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Evaluating the electronic structure of formal Ln^{II} ions in Ln^{II}(C₅H₄SiMe₃)₃¹⁻ using XANES spectroscopy and DFT calculations†

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The isolation of [K(2.2.2-cryptand)][Ln(C₅H₄SiMe₃)₃], formally containing Ln^{II}, for all lanthanides (excluding *Pm*) was surprising given that +2 oxidation states are typically regarded as inaccessible for most 4f-elements. Herein, X-ray absorption near-edge spectroscopy (XANES), ground-state density functional theory (DFT), and transition dipole moment calculations are used to investigate the possibility that Ln(C₅H₄SiMe₃)₃¹⁻ (Ln = Pr, Nd, Sm, Gd, Tb, Dy, Y, Ho, Er, Tm, Yb and Lu) compounds represented molecular Ln^{II} complexes. Results from the ground-state DFT calculations were supported by additional calculations that utilized complete-active-space multi-configuration approach with second-order perturbation theoretical correction (CASPT2). Through comparisons with standards, Ln(C₅H₄SiMe₃)₃¹⁻ (Ln = Sm, Tm, Yb, Lu, Y) are determined to contain 4f⁶ 5d⁰ (Sm^{II}), 4f¹³ 5d⁰ (Tm^{II}), 4f¹⁴ 5d⁰ (Yb^{II}), 4f¹⁴ 5d¹ (Lu^{II}), and 4d¹ (Y^{II}) electronic configurations. Additionally, our results suggest that Ln(C₅H₄SiMe₃)₃¹⁻ (Ln = Pr, Nd, Gd, Tb, Dy, Ho, and Er) also contain Ln^{II} ions, but with 4f⁷ 5d¹ configurations (not 4fⁿ⁺¹ 5d⁰). In these 4f⁷ 5d¹ complexes, the C_{3h}-symmetric ligand environment provides a highly shielded 5d-orbital of a' symmetry that made the 4f⁷ 5d¹ electronic configurations lower in energy than the more typical 4fⁿ⁺¹ 5d⁰ configuration.

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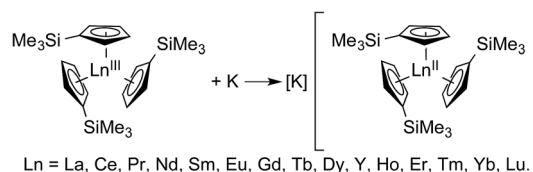
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Introduction

Recent advances in rare-earth metal reduction chemistry have revealed a surprisingly new series of molecular complexes that contained all the rare earth metals in the formal oxidation state of +2,¹ as defined by Parkin and Karen, (Scheme 1).^{2,3} These results were unexpected given that the +2 oxidation state had

only been observed with six rare earth metals in molecules (Eu, Yb, Sm, Tm, Dy, and Nd). Observing this +2 oxidation state for the other lanthanides was unexpected because the -2.7 to -3.9 V *versus* standard hydrogen electrode (SHE) Ln^{III}/Ln^{II} reduction potentials seemed too negative to allow Ln^{II} ions to exist in solution.⁴ In the solid state, only the six lanthanides listed above were known to form +2 salts. For the other metals, compounds like LnX₂ (Ln = La, Ce, Pr, Gd, and Y; X = halide) with formal +2 oxidation states had been observed, but subsequent analyses revealed that they contain +3 ions and a delocalized electron in a conduction band, *i.e.* Ln^{III}(X¹⁻)₂(e¹⁻).⁵

The new Ln(C₅H₄SiMe₃)₃¹⁻ compounds, containing the putative +2 ions, were synthesized by potassium reduction of



Scheme 1 A general reaction scheme for generating Ln^{II}(C₅H₄SiMe₃)₃¹⁻ containing salts. Accessing these compounds in crystalline form requires complexation of the potassium cation by 18-crown-6 or 2.2.2-cryptand.¹

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† Electronic supplementary information (ESI) available: The results from quantum chemical *ab initio* FEFF9.6 code calculations for Ln^{II} L₃-edges from Ln(C₅H₄SiMe₃)₃^{x-} (Ln = Sm, Tm, Y; x = 0, 1) and second derivative analysis of the lanthanides are in the ESI.† Our branching ratio analysis, the PBE//TZP optimized ground-state geometrical xyz coordinates and XC//TZ2P (XC = PBE, BLYP, B3LYP, BHandHLYP), single-point calculated total bonding energies of Ln(C₅H₄SiMe₃)₃^{1-/0} (Ln = Sm, Ho), and relative single-point energy difference in kcal mol⁻¹ between 4f¹⁰ 5d¹ and 4f¹¹ 5d⁰ in Ho^{II}(C₅H₄SiMe₃)₃¹⁻ from different functional results at the PBE//TZP optimized ground-state geometries are also included. See DOI: 10.1039/c7sc00825b

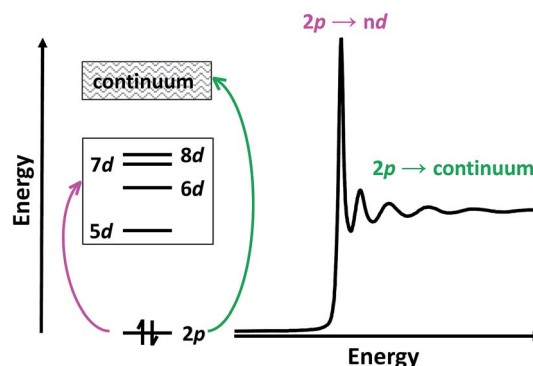
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trimethylsilylcyclopentadienyl lanthanide(III) complexes, $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (Scheme 1). More detailed synthetic descriptions for these $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ anions, as well as related $\text{Ln}[\text{C}_5\text{H}_3(\text{SiMe}_3)_2]_3^{1-}$ complexes prepared by Lappert and coworkers, have been previously discussed.⁶ The new $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ complexes were unusual in that their $\text{Ln}-\text{C}_{\text{centroid}}$ distances were only 1% (0.020–0.032 Å) longer than their Ln^{III} precursors, $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. Larger variations, by an order of magnitude (0.1 to 0.2 Å), were expected based on previous comparisons between conventional Ln^{II} versus Ln^{III} structures, which historically provided a diagnostic for the +2 oxidation state. Consistent with this traditional expectation, $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ bond lengths for $\text{Ln} = \text{Eu}, \text{Yb}, \text{Sm}, \text{and Tm}$ were 0.10–0.20 Å (~6%) longer than their +3 analogs.⁷ The unusually short bond lengths in the La, Ce, Pr, Nd, Gd, Tb, Dy, Y, Ho, Er, and Lu complexes led to skepticism about the presence of the +2 oxidation state across the $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ series, suggesting that the salts might contain +3 metals with an electron delocalized into ligand-based orbitals. This scenario was – in a sense – reminiscent of the LnX_2 compounds (discussed above).⁵ An alternative description, based on subsequent theoretical analyses, proposed that the small differences in bond distances for La, Ce, Pr, Nd, Gd, Tb, Dy, Y, Ho, Er, and Lu complexes were a direct result of the metal ions having an unusual $4f^n 5d^1$ electronic configuration, rather than the traditionally expected $4f^{n+1} 5d^0$ configuration known for $\text{Eu}^{\text{II}}, \text{Yb}^{\text{II}}, \text{Sm}^{\text{II}}, \text{and Tm}^{\text{II}}$.

Attempts have been made to validate the theoretical conclusions using electronic absorption spectroscopy and magnetic susceptibility.⁸ Although the UV-vis analyses showed intense bands that were consistent with the $4f^n 5d^1$ configurations, forbidden $4f \rightarrow 4f$ transitions typically used as diagnostics for lanthanide oxidation states were not experimentally resolved.^{1,5,9} Similarly, the magnetic studies showed complicated magnetic behavior that could not be ubiquitously rationalized for all the lanthanides using simple models.^{8d} For these reasons, it was of great interest to evaluate the electronic structure of the $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ complexes using a combination of X-ray absorption near-edge spectroscopy (XANES) and transition dipole moment density functional theory (DFT). There is an emerging body of literature demonstrating the power of cooperative XANES and DFT analyses in evaluating bonding and electronic structure in inorganic compounds.¹⁰ As such, we have recently used this approach to uniquely characterize the electronic structures of a wide variety of f-element species.¹¹

Herein, we describe the use of a combination of XANES and transition dipole moment DFT calculations to evaluate the possibility that the $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($\text{Ln} = \text{Pr}, \text{Nd}, \text{Sm}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Y}, \text{Ho}, \text{Er}, \text{Tm}, \text{Yb}$ and Lu) compounds represent molecular Ln^{II} complexes. In the XANES experiment, an analyte is exposed to high-energy X-rays that excite core electrons to higher, unoccupied states. At the $\text{Ln } L_{3,2}$ -edges, there is an edge-jump consisting of electric-dipole allowed transitions from $\text{Ln } 2p$ -orbitals to unoccupied states that contain metal d-character. Moving to higher energies, core electrons are excited into the continuum (Scheme 2). Given that $\text{Ln } L_{3,2}$ -edge XANES probes transitions to $\text{Ln } 5d$ -orbitals, this spectroscopic approach provides a particularly sensitive and accurate method for



Scheme 2 Cartoon depicting the origin of $L_{3,2}$ -edge XANES transitions.

directly characterizing $5d$ -orbital occupancies for the alleged $4f^n 5d^1$ ions in $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($\text{Ln} = \text{La}, \text{Ce}, \text{Pr}, \text{Nd}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{Er}, \text{and Lu}$) anions. To guide interpretations of these XANES spectra, appropriate ground-state DFT models were developed that formed a basis for extracting probability amplitudes from the transition dipole moments between the calculated excited-states and the ground-state. Combined, these computational and experimental efforts allow the influence of $4f^{n+1} 5d^0$ versus $4f^n 5d^1$ electronic configurations on the lanthanide $L_{3,2}$ -edge XANES spectra to be determined for the first time.

To best characterize the electronic structure of the $[\text{K}(2.2.2\text{-cryptand})][\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ salts containing new Ln^{II} ions, XANES and DFT studies are also reported with the compounds containing traditional +2 ions (*i.e.* $\text{Sm}^{\text{II}}, \text{Tm}^{\text{II}}, \text{and Yb}^{\text{II}}$) whose electronic configurations were well defined as $4f^6, 4f^{13}, \text{and } 4f^{14}$, respectively. These results provide a foundation for analyses of the other $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ anions. For comparison, studies of the neutral $4f^n 5d^0$ Ln^{III} complexes, $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$, are also reported because the metal oxidation state in these compounds is unambiguously +3. These combined efforts lead to a definitive description of the electronic structure and bonding in the $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ complexes. For the convenience of the reader in the rest of the paper, we refer to compounds with formal +3 oxidation states as $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and formal +2 oxidation states as $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$. When discussing both, the Roman numerals are omitted and $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$) is used.

Results

$\text{Sm } L_{3,2}$ -edge XANES

The background-subtracted and normalized $\text{Sm } L_{3,2}$ -edge XANES spectra from $[\text{K}(2.2.2\text{-cryptand})][\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ and $\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$ are shown in Fig. 1. Each spectrum contains large edge features near 6715 eV (L_3) and 7310 eV (L_2) and small post-edge shoulders near 6725 and 7320 eV that are superimposed on step-like absorption thresholds. The $L_{3,2}$ -edge positions were characterized by their peak maxima, where the first derivatives of the data equaled zero (Table 1). Given the sharp characteristics of these peaks, we find that the peak maximum provides a more useful metric than the inflection point, which is commonly used to evaluate actinide absorption edges. The $L_{3,2}$ -edge peak maxima for $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ at



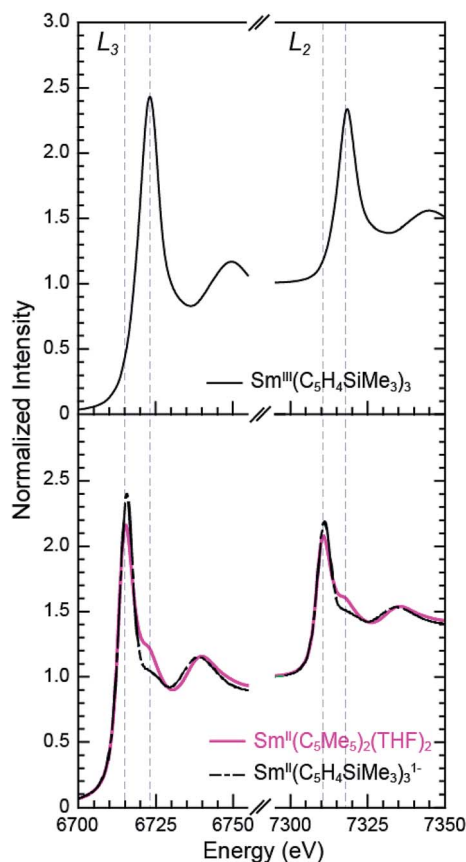


Fig. 1 The background-subtracted and normalized Sm $L_{3,2}$ -edge XANES spectra obtained from $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (top, black trace), $\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$ (bottom, pink trace), and $[\text{K}(2.2.2\text{-cryptand})][\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ (bottom, black dashed trace).

