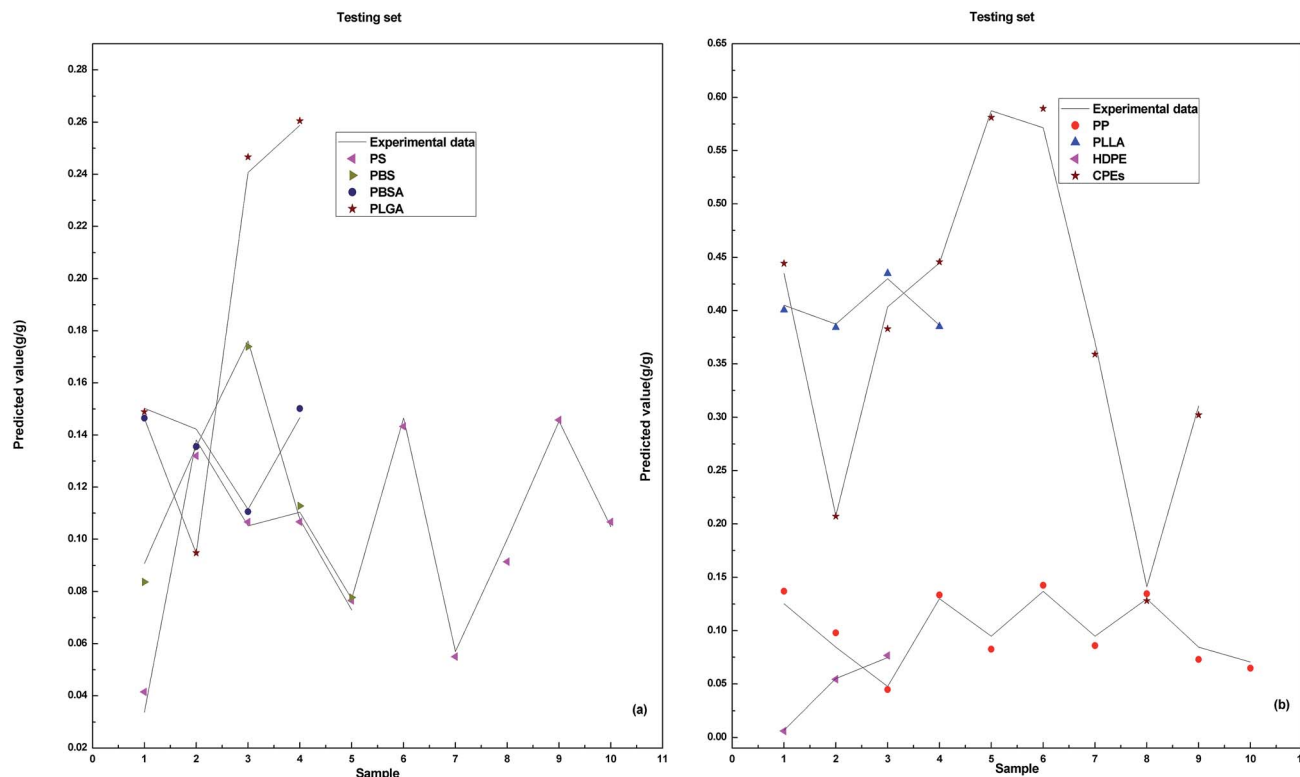


Fig. 4 Correlations between the predicted value and experimental data for testing set.

experimental data points, indicating that the predicted values are close to the experimental values. The distribution of data points of the test set is more dispersed than that of the training set and verification set, indicating that the prediction accuracy of the test set is slightly inferior. Table 3 shows the values of the prediction indexes for each data set.

DP-DT-PSO RBF ANN model has the superior performance (including accuracy and correlation) in the training set, validation set, and test set. In the three sets, the validation set shows the better performance: the high prediction accuracy and correlation. The statistical parameters of the model in the prediction experiment for each polymer are provided in Table 4. It is shown that the prediction accuracy of the model in each polymer is almost the same. The correlation between predictive value and experimental value is also better (>0.99). The statistical results verify the better comprehensive generalization ability of the model.

4.2 Comparative analysis with other models

The above experiments show the good performance of the model DP-DT-PSO RBF ANN in solution prediction. In order to evaluate the performance of DP-DT-PSO RBF ANN model and other models, the three common models, RBF ANN, PSO BP ANN and PSO RBF ANN were respectively used to perform the dissolution prediction for comparison. The running convergence curve of each model is shown in Fig. 5.