6715.6 and 7311.1 eV are nearly identical to the 6715.2 and 7310.7 eV values determined for $\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$ and similar to the other Sm^{II} $L_{3,2}$ -edge XANES spectra reported previously (Table 1).¹²

The Sm $L_{3,2}$ -edge XANES spectra obtained from $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and $\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$ are also compared with $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ in Fig. 1. The $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ $L_{3,2}$ -edge spectra differ from the Sm^{II} spectra in that the edge features are shifted by approximately 7–8 eV to higher energies at 6723.2 and 7318.4 eV, Table 1. The differences in edge-positions for $4f^6 5d^0$ (+2) and $4f^5 5d^0$ (+3) samarium species are not unique to this suite of samarium cyclopentadienyl compounds.^{10,13} For instance, the Sm L_3 -edge XANES spectra obtained from $\text{Sm}^{\text{II}}[\text{N}(\text{SiMe}_3)_2](\text{THF})_2$ and $\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_3$, also exhibit a Sm L_3 -edge energy difference of 7–8 eV (Fig. 2 and Table 1). These results demonstrate that samarium 4f-orbital occupancy ($4f^6 5d^0$ versus $4f^5 5d^0$) influences the peak position more substantially than the ligand identity, as changing cyclopentadienide in $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ to amido ligands in $\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_3$ only shifts the L_3 -edge peak maximum to lower energy by 0.4 eV.

Comparisons between the Sm^{II} and Sm^{III} spectra provide insight into the origin of the small post-edge shoulders near 6725 and 7320 eV observed in all of the Sm^{II} spectra. As shown by the dashed lines in Fig. 1 and 2, this post-edge feature

Table 1 Comparison of the peak maxima for $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$, $[\text{K}(2.2.2\text{-cryptand})][\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ ($\text{Ln} = \text{Pr}, \text{Nd}, \text{Sm}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Y}, \text{Ho}, \text{Er}, \text{Tm}, \text{Yb}$ and Lu), $\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$, $\text{Sm}^{\text{II}}[\text{N}(\text{SiMe}_3)_2](\text{THF})_2$, $\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_3$, $\text{TmI}_2(\text{THF})_3$, and $\text{TmI}_3(\text{THF})_{3.5}$. When possible, spectral differences between analogous Ln^{II} and Ln^{III} compounds have been included

Compound	Edge	Peak position (eV) ^a	$\Delta(\text{Ln}^{\text{III}}-\text{Ln}^{\text{II}})$ peak position (eV)
$\text{Pr}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_2	6444.5	0.2
$\text{Pr}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_2	6444.7	
$\text{Nd}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_2	6728.5	0.3
$\text{Nd}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_2	6728.8	
$\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	6715.6	7.6
$\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_2	7311.1	7.3
$\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	6723.2	
$\text{Sm}^{\text{II}}[\text{N}(\text{SiMe}_3)_2](\text{THF})_2$	L_3	6715.0	7.8
$\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_3$	L_3	6722.8	
$\text{Sm}^{\text{II}}(\text{C}_5\text{Me}_5)_2(\text{THF})_2$	L_3	6715.2	—
	L_2	7310.7	
$\text{Gd}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	7248.6	0.3
$\text{Gd}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	7248.9	
$\text{Tb}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	7520.3	0.9
	L_2	8258.1	1.0
$\text{Tb}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	7521.2	
	L_2	8259.1	
$\text{Dy}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	7798.1	0.4
$\text{Dy}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	7798.5	
$\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	8075.6	0.5
	L_2	8922.3	0.3
$\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	8076.1	
	L_2	8922.6	
$\text{Er}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	8364.0	0.5
$\text{Er}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	8364.5	
$\text{Tm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	8647.5	7.0
	L_2	9617.1	6.6
$\text{Tm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	8654.5	
	L_2	9623.7	
$\text{Tm}^{\text{II}}\text{I}_2(\text{THF})_3$	L_3	8646.3	7.7
	L_2	9616.0	7.0
$\text{Tm}^{\text{III}}\text{I}_3(\text{THF})_{3.5}$	L_3	8653.8	
	L_2	9623.0	
$\text{Yb}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	8942.7	7.3
$\text{Yb}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	8950.0	
$\text{Lu}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	L_3	9244.4	1.9
$\text{Lu}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	L_3	9246.3	
$\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	K	17 052.6, ^a 17 047.3 ^b	1.0
$\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	K	17 053.6, ^a 17 048.7 ^b	

^a The peak position points were defined as the first point at which the first derivative of the data equaled zero. ^b Because the yttrium measurements were made at the Y K-edge, inflection points for $\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $[\text{K}(2.2.2\text{-cryptand})][\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ are reported.

corresponds to the peak maximum of Sm^{III} . While the magnitude of this feature is invariant with temperature between 8 and 100 K, it shows significant intensity changes during our attempts to reproduce the data, e.g. from sample-to-sample. Hence, we attribute this feature to a small amount of Sm^{III} contamination, which likely arose from unwanted reactions with small amounts of O_2 or H_2O . Despite our best attempts, we were unsuccessful in obtaining completely pure Sm^{II} spectra; (1) analytes were shipped to the synchrotron cold and under vacuum, (2) XANES-samples were prepared at low temperature



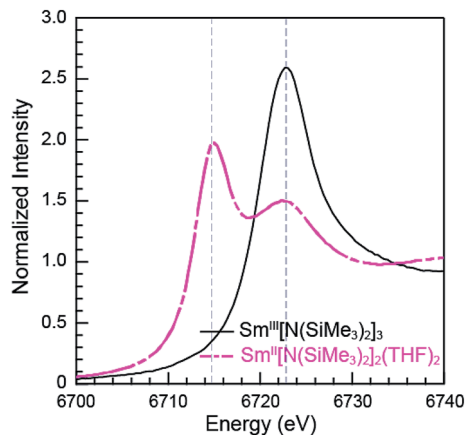


Fig. 2 The background-subtracted and normalized Sm L_3 -edge XANES measurements obtained from the $\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_3$ (black trace) and $\text{Sm}^{\text{III}}[\text{N}(\text{SiMe}_3)_2]_2(\text{THF})_2$ (pink dashed trace).

with rigorous exclusion of air and moisture immediately before the experiment, and (3) measurements were obtained rapidly (low temperature, under vacuum) using an unfocused beam. While it is difficult to identify what caused this contamination, the decomposition rate from X-ray radiolysis under our experimental conditions is slow. For example, when samples are cooled under vacuum (8 to 100 K; 10^{-7} Torr), the Sm^{II} spectra are unchanged after 3 hours of exposure to X-rays using an unfocused beam on SSRL's beam line 11-2. These results suggest that the Sm^{III} species is not being generated during the XANES data acquisition. However, we identified under different experimental conditions – using a focused beam at room temperature under an argon atmosphere on SSRL's beam line 6-2 – that complete conversion of $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ to Sm^{III} occurred in less than 10 seconds.

Tm and Yb $L_{3,2}$ -edge XANES

The background-subtracted and normalized Tm $L_{3,2}$ -edge XANES spectra from +2 and +3 thulium compounds are shown in Fig. 3. As observed for the samarium compounds in Fig. 1 and 2, spectra from the $[\text{K}(2.2.2\text{-cryptand})][\text{Tm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ and $\text{Tm}^{\text{II}}\text{I}_2(\text{THF})_3$ compounds display two main features. There are pronounced peaks near 8645 eV (L_3) and 9615 eV (L_2) and higher energy post-edge shoulders at approximately 8655 eV and 9625 eV. Comparisons with +3 thulium compounds – namely, $\text{Tm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $\text{Tm}^{\text{III}}\text{I}_3(\text{THF})_{3.5}$ – lead us to interpret the Tm^{II} spectra in analogy to the Sm^{II} results described above. For instance, the large edge-features for $\text{Tm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and $\text{Tm}^{\text{II}}\text{I}_2(\text{THF})_3$ are about 7 eV lower in energy than the edge features from $\text{Tm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $\text{Tm}^{\text{III}}\text{I}_3(\text{THF})_{3.5}$, Table 1. The spectral shapes and the trend toward lower energy for the Ln^{II} L_3 -edges from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Tm}, \text{Sm}; x = 0, 1$) are consistent with models of the data generated using quantum chemical *ab initio* FEFF9.6 code based on the multiple scattering theory (see Fig. S1 and S2†).¹⁴

As observed in the Sm^{II} $L_{3,2}$ -edge XANES experiments, the Tm^{II} spectra contain post-edge shoulders associated with small

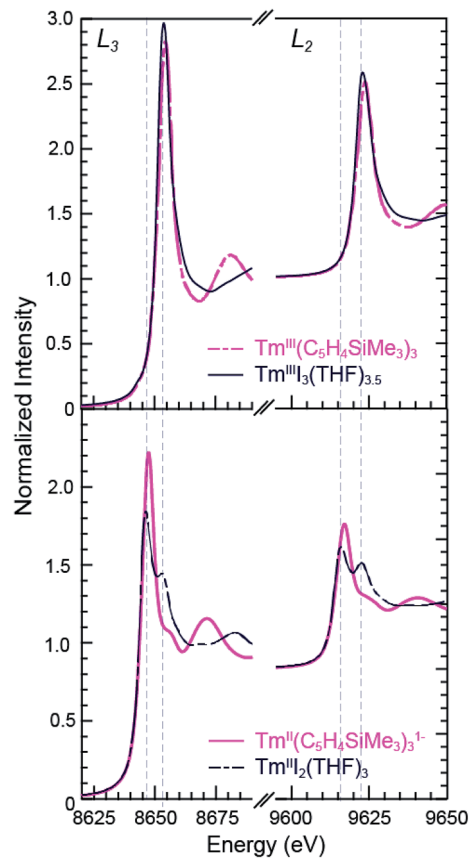


Fig. 3 The background-subtracted and normalized Tm $L_{3,2}$ -edge XANES spectra obtained from $\text{Tm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (top, pink dashed trace), $\text{Tm}^{\text{III}}\text{I}_3(\text{THF})_{3.5}$ (top, black trace), $[\text{K}(2.2.2\text{-cryptand})][\text{Tm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ (bottom, pink trace), and $\text{Tm}^{\text{II}}\text{I}_2(\text{THF})_3$ (bottom, black dashed trace).

amounts of +3 thulium contamination. Variable temperature XANES experiments conducted between 8 and 100 K on these thulium compounds using a small excitation beam (1×1 mm) that was rastered across the sample show small variations in peak intensities. However, because the changes are not reversible and not reproducible, we attribute the slight variances to sample decomposition. Nevertheless, the compounds seem quite stable to X-ray radiation damage on the XANES experimental time scale (10 s to 1.5 h) under our experimental conditions; low temperature (8–100 K), under vacuum (10^{-7} Torr), and in an unfocused beam on SSRL's beam line 11-2.

Despite minor Ln^{III} contamination in the Sm^{II} and Tm^{II} spectra, these results provide confidence and credibility in our abilities to manipulate extremely air and moisture sensitive organometallic complexes at the SSRL synchrotron facility. We remind the reader of the sensitivity of the $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ compounds to hydrolysis, the highly reducing nature of Sm^{II} and Tm^{II} (which have standard reduction potentials of -1.5 and -2.3 V versus SHE),⁴ and of the light sensitivity of $\text{Tm}^{\text{III}}\text{I}_3(\text{THF})_{3.5}$. As noted previously,^{12,15} the consistent 7–8 eV shift between Ln^{II} and Ln^{III} containing compounds highlights the utility of overcoming these sample handling challenges for characterizing $\text{Tm}^{\text{II}} 4f^{13} 5d^0$ versus $\text{Tm}^{\text{III}} 4f^{12} 5d^0$ electronic



configurations using $L_{3,2}$ -edge XANES spectroscopy. Note that while not explicitly described here in detail, Fig. 4 shows that similar results were observed for ytterbium, whose spectrum, also displayed a peak maxima shift of ~ 7 eV upon moving from Yb^{II} ($4f^{14} 5d^0$) to Yb^{III} ($4f^{13} 5d^0$).

$\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ Ln $L_{3,2}$ -edge (Ln = Pr, Nd, Gd, Tb, Dy, Ho, and Er) XANES

The samarium, thulium, and ytterbium $L_{3,2}$ -edge measurements described above provide an experimental basis for using XANES spectroscopy to evaluate the recently discovered $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ (Ln = Pr, Nd, Gd, Tb, Dy, Ho, and Er) compounds.^{1a,1b,8} Fig. 4 compares the background-subtracted

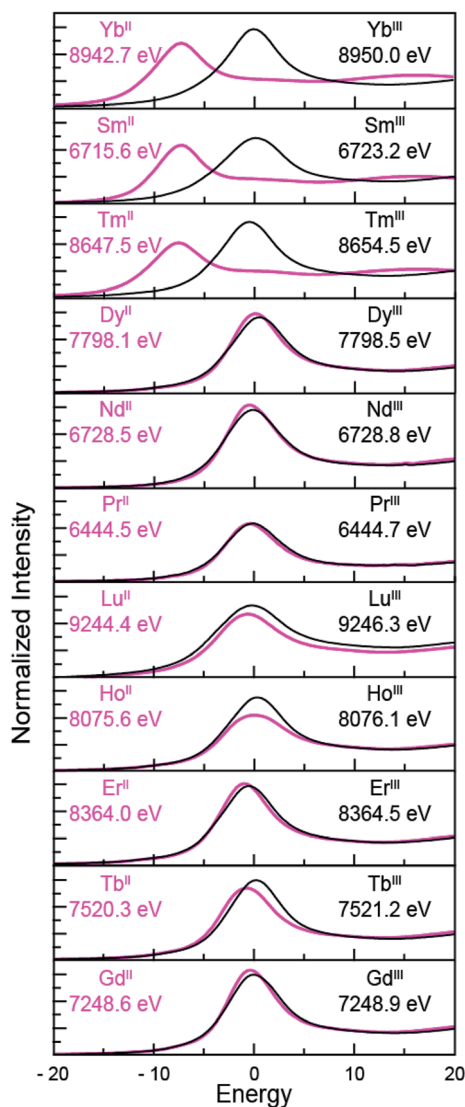


Fig. 4 The background-subtracted and normalized L-edge XANES spectra obtained from $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (black traces) and $[\text{K}(2.2.2\text{-cryptand})][\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ (pink traces) for Ln = Yb, Sm, Tm, Dy, Nd, Pr, Lu, Ho, Er, Tb and Gd. All spectra were collected at the Ln $L_{3,2}$ -edge except Nd and Pr, which were collected at the $L_{2,1}$ -edge. Peak maxima are shown in each pane. The spectra have been ordered from top to bottom based on increasing general reduction potentials.^{4,16}

and normalized Ln $L_{3,2}$ - or $L_{2,1}$ -edge XANES spectra from $[\text{K}(2.2.2\text{-cryptand})][\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ with $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. In this figure, the spectra are ordered from top to bottom as a function of increasing standard reduction potential, as determined by Morss and Mikheev.^{4,16} These data display rising-edge features similar to the samarium and thulium spectra described above. However, in stark contrast to the samarium, thulium, and ytterbium spectra, the L-edge peak maxima from the other $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ anions are quite similar in energy to the neutral $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ compounds. As shown in Fig. 4 and Table 1, small shifts in $L_{3,2}$ -edge inflection points are observed for the other $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$) compounds, ranging from 0.2 to 1.0 eV.

To evaluate the likelihood that the spectra obtained from $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ (Ln = Pr, Nd, Gd, Tb, Dy, Ho, and Er) compounds were indeed correct, a series of control experiments were conducted. Herein we limit the discussion explicitly to the $\text{Ho}^{\text{II}}/\text{Ho}^{\text{III}}$ case. The first control experiment involved analyzing the Ho^{II} and Ho^{III} samples by electronic absorption spectroscopy before and after the Ho $L_{3,2}$ -edge XANES experiment. Because the $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ UV-vis spectrum is distinct from the $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ precursor, electronic absorption spectroscopy provides a robust method for confirming the presence of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ during the XANES experiment. First, an aliquot of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ was characterized by UV-visible spectroscopy (black trace, Fig. 5; pre-XANES). The spectrum showed the characteristic and broad charge transfer band associated with $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$. Moreover, no detectible Ho^{III} was observed. For comparison, the spectrum from $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ is shown as a grey trace. A second aliquot of the $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ was diluted in BN and the Ho $L_{3,2}$ -edge XANES experiment was conducted. Subsequently, the sample – $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and BN – was transferred to a Teflon sealable quartz cuvette and the mixture was again characterized by UV-visible spectroscopy (pink trace, post-XANES).

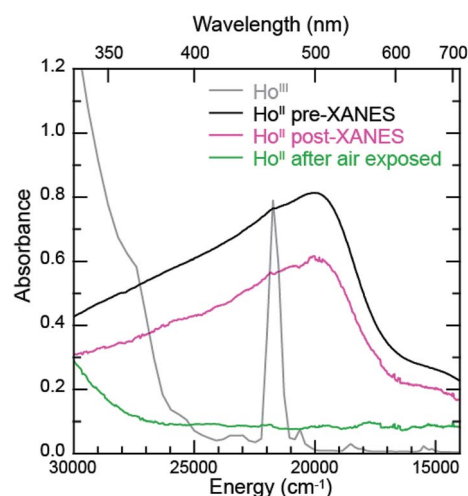


Fig. 5 The background-subtracted UV-vis spectra obtained from $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (grey trace) and $[\text{K}(2.2.2\text{-cryptand})][\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$. Data from $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ were collected (1st) before XANES analysis (black trace), (2nd) after XANES analysis (pink trace), and (3rd) after XANES analysis and exposure to air (green trace).



Unfortunately, because of constraints associated with the XANES holder, this transfer was not quantitative and the overall amount of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ in the cuvette was unknown. A 20% loss during the transfer is possible. Hence, the intensities in the pre-XANES spectrum cannot be directly compared with those from the post-XANES spectrum. Additionally, the BN in the post-XANES spectrum is insoluble and artificially increases the overall UV-visible baseline due to scattering effects. For data comparison, the post-XANES spectrum was background-subtracted to place overall peak heights on the same approximate absorbance scale. Regardless, this experiment unambiguously demonstrates that no detectable amount of $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ was observed before or after the synchrotron experiment. One cannot rule out the possibility of insoluble Ho^{III} contaminants. For example, exposing a Teflon sealable cuvette containing the $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ post-XANES samples to air for 2 s caused an immediate loss of Ho^{II} signal and no ingrowth of $\text{Ho}^{\text{III}} 4f \rightarrow 4f$ transitions. However, when one considers loss of sample during the transfer from the XANES holder to the cuvette, this control experiment suggests that after the $\text{Ho} L_{3,2}$ -edge experiment >80% of the sample was in the form of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$.

Additional support that the $\text{Ho} L_{3,2}$ -edge XANES spectra obtained from $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ was representative of the Ho^{II} organometallic was gleaned from a series of X-ray absorption decomposition experiments. For example, exposing $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, whose absorption peak is at 8075.6 eV, after XANES analysis to air caused the peak position to shift by 0.5 eV to 8076.0 eV, matching the 8076.1 eV peak in $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. Analysis of the second derivative of the Ln^{II} versus Ln^{III} data additionally revealed a spectral diagnostic for the unconventional $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ (Fig. S4†) compounds. For example, all of the +3 $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ precursors contain a minimum in the second derivative approximately 2 eV lower in energy than the corresponding absorption peak. For Sm, Tm, and Yb, this feature is also persists after reduction to the $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ complex. However, reduction to form unconventional divalents, $\text{Ln} = \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{Er}, \text{and Lu}$, caused the pre-edge features to disappear from the $L_{3,2}$ -edges XANES spectra. This observation is documented by the 2nd derivative plots shown in Fig. 6 for $\text{Ho}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 1, 0$) (see ESI† for the other $L_{3,2}$ -edge 2nd derivative spectra). We remind the reader that a minimum in the 2nd derivative indicates the presence of a peak in the XANES data. Fig. 6 shows the pre-edge peak at 8073.0 eV for $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. If the transition corresponds to a $\text{Ln} 2p \rightarrow 5d$ excitation, 5d-orbital population in $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ would shift this feature higher in energy (owing to electron pairing energy) and make it more difficult to resolve. Consistent with this proposition, for Sm, Tm, and Yb analytes – which have $4f^n 5d^0$ (for +3 metals) and $4f^{n+1} 5d^0$ (for +2 metals) electronic configurations with empty 5d orbitals (for both +3 and +2 metals) – pre-edge features were observed in both the +3 and +2 spectra. Regardless of its identity, this pre-edge feature is unexpectedly sensitive to the amount of Ln^{III} present in the Ln^{II} sample, as demonstrated by the $\text{Ho} L_{3,2}$ -edge XANES measurement made on a 1 : 1 mixture of $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, Fig. 6, which showed the pre-edge feature had a lower intensity than the pure Ho^{III} starting material. The absence of the extra

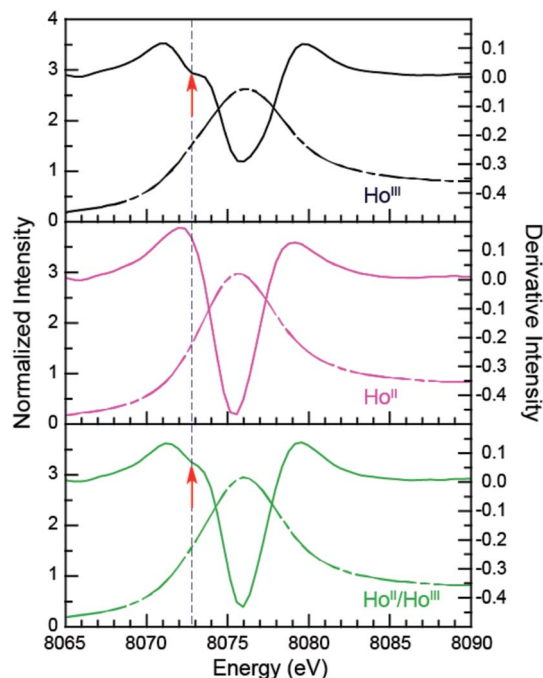


Fig. 6 The background-subtracted and normalized $\text{Ho} L_{3,2}$ -edge XANES measurements obtained from $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (black dashed trace), $[\text{K}(2.2.2\text{-cryptand})][\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ (pink dashed trace) complexes, and a mixture of Ho^{III} and Ho^{II} samples (green dashed trace). Second derivatives of the data are provided as solid traces. Note the pre-edge features (labeled with a red arrow) that are manifested as a minimum in the 2nd derivative.

feature in the $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ $L_{3,2}$ -edge XANES spectra provides a fortuitous alternative fingerprint for the Ln^{II} compounds with $4f^n 5d^1$ electronic configurations. This is especially valuable when one considers that L_3/L_2 absorption peak area comparisons and branching ratio analyses were inconclusive (Table S1†), even for the Sm, Tm, and Yb analytes.

$\text{M}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ K- and $L_{3,2}$ -edge XANES ($\text{M} = \text{Y}, \text{Lu}$)

The subtle rising edge energy shifts between $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ are reminiscent of those accompanying changes in oxidation state for transition metals (K- and L-edges),^{17,18} not lanthanides. For example, changes in d-orbital occupancy only shift the K- and L-edges for transition metals by a few electron volts, which pales in comparison to the 7 eV shifts that accompany oxidation state changes in 4f-element chemistry. For example, the Y K-edge XANES data from $\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and $\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ show a 1.4 eV inflection point shift (Fig. 7, Table 1), which is consistent with the computational results generated using quantum chemical *ab initio* FEFF9.6 code based on the multiple scattering theory (see Fig. S3†).¹⁴ Hence, both experiment and theory indicate that $\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ has a $4d^0$ electronic configuration and $\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ a $4d^1$ configuration. These Y K-edge XANES results agree with the previous analyses of $\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ (UV-vis, EPR, structural metrics)^{1c} and – to the best of our knowledge –



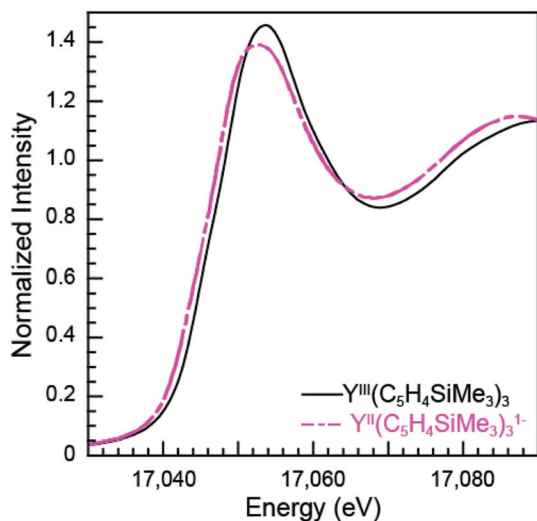


Fig. 7 The background-subtracted and normalized Y K-edge XANES measurements obtained from $\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (black trace) and $[\text{K}(2.2\text{-cryptand})][\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]^{-1}$ (pink dashed trace) complexes.

represent the first Y K-edge XANES spectrum of a molecule containing Y^{II} . Also consider data from the $\text{Lu}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$) pair. Lutetium in the +3 oxidation state has a full 4f-shell. Hence reduction of $\text{Lu}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$, with a $4f^{14} 5d^0$ electron configuration, has to generate a $4f^{14} 5d^1$ configuration in $\text{Lu}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$. Consistent with 5d-orbital occupation, the peak maxima difference between Lu^{III} and Lu^{II} in the Lu $L_{3,2}$ -edge XANES was small, measured at 1.9 eV.