The DP-DT-PSO RBF ANN model tends to be stable at nearly 200 iterations, the PSO RBF ANN model tends to be stable at 400 iterations. PSO BP ANN is stable at about 300 iterations. The convergence precision of these models is decreased according to the order: DP-DT-PSO RBF ANN $>$ PSO BP ANN $>$ PSO RBF ANN $>$ RBF ANN. The convergence accuracy of DP-DT-PSO RBF ANN model is close to 0. From the comprehensive analysis of accuracy

Table 3 Values of ARD, R^2 , RMSEP of DP-DT-PSO RBF ANN for different subsets

Subset	ARD	R^2	RMSEP
Training subset	0.0038	0.9962	0.0153
Validation subset	0.0036	0.9967	0.0151
Testing subset	0.0043	0.9954	0.0161
Average value	0.0039	0.9961	0.0155

Table 4 Values of ARD, R^2 , RMSEP for different polymers

Polymer	ARD	R^2	RMSEP
PP	0.0025	0.9962	0.0158
PLLA	0.0039	0.9959	0.0134
HDPE	0.0048	0.9961	0.0138
CPEs	0.0038	0.9969	0.0161
PS	0.0039	0.9957	0.0145
PBS	0.0048	0.9961	0.0156
PBSA	0.0039	0.9957	0.0214
PLGA	0.0035	0.9961	0.0134

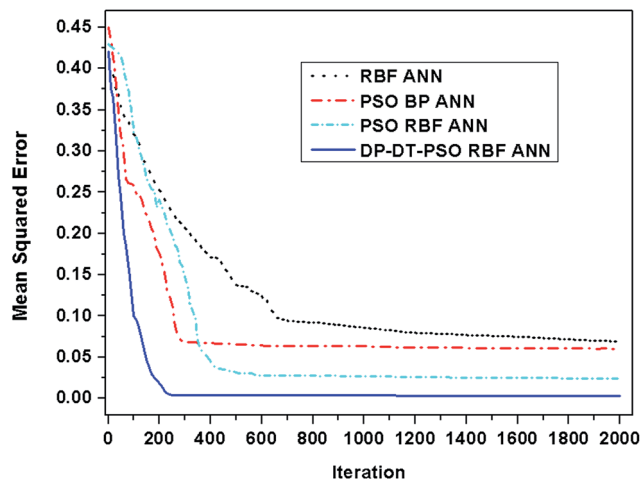


Fig. 5 Curve of mean squared error vs. iteration.

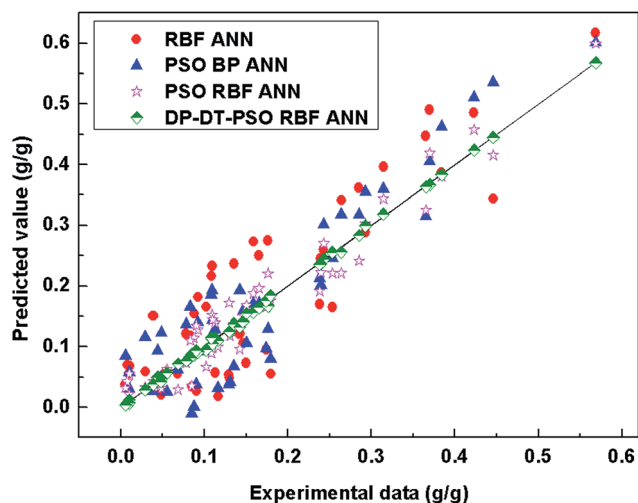


Fig. 6 Predicted values vs. experimental data.

and speed, the performance of DP-DT-PSO RBF ANN model is the best. Fig. 6 shows the relationship between the prediction data points and the experimental data points of each compare modes.

Table 6 Computation time of the comparison models

Model	Computation time (s)
RBF ANN	36
PSO BP ANN	41
PSO RBF ANN	43
This paper	56

The distribution of prediction data points shows significant differences among these models. The prediction data points of DP-DT-PSO RBF ANN model are the closest to the straight, indicating the high consistency with the experimental data. No abnormal prediction data point is found, indicating the better stability. The distance between the prediction data points and the straight line indicates that the accuracy of DP-DT-PSO RBF ANN model is also higher. Table 5 shows the data of the evaluation indexes for each model in the solubility prediction for various polymers.

Table 5 shows the prediction statistics of each model, including the data of accuracy and correlation. The model DP-DT-PSO RBF ANN has a better comprehensive performance including precision and correlation. Table 6 provides the average computation time required for 5 runs of each model.

The computation time of RBF ANN model is the shortest. The computation time of DP-DT-PSO RBF ANN model is longer. Intelligent algorithm intervention is bound to consume more computation time. In addition, the diffusion theory is introduced into the DP-DT-PSO RBF ANN model for improving the algorithm. In each iteration, the thermodynamic parameters of the system will be recalculated. In addition, the intelligent algorithm itself belongs to the optimization method, and also requires the longer computation time. Its computation time is within the acceptable range.

4.3 Result analysis

In terms of efficiency, accuracy and correlation, the results show that the DP-DT-PSO RBF ANN model presents a good performances for solubility prediction with high accuracy and good

Table 5 Values of ARD, R^2 , RMSEP of the comparison models

	Model	PP	PLLA	HDPE	CPEs	PS	PBS	PBSA	PLGA	Average
ARD	RBF ANN	0.0098	0.0094	0.0088	0.0092	0.0093	0.0092	0.0091	0.0087	0.0092
	PSO BP ANN	0.0052	0.0072	0.0082	0.0077	0.0056	0.0082	0.0089	0.0065	0.0072
	PSO RBF ANN	0.0046	0.0055	0.0054	0.0046	0.0054	0.0066	0.0061	0.0056	0.0055
	This paper	0.0025	0.0039	0.0048	0.0038	0.0039	0.0048	0.0039	0.0035	0.0039
R^2	RBF ANN	0.9643	0.9643	0.9651	0.9614	0.9612	0.9598	0.9601	0.9618	0.9623
	PSO BP ANN	0.9813	0.9815	0.9816	0.9812	0.9821	0.9912	0.9811	0.9834	0.9829
	PSO RBF ANN	0.9885	0.9882	0.9863	0.9817	0.9811	0.9812	0.9817	0.9821	0.9839
	This paper	0.9962	0.9959	0.9961	0.9969	0.9957	0.9961	0.9957	0.9961	0.9961
RMSEP	RBF ANN	0.0579	0.0615	0.0721	0.0852	0.0789	0.0566	0.0796	0.0886	0.0726
	PSO BP ANN	0.0335	0.0345	0.0368	0.0369	0.0389	0.0411	0.0485	0.0512	0.0402
	PSO RBF ANN	0.0312	0.0352	0.0362	0.0338	0.0412	0.0431	0.0458	0.0437	0.0388
	This paper	0.0158	0.0134	0.0138	0.0161	0.0145	0.0156	0.0214	0.0134	0.0155



correlations, the predicted data are agree with the experimental values. Fig. 7 depicts the corrections between predicted values and experimental data under the different temperatures and pressure.

As we can see from Fig. 7, the solubility of HDPE, PS, PP, CPES, PBS and PBSA is proportional to the pressure and inversely proportional to the temperature. As shown in Fig. 7(b) and (i), with the increase in the pressure, the solubility of the