Taken in the context of these $\text{Y}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ and $\text{Lu}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$) XANES measurements – alongside (1st) the experiments we conducted showing our XANES samples contained only marginal quantities of Ln^{III} decomposition products, and (2nd) previously reported UV-vis data, structural metrics, previous computational results – the most plausible interpretations of these Ln $L_{3,2}$ -edge XANES data (Fig. 4) is that reduction of $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ to form an unconventional $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ compound resulted in addition of an electron into a highly shielded 5d-orbital to generate a $4f^n 5d^1$ electronic configuration, not $4f^{n+1} 5d^0$. Although we anticipate that the spectra in Fig. 4 contain some Ln^{III} contamination – in analogy to the Sm^{II} and Tm^{II} spectra in Fig. 1 to 3 – the computational results below provide even more support for the alternative electronic configuration.

Electronic structure calculations

To better understand the origin for the spectroscopic differences between $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ versus $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, electronic structure calculations were conducted on a subset of $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) complexes. This analysis compares $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, which is unambiguously +2, with $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, where the electronic configuration is ambiguous. Calculations for the $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ compounds were restricted to just Sm and Ho, as a follow-on manuscript will compare theoretical results from the other Ln^{II} compounds with other +2 lanthanide and

actinide species. Initially, DFT/PBE calculations were conducted to optimize the geometric structures of $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 1, 0$), see Table 2 for a comparison of experimental and calculated distances and Table S2 (ESI[†]) for the coordinates. The computational results reveal a ground-state $4f^5 5d^0$ electronic configuration (sextet state) for $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and a $4f^6 5d^0$ (septet state) configuration for $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$. Because of the near-degeneracy of 4f-orbitals and the accompanying marginal participation in metal–ligand covalent bonding,^{11c} varying 4f-occupations of the ground-state spin multiplicity has little effects on the geometric structures and spectra. The average 2.513 Å $\text{Sm}^{\text{III}}\text{-C}_{\text{centroid}}$ distance is calculated to be 0.092 Å shorter than the 2.605 Å $\text{Sm}^{\text{II}}\text{-C}_{\text{centroid}}$ distance. This difference is consistent with the differences in Sm^{III} versus Sm^{II} ionic radii¹⁹ and changes in electrostatic interactions between Sm^{III} versus Sm^{II} with $\text{C}_5\text{H}_4\text{SiMe}_3^{1-}$ anions. These calculated distances compare well with experimental results^{8a} and are within the typical error of 2% observed for GGA functionals.

Consistent with previous hybrid DFT calculations that employed no less than 25% Hartree–Fock (HF) exchange,^{1b} our calculations show the ground-state electronic structure of $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ is $4f^{10} 5d^0$ (quintet state), whereas $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ has a $4f^{10} 5d^1$ configuration (sextet state). For example, calculations with the BHandHLYP functional show the $4f^{10} 5d^1$ electronic configuration is 27 kcal mol⁻¹ more stable than the alternative $4f^{11} 5d^0$ configuration (quartet state). In contrast, calculations with functionals that included less HF exchange (PBE, BLYP, and B3LYP) incorrectly predict the alternative $\text{Ho}^{\text{II}} 4f^{11} 5d^0$ configuration as the ground-state (see details in Tables S2 and S3 of the ESI[†]).^{1b,1c,8a,8d} That is to say, GGA and hybrid functionals with lower HF exchange

Table 2 The DFT/PBE calculated and experimental Ln– $\text{C}_{\text{centroid}}$ (Cnt) distances (Å) from $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($\text{Ln} = \text{Sm}, \text{Ho}$). Structural metrics from $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ with $4f^{10} 5d^1$ versus $4f^{11} 5d^0$ electronic configurations were also compared

$\text{Sm}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$)	$\text{Sm}^{\text{III}}, 4f^5 5d^0$		$\text{Sm}^{\text{II}}, 4f^6 5d^0$		$\Delta(\text{Sm}^{\text{II}}\text{-Sm}^{\text{III}})$		
	PBE	Exp ^{8a}	PBE	Exp ^{8a}	PBE	Exp ^{8a}	
	Sm–Cnt1	2.508	2.459	2.610	2.603	0.102	0.144
Sm–Cnt2	2.512	2.459	2.595	2.607	0.083	0.148	
Sm–Cnt3	2.519	2.464	2.609	2.615	0.090	0.151	
Avg(Sm–Cnt)	2.513	2.461	2.605	2.608	0.092	0.147	
$\text{Ho}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$)	$\text{Ho}^{\text{III}}, 4f^{10} 5d^0$		$\text{Ho}^{\text{II}}, 4f^{10} 5d^1$		$\text{Ho}^{\text{II}}, 4f^{11} 5d^0$		$\Delta[\text{Ho}^{\text{II}}(4f^{10} 5d^1)\text{-Ho}^{\text{III}}]$
	PBE	Exp ^{1b}	PBE	Exp ^{1b}	PBE		
	Ho–Cnt1	2.438	2.391	2.477	2.417	2.536	
Ho–Cnt2	2.441	2.393	2.461	2.420	2.509	0.020	0.027
Ho–Cnt3	2.448	2.398	2.481	2.432	2.517	0.033	0.034
Avg(Ho–Cnt)	2.442	2.394	2.473	2.423	2.521	0.031	0.029



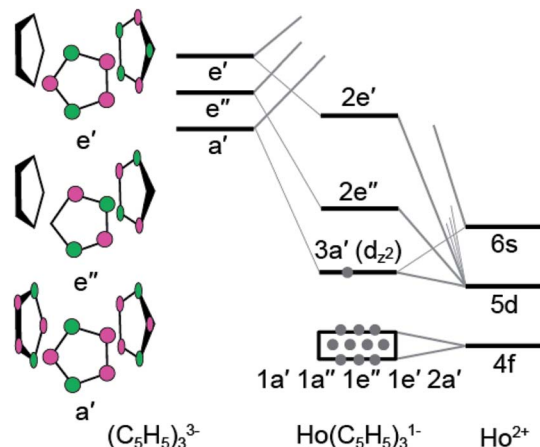


Fig. 8 A qualitative molecular orbital diagram showing molecular orbital interactions in C_{3h} -symmetry for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$.

Table 3 The spin multiplicity ($2S + 1$), total S^2 , Mulliken net spin density for lanthanide atomic electron valence orbitals (s, d, f) calculated for $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) using DFT/BHandHLYP

Compound		$2S + 1$	S^2	Spin	s	d	f
$\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	$4f^5 5d^0$	6	8.77	5.14	0.01	0.09	5.03
$\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	$4f^6 5d^0$	7	12.01	6.04	0.01	0.06	5.96
$\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$	$4f^{10} 5d^0$	5	6.00	4.04	0.00	0.04	3.97
$\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	$4f^{10} 5d^1$	6	8.76	4.86	0.22	0.62	3.98
$\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$	$4f^{11} 5d^0$	4	3.76	3.02	0.00	0.01	3.01

percentages fail to give the correct $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ spin state, which is likely attributable to the delocalization error.^{20,21} Many reports have described how increasing HF exchange

improves the calculated energetics by DFT-based methods such as excitation energy,²² thermochemical kinetics,²³ reaction barriers,²⁴ and electron detachment energy.²⁵ Consistently, our DFT/PBE calculated $\text{Ho}^{\text{III}}(4f^{10} 5d^0)\text{-C}_{\text{centroid}}$ and $\text{Ho}^{\text{II}}(4f^{10} 5d^1)\text{-C}_{\text{centroid}}$ distances are in excellent agreement with experimental values (Table 2), while the $\text{Ho}^{\text{II}}(4f^{11} 5d^0)\text{-C}_{\text{centroid}}$ distances are longer than the experimental results by $\sim 0.1 \text{ \AA}$.^{1b,1c} These results provide confidence in assigning Ho^{II} as having a $4f^{10} 5d^1$ electronic configuration. We refer the interested reader to the experimental section for details of the electronic structure calculation.

To better understand the unusual electronic configuration of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, we found it instructive to interpret the DFT calculations using traditional molecular orbital descriptions derived from group theory considerations of $\text{M}(\text{C}_5\text{H}_5)_3$ in C_{3h} -symmetry. Hence, a qualitative MO level diagram for the C_{3h} - $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$ anion is provided in Fig. 8. As the molecular orbital interactions associated with $\text{Ln}^{\text{III}}(\text{C}_5\text{R}_5)_3$ ($\text{R} = \text{H}$ or alkyl) have been the subject of numerous theoretical and spectroscopic studies,²⁶ this discussion is confined to those orbitals most relevant to the Sm and Ho $L_{3,2}$ -edge XANES measurements. In contrast to previous theoretical results for $\text{M}^{\text{III}}(\text{C}_5\text{H}_5)_3$ in D_{3h} - or C_{3v} -symmetry,^{26b,c,d,g,h,i,j} we find it more appropriate to describe the MO-interaction using C_{3h} -symmetry, as this designation more closely mimics data from the crystal structure of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$.

In the C_{3h} -point group, symmetry allowed mixing between the metal 5d- and cyclopentadienyl π -orbitals – perpendicular to the ring planes – generates bonding interactions of a' , e' , and e'' symmetries, which were σ - π - and δ -bonding with respect to the metal-cyclopentadienyl centroid axes, Fig. 8. Superimposed on this molecular orbital picture, and at lower energy, are $\text{Ln}(\text{C}_5\text{H}_5)$ σ -, π - and δ -bonding orbitals of a' , a'' , e' , and e'' symmetries that originate from mixing between the 4f-orbitals

Table 4 Ground-states configurations from $\text{Ln}(\text{C}_5\text{H}_5)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) complexes from CASPT2/CASSCF calculations.^a Geometries relied on the DFT/PBE optimized geometries of $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$. However, for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$ two geometries were investigated that were derived from the calculated $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_3\text{SiMe}_3)_3^{1-}$ structures with either $4f^{10} 5d^1$ or $4f^{11} 5d^0$ electronic configurations

Ground-state	Configurations
$\text{Sm}^{\text{III}}(\text{C}_5\text{H}_5)_3^{1-}$ X^7A	100%($1a^1 2a^1 3a^1 4a^1 5a^1 6a^1 7a^0$)
$\text{Sm}^{\text{III}}(\text{C}_5\text{H}_5)_3$ X^6A	58%($1a^1 2a^1 3a^1 4a^1 5a^1 6a^0 7a^0$) + 41%($1a^1 2a^1 3a^1 4a^0 5a^0 6a^1 7a^1$)
$\text{Ho}(\text{C}_5\text{H}_5)_3^{1-}$; geometry from $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($4f^{10} 5d^1$) ^b X^6A	71%($1a^2 2a^2 3a^1 4a^2 5a^1 6a^1 7a^1 8a^1$) + 21%($1a^2 2a^1 3a^2 4a^1 5a^2 6a^1 7a^1 8a^1$) + 7%($1a^1 2a^2 3a^2 4a^1 5a^1 6a^2 7a^1 8a^1$)
$\text{Ho}(\text{C}_5\text{H}_5)_3^{1-}$; geometry from $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($4f^{11} 5d^0$) ^b X^6A	70%($1a^2 2a^2 3a^1 4a^2 5a^1 6a^1 7a^1 8a^1$) + 21%($1a^2 2a^1 3a^2 4a^1 5a^2 6a^1 7a^1 8a^1$) + 7%($1a^1 2a^2 3a^2 4a^1 5a^1 6a^2 7a^1 8a^1$)
$\text{Ho}(\text{C}_5\text{H}_5)_3$ X^3A	65%($1a^2 2a^1 3a^2 4a^1 5a^2 6a^1 7a^1 8a^0$) + 20%($1a^2 2a^2 3a^1 4a^2 5a^1 6a^1 7a^1 8a^0$) + 5%($1a^1 2a^2 3a^2 4a^1 5a^1 6a^1 7a^2 8a^0$) + 2%($1a^2 2a^1 3a^1 4a^1 5a^2 6a^2 7a^1 8a^0$) + 1%($1a^2 2a^1 3a^1 4a^2 5a^2 6a^1 7a^1 8a^0$) + 1%($1a^2 2a^1 3a^2 4a^1 5a^1 6a^1 7a^2 8a^0$) + 1%($1a^2 2a^2 3a^2 4a^1 5a^1 6a^1 7a^1 8a^0$)