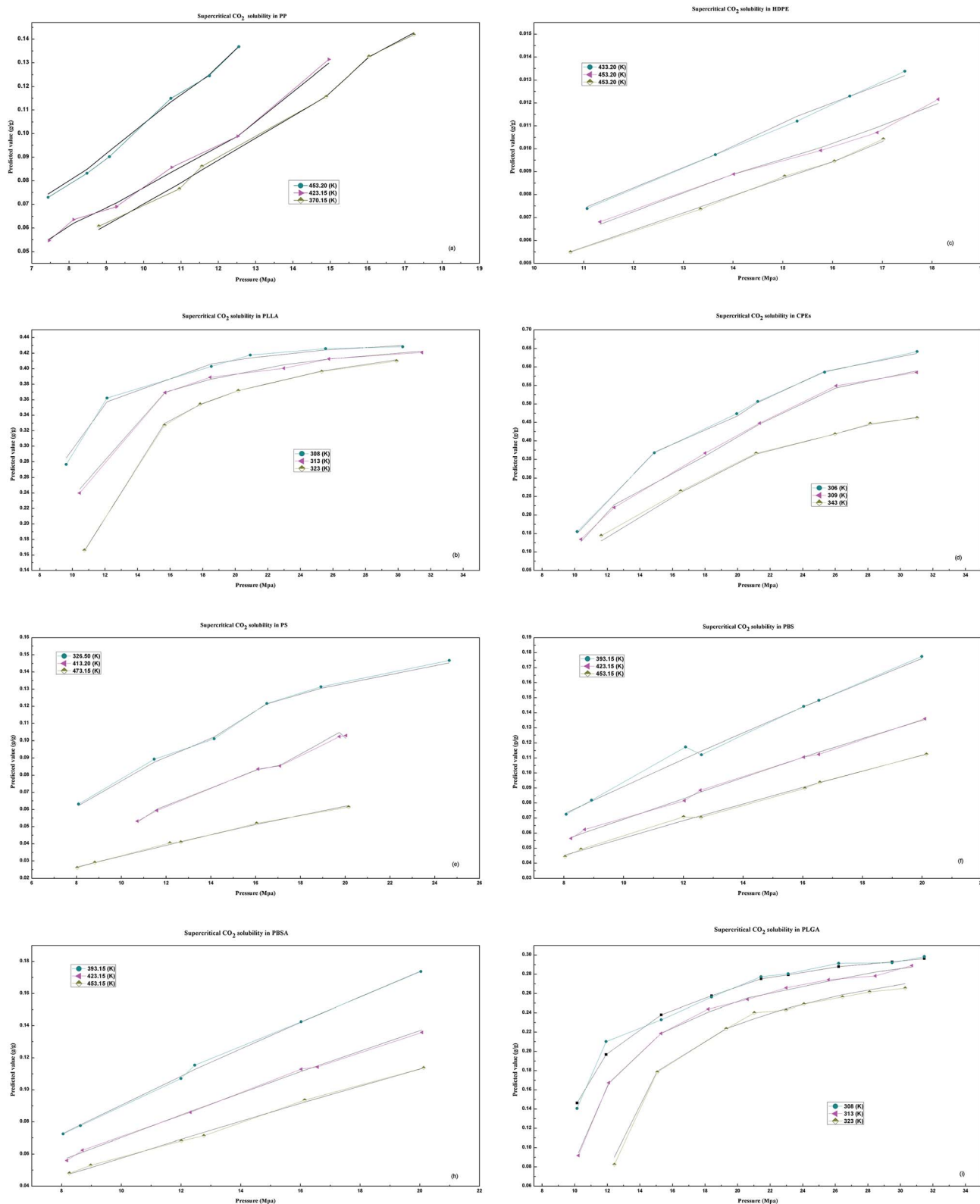


Fig. 7 Variation of solubility with pressure and temperature.



two polymers PLLA and PLGA increases firstly and then stabilizes and decreases with the increase in the temperature. The solubility prediction trend is concordant with the experimental.

5. Conclusions

(1) The prediction performance of DP-DT-PSO RBF ANN model is good.

Based on the theory of diffusion, dissolution and particle evolution, an improved evolutionary algorithm based on diffusion theory is proposed in this paper to train the radial artificial neural network, so that the solution prediction model is obtained. A good prediction performance of DP-DT-PSO RBF ANN model is demonstrated by predicting the solubility examples of ScCO₂ in 8 polymers.

(2) The expandability of DP-DT-PSO RBF ANN model is better.

By dissolubility prediction, this paper verifies the advantages of DP-DT-PSO RBF ANN model. The model can be extended to prediction fields of other chemical and physical properties. It can be applied to experimental data processing, control and optimization of experimental parameters. The scalability is better.

(3) DP-DT-PSO RBF ANN model realizes the solubility prediction by training experimental data and obtaining relevant rules. In the future research, the essence of diffusion and dissolution can be further discussed and the theoretical calculation method will be put forward. We will pay more attention to the development of this field and study the theoretical calculation method with the higher efficiency and better performance.

Conflicts of interest

There are no conflicts to declare.

Abbreviations

PVT	Pressure, volume, temperature
ANN	Artificial neural network
RBF	Radial basis function
PSO	Particle swarm optimization
DP-DT-PSO	Double population particle swarm algorithm based on diffusion theory
PBS	Poly(butylene succinate)
PBSA	Poly(butylene succinate-co-adipate)
PP	Polypropylene
PS	Polystyrene
CPEs	Carboxylated polyesters
PLLA	Poly(L-lactide)
PLGA	Poly(D,L-lactide-co-glycolide)
HDPE	High-density polyethylene
ARD	Average relative deviation
R ²	Squared correlation coefficient
MSE	Mean square error
RMSEP	Root mean square error of prediction

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References

- 1 A. A. Cockram, T. J. Neal, M. J. Derry, O. O. Mykhaylyk, N. S. J. Williams, M. W. Murray, S. N. Emmett and S. P. Armes, *Macromolecules*, 2017, **50**, 796–802.
- 2 F. N. Azad, M. Ghaedi, A. Asfaram, A. Jamshidi, G. Hassani, A. Goudarzi, M. H. A. Azghandi and A. Ghaedi, *RSC Adv.*, 2016, **6**, 19768–19779.
- 3 M. T. Jacobsen, M. E. Petersen, X. Ye, M. Galibert, G. H. Lorimer, V. Aucagne and M. S. Kay, *J. Am. Chem. Soc.*, 2016, **138**, 11775–11782.
- 4 M. Li, W. Wu, B. Chen, Y. Wu and X. Huang, *RSC Adv.*, 2017, **7**, 35274–35282.
- 5 H. Ziaee, S. M. Hosseini, A. Sharafpoor, M. Fazavi, M. M. Ghiasi and A. Bahadori, *J. Taiwan Inst. Chem. Eng.*, 2015, **46**, 205–213.
- 6 Y. S. Zhao, J. B. Gao, Y. Huang, R. M. Afzal, X. P. Zhang and S. J. Zhang, *RSC Adv.*, 2016, **6**, 70405–70413.
- 7 J. J. van Franeker, G. H. L. Heintges, C. Schaefer, G. Portale, W. Li, M. M. Wienk, P. van der Schoot and R. A. J. Janssen, *J. Am. Chem. Soc.*, 2015, **137**, 11783–11794.
- 8 J. A. Lazzús, F. Cuturrufo, G. Pulgar-Villarroel, I. Salfate and P. Vega, *Ind. Eng. Chem. Res.*, 2017, **56**, 6869–6886.
- 9 S. Arefi-Oskoui, A. Khataee and V. Vatanpour, *ACS Comb. Sci.*, 2017, **19**, 464–477.
- 10 J. R. Kramer and T. J. Deming, *J. Am. Chem. Soc.*, 2014, **136**, 5547–5550.
- 11 R. Galvelis and Y. Sugita, *J. Chem. Theory Comput.*, 2017, **13**, 2489–2500.
- 12 Y. Xu, D. Koo, E. A. Gerstein and C.-S. Kim, *Polymer*, 2016, **84**, 121–131.
- 13 L. P. Barron and G. L. McEneff, *Talanta*, 2016, **147**, 261–270.
- 14 H. T. Liu, Y. W. Wang, W. D. Huang and M. Lei, *Environ. Sci. Pollut. Res.*, 2016, **23**, 24230–24236.
- 15 X. Wang, W. Cai, J. Lu and Y. Sun, *Ind. Eng. Chem. Res.*, 2014, **53**, 19293–19303.
- 16 A. Esmacili, E. Hejazi and Y. Vasseghian, *RSC Adv.*, 2015, **5**, 91776–91784.
- 17 Y. Bakhbakhi, *Math. Comput. Model.*, 2012, **55**, 1932–1941.
- 18 A. Z. Hezave, S. Raeissi and M. Lashkarbolooki, *Ind. Eng. Chem. Res.*, 2012, **51**, 9886–9893.
- 19 F. Gharagheizi, A. Eslamimanesh, A. H. Mohammadi and D. Richon, *Ind. Eng. Chem. Res.*, 2011, **50**, 221–226.
- 20 A. Eslamimanesh, F. Gharagheizi, A. H. Mohammadi and D. Richon, *Chem. Eng. Sci.*, 2011, **66**, 3039–3044.
- 21 H. Pahlavan-zadeh, S. Nourani and M. Saber, *J. Chem. Thermodyn.*, 2011, **43**, 1775–1783.
- 22 D. Granato, M. D. Carrapeiro, V. Fogliano and S. M. van Ruth, *Trends Food Sci. Technol.*, 2016, **52**, 31–48.
- 23 A. S. Ghareb, A. Abu Bakar and A. R. Hamdan, *Expert Syst. Appl.*, 2016, **49**, 31–47.