^a 1a-7a are 4f orbitals, and 8a is 5d orbital. ^b Refer to the DFT/PBE calculated ground-state geometries for $\text{Ho}^{\text{II}}(4f^{10} 5d^1)$ and $\text{Ho}^{\text{II}}(4f^{11} 5d^0)$, respectively, shown in Table 2.



and cyclopentadienyl π -orbitals. In general, the $\text{Ln}(\text{C}_5\text{H}_4\text{-SiMe}_3)_3^{x-}$ ($x = 0, 1$) compounds exhibit little 4f- and cyclopentadienyl orbital mixing, such that the seven primarily 4f-orbitals span a narrow energy range. In contrast, substantial mixing occurs between the Ln 5d- and cyclopentadienyl π -orbitals, with the exception of the 5d-orbital of $3a'$ symmetry (d_{z^2}). Consistent with previous reports,^{1b} metal-cyclopentadienyl mixing is inhibited in this $3a'$ orbital by poor spatial overlap. Hence, the $3a'$ orbital is primarily composed of 5d- and 6s-character and best described as a non-bonding 5d-orbital. For $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ ($4f^5 5d^0$), $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ ($4f^6 5d^0$), and $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ ($4f^{10} 5d^0$), the $3a'$ orbital is empty. As testament, the BHandHLYP calculations at PBE optimized ground-state geometries show the Mulliken net spin densities²⁷ to be almost exclusively distributed on 4f-orbitals (Table 3). Meanwhile, for the $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ anion ($4f^{10} 5d^1$), significant 6s- and 5d-spin density distribution associated with the highest alpha spin occupied orbital indicates that the $3a'$ orbital is singly occupied. A summary of the spin multiplicity results is provided in Table 3. The differences between the $4f^{n+1} 5d^0$ versus $4f^n 5d^1$ electronic configurations of the Ln^{II} ions is observed to influence the metal-cyclopentadienyl bond distances and, as discussed below, is found to significantly impact the Ln L_3 -edge XANES spectrum.

To support the results from the ground-state DFT calculations, CASPT2/CASSCF calculations were performed on the ground-states and low excited-states of simplified $\text{Ln}(\text{C}_5\text{H}_5)_3^{x-}$ ($\text{Ln} = \text{Sm, Ho}; x = 0, 1$) complexes. The DFT/PBE optimized geometries of $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ were used; however, to reduce the computational cost SiMe_3 substituents were replaced with protons having C-H bond lengths of 1.088 Å. Two possibilities were investigated for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$. The first was associated with the calculated structure of $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ with a $4f^{10} 5d^1$ ground-state electronic configuration. The second investigated $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$ geometry was based on the calculated $4f^{11} 5d^0$ $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ structure. Although efforts were made to include all the seven 4f and five 5d orbitals into the active space, the converged CASSCF results for $\text{Sm}(\text{C}_5\text{H}_5)_3^{x-}$ ($x = 0, 1$) showed that the five 5d-orbitals were not correlated and removed from the active space. Meanwhile for $\text{Ho}(\text{C}_5\text{H}_5)_3^{x-}$ ($x = 0, 1$), only the $5d_{z^2}$ -orbital remained in the active space. Hence, the active space calculations were adjusted to include all seven 4f-orbitals for $\text{Sm}(\text{C}_5\text{H}_5)_3^{x-}$ ($x = 0, 1$) and an additionally $5d_{z^2}$ -orbital for $\text{Ho}(\text{C}_5\text{H}_5)_3^{x-}$ ($x = 0, 1$). The results generated a complete active space of 6-electrons with 7-orbitals for $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$, 5-electrons and 7-orbitals for $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_5)_3$, 11-electrons and 8-orbitals for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$, and 10-electrons with 8-orbitals for $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_5)_3$.

Although subtle differences were observed, the ground-state electronic structure results from the CASPT2/CASSCF calculations are similar to those obtained by DFT (Table 4). The “core-like” and nearly degenerated 4f-orbitals resulted in different 4f-occupations with nearly the same energies. The CASPT2/CASSCF results show that $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_5)_3$ has ground sextet state of $4f^5$ configurations and that $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_5)_3^{1-}$ has ground septet state of $4f^6$ configuration, which are the same as DFT results. In the holmium case, $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_5)_3$ has ground quintet state of $4f^{10}$

$5d^0$. For Ho^{II} , both geometries showed a sextet with $4f^{10} 5d^1$ configurations. These Ho^{II} and Ho^{III} results were identical to the DFT calculations. Hence, in terms of evaluating ground-state electronic structures for the $\text{Ln}(\text{C}_5\text{H}_5)_3^{x-}$ ($x = 0, 1$), the CASPT2/CASSCF results are in excellent agreement with the reported DFT results from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$).

Spectral simulations

The open-shell Sm and Ho L_3 -edge XANES spectra from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm, Ho}; x = 0, 1$), were calculated using the transition dipole moment approach based on the Kohn-Sham ground-state molecular orbitals. Using this method the core excitation energies were calculated as the energy differences between occupied and virtual orbitals. Previous studies have demonstrated that this approach provides a sound basis for interpreting the experimental XANES spectra.²⁸ BHandHLYP simulated Ln L_3 -edge XANES spectra from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ are compared with experimental results in Fig. 9 and 10. In

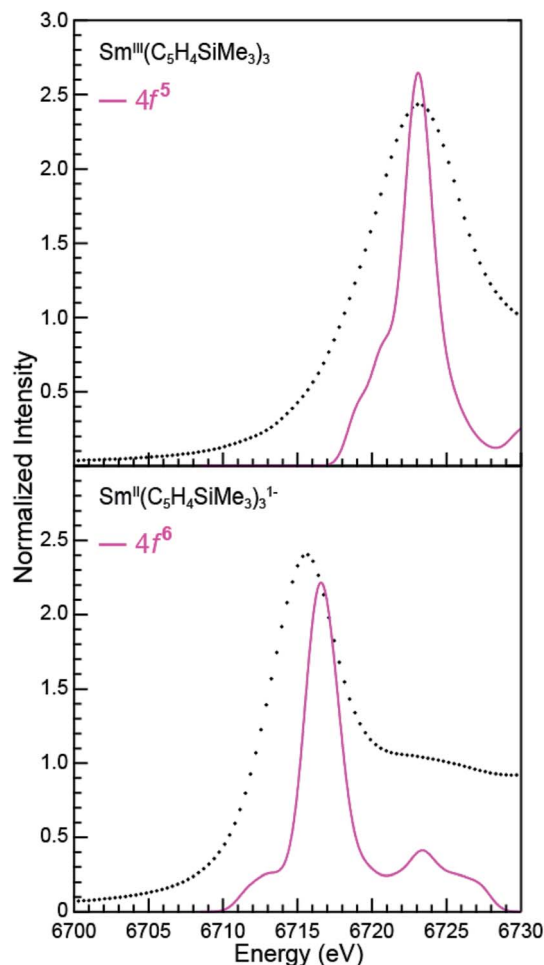


Fig. 9 A comparison of the experimental (●) and transition dipole moment calculations (pink traces) for the Sm L_3 -edge XANES measurements obtained from $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ (top) and $[\text{K}(2.2.2\text{-cryptand})][\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3]$ (bottom). The calculated spectra were shifted by a constant 241.49 eV, which aligned the $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ experimental and calculated edge peak.



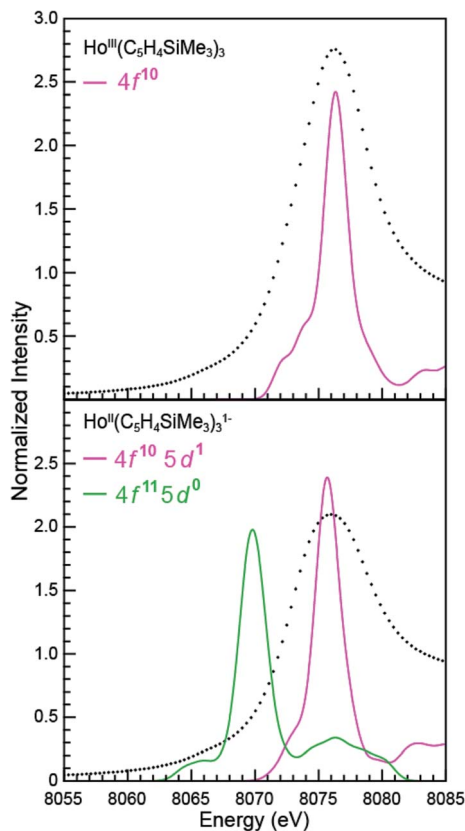


Fig. 10 A comparison of the experimental (●) and transition dipole moment calculations (pink and green traces) for the Ho L₃-edge XANES obtained from Ho^{III}(C₅H₄SiMe₃)₃ (top) and [K(2.2.2-cryptand)] [Ho^{II}(C₅H₄SiMe₃)₃]¹⁻ (bottom). The calculated spectra were shifted by a constant 348.17 eV, which aligned the Ho^{III}(C₅H₄SiMe₃)₃ experimental and calculated edge peak.

these figures, the calculated spectra were shifted by a constant 241.49 eV (Sm) and 348.17 eV (Ho) to line up the Ln^{III}(C₅H₄SiMe₃)₃ L₃-edge peaks, which in turn accounts for omission of the atomic and extra-atomic relaxation associated with the core excitation, relativistic stabilization, and errors associated with the functionals.^{29,30} In the Ln^{II} cases, two options were explored, transitions that involved conventional electronic configurations, Ln^{II} 2p⁶...4fⁿ⁺¹ 5d⁰ → Ln^{II} 2p⁵...4fⁿ⁺¹ 5d¹, and alternatives that involved 5d-orbital occupations, Ln^{II} 2p⁶...4fⁿ 5d¹ → Ln^{II} 2p⁵...4fⁿ 5d². The resulting near edge energies are

summarized in Table 5 alongside analogous values acquired using PBE, BLYP, and B3LYP functionals.

The theoretical analyses reveal the primary contributions to the Ln L₃-edge XANES spectra are electric dipole allowed excitations from Ln 2p-orbitals to unoccupied states that contain metal d-character. Of the functionals explored, the L₃-edge energy differences calculated using BHandHLYP were in best agreement with the experiment. For example, in the Sm(C₅H₄SiMe₃)₃^{x-} case, where the 4f- and 5d-orbital occupancies are well established, energy differences between the Sm^{III} (4f⁵ 5d⁰) and Sm^{II} (4f⁶ 5d⁰) L₃-edge positions are calculated to be 6.5 eV, which is in good agreement with the measured value of 7.6 eV. Results from the B3LYP calculations modestly agree with the experimental data, while larger deviations are observed using BLYP and PBE. The two GGA functionals, BLYP and PBE, without any HF exchange give the same L₃-edge energy difference. This comparison (BHandHLYP, B3LYP, BLYP, and PBE) unambiguously shows the importance of Hartree-Fock (HF) exchange in computationally evaluating L₃-edge XANES spectra. This result highlights the importance of high HF exchange in correctly capturing electron transition energies and is consistent with conclusions from previous theoretical studies.²²

Calculations on Ho(C₅H₄SiMe₃)₃^{x-} are similar to those from Sm(C₅H₄SiMe₃)₃^{x-} in that the BHandHLYP provides the best agreement with the experimental data (Table 5), e.g. energy differences between the Ho^{III} (4f¹⁰ 5d⁰) and Ho^{II} (4f¹⁰ 5d¹) L₃-edge peak maxima are calculated to be 0.7 eV and measured to be 0.5 eV. The Ho(C₅H₄SiMe₃)₃^{x-} calculations differ in that they invoke the Ho^{II} low energy 4f¹⁰ 5d¹ ground-state electronic configuration. We note that calculations involving the higher energy 4f¹¹ 5d⁰ Ho^{II} electronic configuration grossly overestimate the Ho^{III}/Ho^{II} L₃-edge energy by 6.5 eV.