- 24 R. Xia, X. Huang and M. Li, *J. Appl. Polym. Sci.*, 2016, **133**, 44252.
- 25 M. Tang, C.-E. Hu, Z.-L. Lv, X.-R. Chen and L.-C. Cai, *J. Phys. Chem. A*, 2016, **120**, 9489–9499.
- 26 C.-P. Chou, Y. Nishimura, C.-C. Fan, G. Mazur, S. Irle and H. A. Witek, *J. Chem. Theory Comput.*, 2016, **12**, 53–64.
- 27 M. Saidi-Mehrabadi, S. Dehnavi-Arani, F. Evazabadian and V. Mahmoodian, *Comput. Ind. Eng.*, 2015, **86**, 2–13.
- 28 X. G. Liu and C. Y. Zhao, *AIChE J.*, 2012, **58**, 1194–1202.
- 29 J. A. Lazzus, A. Ponce and L. Chilla, *Fluid Phase Equilib.*, 2012, **317**, 132–139.
- 30 A. Khajeh and H. Modarress, *Expert Syst. Appl.*, 2010, **37**, 3070–3074.
- 31 M. A. Hussain, M. K. Aroua, C. Y. Yin, R. A. Rahman and N. A. Ramli, *Korean J. Chem. Eng.*, 2010, **27**, 1864–1867.
- 32 M. S. Li, X. Y. Huang, H. S. Liu, B. X. Liu and Y. Wu, *J. Appl. Polym. Sci.*, 2013, **130**, 3825–3832.
- 33 M. S. Li, X. Y. Huang, H. S. Liu, B. X. Liu, Y. Wu and F. R. Ai, *Acta Chimica Sinica*, 2013, **71**, 1053–1058.
- 34 Y. Wu, B. X. Liu, M. S. Li, K. Z. Tang and Y. B. Wu, *Chin. J. Chem.*, 2013, **31**, 1564–1572.
- 35 M. S. Li, X. Y. Huang, H. S. Liu, B. X. Liu, Y. Wu and X. Z. Deng, *J. Appl. Polym. Sci.*, 2013, **129**, 3297–3303.
- 36 M. Li, X. Huang, H. Liu, B. Liu, Y. Wu and L. Wang, *RSC Adv.*, 2015, **5**, 45520–45527.
- 37 M. S. Li, X. Y. Huang, H. S. Liu, B. X. Liu, Y. Wu, A. H. Xiong and T. W. Dong, *Fluid Phase Equilib.*, 2013, **356**, 11–17.
- 38 X. Ru-Ting and H. Xing-Yuan, *RSC Adv.*, 2015, **5**, 76979–76986.
- 39 J. Kennedy and R. Eberhart, *Particle swarm optimization*, Perth, Aust, 1995, DOI: 10.1109/icnn.1995.488968.
- 40 X. L. Zhao, M. Turk, W. Li, K. C. Lien and G. Z. Wang, *Applied Soft Computing*, 2016, **48**, 151–159.
- 41 A. Khajeh, H. Modarress and M. Mohsen-Nia, *Iran. Polym. J.*, 2007, **16**, 759–768.
- 42 Y. Sato, K. Fujiwara, T. Takikawa, Sumarno, S. Takishima and H. Masuoka, *Fluid Phase Equilib.*, 1999, **162**, 261–276.
- 43 Z. G. Lei, H. Ohyabu, Y. Sato, H. Inomata and R. L. Smith, *J. Supercrit. Fluids*, 2007, **40**, 452–461.
- 44 D. C. Li, T. Liu, L. Zhao and W. K. Yuan, *Ind. Eng. Chem. Res.*, 2009, **48**, 7117–7124.
- 45 E. Aionicesei, M. Skerget and Z. Knez, *J. Supercrit. Fluids*, 2008, **47**, 296–301.
- 46 M. Skerget, Z. Mandzuka, E. Aionicesei, Z. Knez, R. Jese, B. Znoj and P. Venturini, *J. Supercrit. Fluids*, 2010, **51**, 306–311.
- 47 Y. Sato, M. Yurugi, K. Fujiwara, S. Takishima and H. Masuoka, *Fluid Phase Equilib.*, 1996, **125**, 129–138.
- 48 S. Hilic, S. Boyer, A. Padua and J. Grolier, *J. Polym. Sci., Part B: Polym. Phys.*, 2001, **39**, 2063–2070.
- 49 Y. Sato, T. Takikawa, S. Takishima and H. Masuoka, *J. Supercrit. Fluids*, 2001, **19**, 187–198.
- 50 Y. Sato, T. Takikawa, A. Sorakubo, S. Takishima, H. Masuoka and M. Imaizumi, *Ind. Eng. Chem. Res.*, 2000, **39**, 4813–4819.