To better understand the how 4f- versus 5d-orbital occupancy influence Ln L₃-XANES spectra, the ground-state 2p-orbital energies are plotted alongside the average 5d- and 6d-orbital energies in Fig. 11 for Ln(C₅H₄SiMe₃)₃^{x-} (Ln = Sm, Ho; x = 0, 1). We remind the reader that the major contributors to the Ln(C₅H₄SiMe₃)₃^{x-} L₃-edge XANES spectra result from dipole allowed transitions between core 2p- and unoccupied d-orbitals. Upon reduction of Ln^{III} to Ln^{II}, the 2p-, 5d-, and 6d-orbital energies increase. For both Sm and Ho, adding the electron into the 4f-shell, Ln^{III} (4fⁿ 5d⁰) + 1e¹⁻ → Ln^{II} (4fⁿ⁺¹ 5d⁰), raises the Ln 2p- and 5d-/6d-orbital energies by 11.5–12.0 eV and 5.0–5.5 eV, respectively. These changes in orbital energies account

Table 5 DFT calculated and experimental peak maximum for the Ln(C₅H₄SiMe₃)₃^{x-} (Ln = Sm, Ho; x = 0, 1) XANES spectra

		PBE	BLYP	B3LYP	BHandHLYP	Exp.
Sm ^{III} (C ₅ H ₄ SiMe ₃) ₃	4f ⁵ 5d ⁰	6873.4	6874.8	6910.7	6964.6	6723.2
Sm ^{II} (C ₅ H ₄ SiMe ₃) ₃ ¹⁻	4f ⁶ 5d ⁰	6870.8	6872.2	6906.1	6958.1	6715.6
Δ[Sm ^{III} -Sm ^{II}]		2.6	2.6	4.6	6.5	7.6
Ho ^{III} (C ₅ H ₄ SiMe ₃) ₃	4f ¹⁰ 5d ⁰	8325.6	8327.1	8366.6	8424.5	8076.1
Ho ^{II} (C ₅ H ₄ SiMe ₃) ₃ ¹⁻	4f ¹⁰ 5d ¹	8325.8	8327.3	8366.2	8423.8	8075.6
Ho ^{II} (C ₅ H ₄ SiMe ₃) ₃ ¹⁻	4f ¹¹ 5d ⁰	8322.6	8324.1	8361.1	8418.0	—
Δ[Ho ^{III} -Ho ^{II} (4f ¹⁰ 5d ¹)]		-0.2	-0.2	0.4	0.7	0.5
Δ[Ho ^{III} -Ho ^{II} (4f ¹¹ 5d ⁰)]		3.0	3.0	5.5	6.5	—



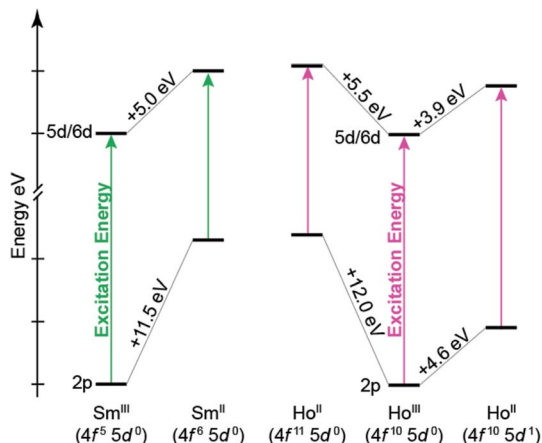


Fig. 11 Quantitative comparison of ground-state 2p- and average 5d/6d-orbital energies from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) for a variety of electronic configurations. The solid arrow represents the excitation energy associated with the Ln L_{3-} edge excitation. To plot both Sm and Ho on the energy scale, the energies associated with the Ln^{III} 2p-orbitals were set to zero.

for $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ L_{3-} edge excitation energy being ~ 7 eV less than that of $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. Adding the electron into 5d-shell, $\text{Ln}^{\text{III}}(4f^n 5d^0) + 1e^{-1} \rightarrow \text{Ln}^{\text{II}}(4f^{n+1} 5d^1)$, also increases the Ln 2p- and 5d-/6d-orbital energies; however, to a lesser extent. Most notably for the 2p-orbitals. For example, the Ho 2p- and 5d/6d-average orbital energies increase by 4.6 eV and 3.9 eV, respectively. This modest energy shift decreases the L_{3-} edge excitation energy for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ by < 1 eV in comparison to $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$. Overall, these results demonstrate that Ln 2p-electrons experienced stronger Coulomb repulsion from Ln 4f-electrons than higher lying 5d-electrons. We additionally correlate the magnitude of this repulsion with the radial distribution of the 4f- versus 5d-orbitals. Because the 4f-orbitals are closer to the nucleus,³¹ increased 4f-orbital occupancy destabilizes the core 2p-orbital energies to a large extent. Meanwhile, occupancy of the more diffuse 5d-orbitals has less impact on the 2p-orbital energies.

Discussion

Herein we describe the use of XANES spectroscopy to characterize the electronic configurations of formally +2 lanthanide compounds of the general formula $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$. Through comparisons with a carefully selected series of standards, including $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$, our XANES results from $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, $\text{Tm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and $\text{Yb}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ demonstrate that these compounds contained Ln^{II} ions with $4f^6 5d^0$ (Sm^{II}), $4f^{13} 5d^0$ (Tm^{II}), and $4f^{14} 5d^0$ (Yb^{II}) electronic configurations. These results are in agreement with previously acquired spectroscopic data, *i.e.* UV-vis, magnetic susceptibility, and the Ln- $\text{C}_{\text{centroid}}$ distances (which were ~ 0.1 Å longer than the Ln^{III} analogue). Consistent with previous studies,^{8a,8d} the measurements highlight the utility of Ln $L_{3,2}$ -edge XANES spectroscopy in characterizing f-orbital occupancies from Ln^{III} ($4f^n 5d^0$) and Ln^{II} ($4f^{n+1} 5d^0$) ions. For example,

changes in 4f-electron occupancy shift the Ln peak maxima in the $L_{3,2}$ -edges by approximately 7 eV. The magnitude of these shifts is impressive in comparison to transition metal K- and L-edge XANES experiments,^{17,18} where changes in d-orbital occupancies are known to shift absorption edges by only a few eV.

The Ln L-edge XANES studies from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Pr}, \text{Nd}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{and Er}; x = 0, 1$) show much smaller shifts in rising-edge energies than the samarium, thulium, and ytterbium analogues. For example, the peak maxima differences between $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ and $\text{Ln}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ range from only 0.2 to 1.0 eV (Table 1). These values are substantially less than the 7–8 eV change expected for an increase in 4f-orbital occupancy, *i.e.* $\text{Ln}^{\text{III}}(4f^n 5d^0) + e^{-1} \rightarrow \text{Ln}^{\text{II}}(4f^{n+1} 5d^0)$. Instead, the 0.2 to 1.0 eV shifts are reminiscent of the those accompanying the reduction of $\text{Y}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3(4d^0)$ to $\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}(4d^1)$ and $\text{Lu}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3(4f^{14} 5d^0)$ to $\text{Lu}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}(4f^{14} 5d^1)$. In these yttrium and lutetium scenarios, the increase in d-orbital occupancy shifts the peak maximum by only ~ 1 eV (inflection point change of 1.4 eV) and 1.9 eV, respectively. These shifts provide strong evidence that the yttrium ion in $\text{Y}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ is best described as +2 with a $4d^1$ electronic configuration and that the lutetium ion in $\text{Lu}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ is +2 with a $4f^{14} 5d^1$. Given that shifts from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Pr}, \text{Nd}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{and Er}; x = 0, 1$) were also small, we initially questioned the possibility that reduction of $\text{Ln}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3(4f^n 5d^0)$ generated a lanthanide ion with a $4f^n 5d^1$ electronic configuration, instead of the more typical $4f^{n+1} 5d^0$ configuration.

To better understand the Ln L_{3-} edge XANES spectra from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$), DFT calculations were conducted on the $\text{Sm}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ and $\text{Ho}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ analytes. Consistent with previous reports, the ground-state DFT calculations show the electronic configurations for $\text{Sm}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$, $\text{Sm}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, and $\text{Ho}^{\text{III}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3$ are $\text{Sm}^{\text{III}} 4f^5 5d^0$, $\text{Sm}^{\text{II}} 4f^6 5d^0$, and $\text{Ho}^{\text{III}} 4f^{10} 5d^0$, respectively. In contrast for $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$, the calculations indicate that the ground-state electronic configuration is $4f^{10} 5d^1$, with the non-bonding $5d_{z^2}$ -orbital of a' -symmetry being singly occupied. CASPT2/CASSCF calculations on the simplified models, $\text{Ln}(\text{C}_5\text{H}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$), were completely consistent with the assignments of the DFT calculations. As such the Ln L_{3-} edge XANES spectra were simulated using transition dipole moment calculations for a variety of electronic configurations, spanning $\text{Ln}^{\text{III}} 4f^n 5d^0$, $\text{Ln}^{\text{II}} 4f^{n+1} 5d^0$, and $\text{Ln}^{\text{II}} 4f^n 5d^1$. For both Sm and Ho, the calculations suggest that reducing $\text{Ln}^{\text{III}}(4f^n 5d^0)$ by adding an electron in the 4f-manifold to generate $\text{Ln}^{\text{II}}(4f^{n+1} 5d^0)$ appreciably shifts the Ln L_{3-} edge by approximately 7 eV. In contrast, reducing $\text{Ln}^{\text{III}}(4f^n 5d^0)$ by adding an electron into the 5d-manifold to generate $\text{Ln}^{\text{II}}(4f^n 5d^1)$ slightly shifts the Ln L_{3-} edge to lower energy (on the order of ~ 1 eV).

Concluding remarks

Our results indicate that the differences in $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) excitation energies stem from electron repulsion between 2p- and either 5d- or 4f-electrons (Fig. 11). For example, increases in Ln 4f-orbital occupation



significantly destabilize the core 2p-orbital energy levels, which decrease the Ln L₃-edge excitation energy by ~7–8 eV. In contrast, increased occupancy for the more diffuse 5d-orbitals has marginal impact on core 2p-energy levels and the Ln L₃-edge excitation energy (0.2–1.9 eV). One might describe the 4f¹⁰ 5d¹ electron configuration in Ho^{II}(C₅H₄SiMe₃)₃¹⁻ as mimicking the 4f¹⁰ electronic configuration in Ho^{III}(C₅H₄SiMe₃)₃, with the extra electron ‘hidden’ in a highly shielded 5d-orbital. We anticipate that this interpretation is quite general and will be used to explain the similar Ln^{II}/Ln^{III} peak maxima shifts and Ln^{II}/Ln^{III}-C_{centroid} bond distances in the other Ln(C₅H₄-SiMe₃)₃^{x-} (Ln = Pr, Nd, Gd, Tb, Dy, Er, and Lu; x = 0, 1) compounds. Hence, our current computational and spectroscopic efforts are focused on evaluating recently reported compounds that contain formally lanthanide(II) and actinide(II) ions.

Among the numerous examples where ligand environments with C₃-symmetry have been exploited to advance transition metal and f-element chemistry,³² our results highlight another extraordinary property associated with a C₃-ligand framework. For example, we identified that the tris-cyclopentadienyl coordination environment provides a mechanism for stabilizing Ln^{II} 4fⁿ 5d¹ electronic configurations through the accessibility of a low-lying 5d-orbital of d' symmetry. The results additionally suggest an electronic structure break between Tm^{II}(C₅H₄-SiMe₃)₃¹⁻ and Dy^{II}(C₅H₄SiMe₃)₃¹⁻. It appears that 4fⁿ⁺¹ 5d⁰ electronic configurations are most stable when the reduction potentials for the lanthanide ions in Ln^{II}(C₅H₄SiMe₃)₃¹⁻ are less than or equal to that of Tm^{II}(C₅H₄SiMe₃)₃¹⁻. Meanwhile, those with reduction potentials greater than or equal to Dy^{II}(C₅H₄-SiMe₃)₃¹⁻ are best described as 4fⁿ 5d¹. While the generality of this interpretation has yet to be determined, we anticipate – based on previous studies on LnX₂ (X = halide) – that the electronic structure breaking point is quite dynamic and can shift to higher reduction potentials, *i.e.* those of Dy^{II} and Nd^{II}, depending in the ligand environment. Our current efforts are focused on identifying the implications of these results on lanthanide reactivity.

Experimental

Sample preparation

The analytes were synthesized at the University of California in Irvine CA with rigorous exclusion of air and moisture.^{1c,8a} The Ln^{III}(C₅H₄SiMe₃)₃,³³ Ln^{II}(C₅H₄SiMe₃)₃^{1-,1,8a} Sm^{II}(C₅Me₅)₂(THF)₂,³⁴ Sm^{III}[N(SiMe₃)₂]₃,³⁵ Sm^{II}[N(SiMe₃)₂]₂(THF)₂,³⁶ Tm^{II}I₂(THF)₂,³⁷ and Tm^{III}I₃(THF)_{3.5} (ref. 38) were prepared as previously described. Analytes were sealed in ampoules and transported in a cooler filled with dry ice to the Stanford Synchrotron Radiation Lightsource (SSRL) where they were stored at –80 °C. Three hours prior to analysis by XAFS, the lanthanide samples were transferred into an argon filled glovebox. The samples were kept cold by preparing them on an aluminum block, which had been plumbed to accommodate flowing helium gas cooled from a dry ice/ethanol bath. Note, all equipment (including the holder, spatulas, wrenches, boron nitride, *etc.*) were cooled on the block prior to sample preparation. Samples were diluted

with boron nitride, which had been dried at elevated temperature (200 °C) under vacuum (10⁻³ Torr) for 48 hours. A mixture of the analyte and BN were weighed out, such that the edge jump for the absorbing atom was calculated to be at ~1 absorption length in transmission (between 8 to 30 mg of sample and ~50 mg of BN). Samples were ground using a Wig-L-Bug®, a Teflon bead, and a polycarbonate capsule. The finely ground powders were pressed as a pellet into a slotted aluminum sample holder. These precautions were taken to minimize self-absorption. The holder was equipped with Kapton windows (1 mil), one was fixed with super glue and the other was Kapton tape. For Pr, Nd, Sm, Gd, Tb, Dy, Y, Ho, Er, Tm, Yb and Lu analytes, the holder was brought out of the glovebox, immediately submerged in liquid nitrogen for transportation to the beam line, and loaded into the cryostat. The cryostat was immediately evacuated and attached to the beamline 11-2 XAFS rail and cooled with either liquid nitrogen or liquid helium.

Data acquisition

The cryostat was attached to the beamline 11-2 XAFS rail (SSRL), which was equipped with three ionization chambers through which nitrogen gas was continually flowed. One chamber (10 cm) was positioned before the cryostat to monitor the incident radiation (I₀). The second chamber (30 cm) was positioned after the cryostat so that sample transmission (I₁) could be evaluated against I₀ and so that the absorption coefficient (μ) could be calculated as ln(I₀/I₁). The third chamber (I₂; 30 cm) was positioned downstream from I₁ so that the XANES of a calibration foil could be measured against I₁. A potential of 1600 V were applied in series to the ionization chambers.

Samples were calibrated to the energy of the first inflection point of a calibration foil, whose spectrum was measured *in situ* from the sample using the transmitted portion of the beam. The measurements were calibrated as follows. The Y K-edges were calibrated to the Y K-edge (17 038.4 eV) of an yttrium foil. The Lu L₃-edge to the Cu K-edge of a copper foil at 8979 eV. The Er and Yb L₃-edges to the Ni K-edge of a nickel foil at 8333 eV. The Tm L₃-edges were calibrated to the Ho L₃-edge at 8070.1 eV. The Dy L₃-edge was calibrated to the Dy L₃-edge of a dysprosium foil at 7790.0 eV. The Ho L₃-edges to the Co K-edge of a cobalt foil at 7709 eV. Sm, Gd, and Tb L-edges to the Fe K-edge of an iron foil at 7111 eV. The Pr, and Nd L-edges to the Cr K-edge of a chromium foil at 5989 eV.

The X-ray absorption near edge spectra (XANES) were measured at the SSRL, under dedicated operating conditions (3.0 GeV, 5%, 500 mA using continuous top-off injections) on end station 11-2. This beamline, which was equipped with a 26-pole, 2.0 tesla wiggler, utilized a liquid nitrogen-cooled double-crystal Si[220] monochromator and employed collimating and focusing mirrors. A single energy was selected from the white beam with a liquid-N₂-cooled double-crystal monochromator utilizing Si[220] (φ = 0) crystals. Harmonic rejection was achieved by detuning the second crystal of the monochromator by 50% at ~600 eV above the absorbing edge. The vertical slit sizes were 1 mm and the beam was unfocused.



Data analysis

Data manipulations and analyses were conducted as previously described.³⁹ Energy calibrations were conducted externally using the first inflection point of the rising edge of the calibration spectrum. Data were analyzed by fitting a line to the pre-edge region, which was subsequently subtracted from the experimental data to eliminate the background of the spectrum. The data were normalized by fitting a first-order polynomial to the post-edge region of the spectrum and setting the edge jump at to an intensity of 1.0.

UV-visible spectroscopy

Prior to transporting the $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ samples to the synchrotron, the compound was characterized by UV-vis, as previously reported.⁸ The sample was first prepared for XANES analysis in an argon-filled glovebox by finely grinding $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ (19.4 mg) with cold anhydrous boron nitride, BN (60.6 mg) for 2 min in polystyrene canisters with plexiglass pestles using a Wig-L-Bug® grinder to obtain a homogeneous fine powder. The sample was loaded within a slotted aluminum holder, whose slot dimensions were $5 \times 20 \times 1$ mm. The holder was equipped with Kapton tape windows (1 mL). This holder was nested within an additional holder, also equipped with Kapton windows (1 mL) that were sealed with indium wire gaskets. This holder is well established as providing robust exclusion of air and moisture. The sample holder was placed on the rail at SSRL's beam line 11-2 and the Ho L_3 -edge spectrum obtained in transition mode at room temperature. After data collection the holder was returned to the glovebox and disassembled. The $\text{Ho}^{\text{II}}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{1-}$ and BN mixture was transferred to a Teflon sealable quartz cuvette with THF (dried over Na/K alloy and benzophenone). The sample was again removed from the glovebox and analyzed using a CARY 50 spectrometer. The UV-vis data were background-subtracted. Owing to the suspended BN, a constant 1.15 absorption value was subsequently subtracted to set the background to zero.

Density functional calculations

Ground-state electronic structure calculations were performed on the $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) using the generalized gradient approximation (GGA) with the PBE exchange–correlation functional⁴⁰ as implemented in the Amsterdam Density Functional (ADF 2014.11).^{41–43} For geometry optimization, the Slater basis sets with the quality of triple- ζ plus one polarization functions (TZP)⁴⁴ were used, with the frozen core approximation applied to the inner shells [$1s^2-4d^{10}$] for Sm and Ho, [$1s^2$] for C, [$1s^2-2p^6$] for Si. All electron TZ2P basis sets were used for spectroscopic simulation by employing the PBE,⁴⁰ BLYP,^{45,46} B3LYP,^{45,46} and BHandHLYP^{47,48} functionals. The latter three functionals combine the LYP⁴⁶ GGA for correlation with three different approximations for exchange, *i.e.*, Becke's GGA (B)⁴⁵ for exchange, the Becke's three-parameter (B3)⁴⁷ hybrid functional including 20% HF exact exchange, and the half-and-half hybrid containing 50% HF exact exchange.⁴⁸ The B3LYP and BHandHLYP functionals were chosen because

they give good performance in excitation energy of charge-transfer states and were commonly used.^{22a,49,50} The BLYP was employed together with B3LYP and BHandHLYP to study the impact of the percentage of HF exchange on the excitation energy and spectral shape. The scalar relativistic (SR) effects were taken into account by the zero-order regular approximation (ZORA).⁵¹ Geometries were fully optimized without symmetry at the SR-ZORA level with the gradient convergence of 10^{-5} , and frequency calculations were carried out to verify the local minimum on the potential energy surface. In the ground-state electronic structure calculations for $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$), the high-spin multiplicity was used for each electron configuration. Specifically, Sm^{III} ($4f^5 5d^0$) had a ground sextet state, and Sm^{II} ($4f^6 5d^0$) had a ground septet state; Ho^{III} ($4f^{10} 5d^0$) has ground quintet state, and Ho^{II} ($4f^{10} 5d^1$) had a ground sextet state, and Ho^{II} ($4f^{11} 5d^0$) had ground quartet state (Table 3).

DFT-simulation of Ln L_3 -edge XANES spectra

The L_3 -edge XANES spectra from $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}, \text{Ho}; x = 0, 1$) were simulated as the Kohn–Sham orbital energy differences, *i.e.*, the energy difference between an occupied orbital and a virtual orbital of the ground-state. For a specific core excitation, the oscillator strength was calculated from the transition dipole approximation between this occupied orbital and the virtual orbital. The core electron excitation was calculated originating from Ln 2p dominated MOs to virtual MOs at the DFT/PBE optimized ground-state geometry. All other excitations from orbitals between the Ln 2p and HOMOs were excluded by restricting the energy range of the occupied orbitals involved in the excitations, so that only excitations from Ln 2p core levels to virtual MOs were allowed. The relaxation due to the core hole was assumed constant. All the calculated transition intensities were evenly broadened with a Gaussian function of full-width at half-maximum of 1.7 eV (*i.e.*, peak width) to emulate the experimental spectra.

FEFF spectral simulations

The $\text{Ln}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($\text{Ln} = \text{Sm}$ and $\text{Tm}; x = 0, 1$) Sm and Tm L_3 -edge and $\text{Y}(\text{C}_5\text{H}_4\text{SiMe}_3)_3^{x-}$ ($x = 0, 1$) Y K-edge XANES spectra and the angular momentum projected density of states were calculated with the FEFF9.6 *ab initio* quantum chemical code based on the multiple scattering theory (see ESI†).¹⁴ The potentials of free atoms were calculated with a relativistic Dirac–Fock atom code part of FEFF9.6. The scattering potentials were calculated self-consistently by overlapping the free atomic densities in the muffin tin approximation within a cluster of 334 atoms (SCF card; UNFREEZF card was not included). The energy dependent exchange Hedin–Lundquist potential was used for the fine structure and the atomic background (EXCHANGE card). The full multiple scattering XANES spectra were calculated for an atomic cluster of 334 atoms centered on the absorbing Sm/Tm/Y atom (FMS and XANES cards). Best agreement between calculation and experiment was found by applying “COREHOLE FSR” option to screen the $2p_{3/2}$ (Sm/Tm) or $1s$ (Y) core-holes. The FOLP card (FOLP 1 1.07) was used for



calculating the Sm spectra, as the overlap of the muffin tin radii was reported to be too large by the program. This value was chosen as it was found for the calculations of the Tm and Y spectra. We have obtained comparable results (not shown here) for Tm by including the f valence states in the self-consistent calculations of the scattering potentials (UNFREEZF card).

CASPT2/CASSCF calculations

Using the complete-active-space multi-configuration approach with second-order perturbation theoretical correction (CASPT2)^{52,53} implemented in Molpro 2015.1 program, *ab initio* WFT calculations were performed.^{54,55} To reduce the computational cost, CASPT2/CASSCF calculations were carried out on the ground-states and low excited-states of the simplified Ln(C₅H₅)₃^{x-} (Ln = Sm, Ho; x = 0, 1) complexes. The DFT/PBE optimized geometries of Ln(C₅H₄SiMe₃)₃^{x-} were used in the calculations. Here the original SiMe₃ substituents, ancillary groups, were replaced with protons having C–H bond lengths of 1.088 Å. For Ho(C₅H₅)₃¹⁻, two geometries derived from Ho^{II} (4f¹¹ 5d⁰) and Ho^{III} (4f¹⁰ 5d¹) were used. We applied the cc-pVDZ basis sets for H and C,⁵⁶ Stuttgart energy-consistent relativistic pseudopotentials ECP28MWB,^{57,58} and the corresponding ECP28MWB-SEG basis for Sm and Ho. Although attempts to include all the seven 4f- and five 5d-orbitals into active space were made, the converged CASSCF results showed that for Sm(C₅H₅)₃^{x-} (x = 0, 1) the five 5d-orbitals are not correlated and were removed out of active space. In contrast for Ho(C₅H₅)₃^{x-} (x = 0, 1), only the 5d_{z²}-orbital remained in the active space. Therefore, the active space was adjusted to include all the 4f-orbitals for Sm(C₅H₅)₃^{x-} (x = 0, 1) and additionally the 5d_{z²}-character orbital for Ho(C₅H₅)₃^{x-} (x = 0, 1). In the CASPT2 calculations, the ionization-potential/electron-affinity corrected zeroth-order Hamiltonian was used with an IPEA shift of 0.25 a.u.⁵⁹ The 1s-core orbitals of the C atoms, and 4s-, 4p-, 4d-orbitals of the Sm and Ho atoms were kept frozen in the CASPT2 calculations.

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References

- (a) M. R. MacDonald, J. W. Ziller and W. J. Evans, *J. Am. Chem. Soc.*, 2011, **133**, 15914; (b) M. R. MacDonald, J. E. Bates, M. E. Fieser, J. W. Ziller, F. Furche and W. J. Evans, *J. Am. Chem. Soc.*, 2012, **134**, 8420; (c) M. R. MacDonald, J. E. Bates, J. W. Ziller, F. Furche and W. J. Evans, *J. Am. Chem. Soc.*, 2013, **135**, 9857.
- G. Parkin, *J. Chem. Educ.*, 2006, **83**, 791.
- P. Karen, *Angew. Chem., Int. Ed.*, 2015, **54**, 4716.
- L. R. Morss, *Chem. Rev.*, 1976, **76**, 827.
- (a) G. Meyer and M. S. Wickleder, *Handbook on the Physics and Chemistry of Rare Earths*, Elsevier Science B. V., Amsterdam, 2000, vol. 28; (b) J. D. Corbett, *Rev. Chim. Miner.*, 1973, **10**, 239.
- P. B. Hitchcock, M. F. Lappert, L. Maron and A. V. Protchenko, *Angew. Chem., Int. Ed.*, 2008, **47**, 1488.
- (a) W. J. Evans, N. T. Allen and J. W. Ziller, *J. Am. Chem. Soc.*, 2000, **122**, 11749; (b) M. N. Bochkarev, I. L. Fedushkin, S. Dechert, A. A. Fagin and H. Schumann, *Angew. Chem., Int. Ed.*, 2001, **40**, 3176; (c) M. N. Bochkarev, *Coord. Chem. Rev.*, 2004, **248**, 835; (d) W. J. Evans and S. E. Foster, *J. Organomet. Chem.*, 1992, **433**, 79; (e) Y. K. Gun'ko, P. B. Hitchcock and M. F. Lappert, *Chem. Commun.*, 1998, 1843; (f) L. Huebner, A. Kornienko, T. J. Emge and J. G. Brennan, *Inorg. Chem.*, 2004, **43**, 5659.
- (a) M. E. Fieser, M. R. MacDonald, B. T. Krull, J. E. Bates, J. W. Ziller, F. Furche and W. J. Evans, *J. Am. Chem. Soc.*, 2015, **137**, 369; (b) M. R. MacDonald, M. E. Fieser, J. E. Bates, J. W. Ziller, F. Furche and W. J. Evans, *J. Am. Chem. Soc.*, 2013, **135**, 13310; (c) R. R. Langeslay, M. E. Fieser, J. W. Ziller, F. Furche and W. J. Evans, *Chem. Sci.*, 2015, **6**, 517; (d) K. R. Meihaus, M. E. Fieser, J. F. Corbey, W. J. Evans and J. R. Long, *J. Am. Chem. Soc.*, 2015, **137**, 9855.
- K. Binnemans and C. Görller-Walrand, *Chem. Phys. Lett.*, 1995, **235**, 163.
- (a) A. V. Soldatov, Y. V. Sukhetsky and A. Bianconi, *Nucl. Instrum. Methods Phys. Res., Sect. B*, 1991, **308**, 246; (b) R. K. Singhal and K. B. Garg, *J. Magn. Magn. Mater.*, 1992, **116**, 238; (c) F. Lissner, K. Kraemer, T. Schleid, G. Meyer, Z. Hu and G. Kaindl, *Z. Anorg. Allg. Chem.*, 1994, **620**, 444; (d) M. R. Antonio, J. S. Xue and L. Soderholm, *J. Alloys Compd.*, 1994, **207/208**, 444; (e) Z. Hu, G. Kaindl and G. Meyer, *J. Alloys Compd.*, 1997, **246**, 186; (f) Z. Hu, G. Kaindl and B. G. Muller, *J. Alloys Compd.*, 1997, **246**, 177; (g) C. Den Auwer, M. C. Charbonnel, M. G. B. Drew, M. Grigoriev, M. J. Hudson, P. B. Iveson, C. Madic,



- M. Nierlich, M. T. Presson, R. Revel, M. L. Russell and P. Thuéry, *Inorg. Chem.*, 2000, **39**, 1487; (h) M. P. Jensen and A. H. Bond, *J. Am. Chem. Soc.*, 2002, **124**, 9870; (i) I. Jarrige, H. Ishii, Y. Q. Cai, J.-P. Rueff, C. Bonnelle, T. Matsumara and S. R. Shieh, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 075122; (j) Y. G. Choi, K. A. Lee and K. S. Lee, *Met. Mater. Int.*, 2007, **13**, 269; (k) G. Mountjoy, J. M. Cole, T. Brennan, R. J. Newport, G. A. Saunders and G. W. Wallidge, *J. Non-Cryst. Solids*, 2001, **279**, 20; (l) C. H. Booth, D. Kazhdan, E. L. Werkema, M. D. Walter, W. W. Lukens, E. D. Bauer, Y.-J. Hu, L. Maron, O. Eisenstein, M. Head-Gordon and R. A. Andersen, *J. Am. Chem. Soc.*, 2010, **132**, 17537; (m) E. M. Broderick, P. S. Thuy-Boun, N. Guo, C. S. Vogel, J. Sutter, J. T. Miller, K. Meyer and P. L. Diaconescu, *Inorg. Chem.*, 2011, **50**, 2870; (n) J. I. Pacold, D. S. Tatum, G. T. Seidler, K. N. Raymond, X. Zhang, A. B. Stickrath and D. R. Mortensen, *J. Am. Chem. Soc.*, 2014, **136**, 4186; (o) Z. Lin, M. L. Shelby, D. Hayes, K. A. Fransted, L. X. Chen and M. J. Allen, *Dalton Trans.*, 2014, **43**, 16156; (p) J. Kratsch, B. B. Beele, C. Koke, M. A. Denecke, A. Geist, P. J. Panak and P. W. Roesky, *Inorg. Chem.*, 2014, **53**, 8949; (q) H. Asakura, T. Shishido, K. Teramura and T. Tanaka, *J. Phys. Chem. C*, 2015, **119**, 8070.
- 11 (a) S. G. Minasian, J. M. Keith, E. R. Batista, K. S. Boland, C. N. Christensen, D. L. Clark, S. D. Conradson, S. A. Kozimor, R. L. Martin, D. E. Schwarz, D. K. Shuh, G. L. Wagner, M. P. Wilkerson, L. E. Wolfsberg and P. Yang, *J. Am. Chem. Soc.*, 2012, **134**, 5586; (b) L. P. Spencer, P. Yang, S. G. Minasian, R. E. Jilek, E. R. Batista, K. S. Boland, J. M. Boncella, S. D. Conradson, D. L. Clark, T. W. Hayton, S. A. Kozimor, R. L. Martin, M. A. MacInnes, A. C. Olson, B. L. Scott, D. K. Shuh and M. P. Wilkerson, *J. Am. Chem. Soc.*, 2013, **135**, 2279; (c) S. G. Minasian, J. M. Keith, E. R. Batista, K. S. Boland, D. L. Clark, S. A. Kozimor, R. L. Martin, D. K. Shuh and T. Tylliszczak, *Chem. Sci.*, 2014, **5**, 351; (d) X.-D. Wen, M. W. Löble, E. R. Batista, E. Bauer, K. S. Boland, A. K. Burrell, S. D. Conradson, S. R. Daly, S. A. Kozimor, S. G. Minasian, R. L. Martin, T. M. McCleskey, B. L. Scott, D. K. Shuh and T. Tylliszczak, *J. Electron Spectrosc. Relat. Phenom.*, 2014, **194**, 81; (e) M. W. Löble, J. M. Keith, A. B. Altman, S. C. E. Stieber, E. R. Batista, K. S. Boland, S. D. Conradson, D. L. Clark, J. L. Pacheco, S. A. Kozimor, R. L. Martin, S. G. Minasian, A. C. Olson, B. L. Scott, D. K. Shuh, T. Tylliszczak, M. P. Wilkerson and R. A. Zehnder, *J. Am. Chem. Soc.*, 2015, **137**, 2506.
- 12 (a) S. Harder, D. Naglav, C. Ruspici, C. Wickleder, M. Adlung, W. Hermes, M. Eul, R. Pottengen, D. B. Rego, F. Poineau, K. R. Czerwinski, R. H. Herber and I. Nowik, *Chem.–Eur. J.*, 2013, **19**, 12272; (b) J.-H. Agondanou, G. A. Spyroulias, J. Purans, G. Tsikalas, C. Souleau, A. G. Coutsolelos and S. Benazeth, *Inorg. Chem.*, 2001, **40**, 6088; (c) A. P. Menushenkov, R. V. Chernikov, V. V. Sidorov, K. V. Klementiev, P. A. Aleksev and A. V. Rybina, *JETP Lett.*, 2006, **84**, 119; (d) E. Beaurepaire, J. P. Kappler and G. Krill, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1990, **41**, 6768.
- 13 (a) B. Qi, I. Perez, P. H. Ansari, F. Lu and M. Croft, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1987, **36**, 2972; (b) B. K. Agarwal, *X-ray Spectroscopy: An Introduction*, Springer-Verlag, 1991; (c) D. Ravot, C. Godard, J. C. Achard and P. Lagarde, in *Valence Fluctuations in Solids*, ed. H. M. Falicov, North-Holland Publishing Company, Amsterdam, 1981, p. 423.
- 14 J. J. Rehr, J. J. Kas, F. D. Vila, M. P. Prange and K. Jorissen, *Phys. Chem. Chem. Phys.*, 2010, **12**, 5503.
- 15 (a) M. R. Antonio, L. Soderholm and I. Song, *J. Appl. Electrochem.*, 1997, **27**, 784; (b) A. Fuse, G. Nakamoto, N. Ishimatsu and M. Kurisu, *J. Appl. Phys.*, 2006, **100**, 043712–043721; (c) A. I. Lebedev, I. A. Sluchinskaya, S. G. Nikitenko and S. G. Dorofeev, *Phys. Scr.*, 2005, **T115**, 365.
- 16 N. B. Mikheev, L. N. Auerman, I. A. Rumer, A. N. Kamenskaya and M. Kazakevich, *Russ. Chem. Rev.*, 1992, **61**, 990.
- 17 (a) S. Yoshida, T. Tanak, T. Hanada, T. Huraiwa, H. Kanai and T. Funabiki, *Catal. Lett.*, 1992, **12**, 277; (b) A. J. Davenport, A. J. Dent, N. Monir, J. A. Hammons, S. M. Ghahari, P. D. Quinn and R. Rayment, *J. Electrochem. Soc.*, 2011, **5**, C111; (c) I. Almhamid, J. C. Bryan, J. J. Bucher, A. K. Burrell, N. M. Edelstein, E. A. Hudson, N. Kaltsoyannis, W. W. Lukens, D. K. Shuh, H. Nitsche and T. Reich, *Inorg. Chem.*, 1995, **34**, 193.
- 18 (a) M. M. Grush, J. Chen, T. L. Stemmler, S. J. George, C. Y. Ralston, R. T. Stibrany, A. Gelasco, G. Christou, S. M. Gorun, J. E. Penner-Hahn and S. P. Cramer, *J. Am. Chem. Soc.*, 1996, **118**, 65; (b) S. Bertram, G. Kaindl, J. Jove and M. Pages, *Phys. B*, 1989, **158**, 508; (c) R. Kirsch, D. Fellhauer, M. Altmaier, V. Neck, A. Rossberg, T. Fanghanel, L. Charlet and A. C. Scheinost, *Environ. Sci. Technol.*, 2011, **45**, 7267; (d) D. Joseph, A. K. Yadav, S. N. Jha and D. Bhattacharyya, *Bull. Mater. Sci.*, 2013, **36**, 1067; (e) J. Wong, F. W. Lytle, R. P. Messmer and D. H. Maylotte, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1984, **30**, 5596; (f) S. P. Cramer, T. K. Eccles, F. W. Kutzler, K. O. Hodgson and L. E. Mortenson, *J. Am. Chem. Soc.*, 1976, **98**, 1287.
- 19 R. D. Shannon, *Acta Crystallogr.*, 1976, **A32**, 751.
- 20 A. J. Cohen, P. Mori-Sanchez and W. Yang, *Chem. Rev.*, 2012, **112**, 289.
- 21 V. B. Oyeyemi, J. A. Keith, M. Pavone and E. A. Carter, *J. Phys. Chem. Lett.*, 2012, **3**, 289.
- 22 (a) R. J. Magyar and S. J. Tretiak, *J. Chem. Theory Comput.*, 2007, **3**, 976; (b) A. Dreuw and M. Head-Gordon, *Chem. Rev.*, 2005, **105**, 4009; (c) I. H. Nayyar, E. R. Batista, S. Tretiak, A. Saxena, D. L. Smith and R. L. Martin, *J. Chem. Theory Comput.*, 2013, **9**, 1144.
- 23 Y. Zhao and D. G. Truhlar, *Theor. Chem. Acc.*, 2008, **120**, 215.
- 24 (a) J. L. Durant, *Chem. Phys. Lett.*, 1996, **256**, 595; (b) J. Zheng, Y. Zhao and D. G. Truhlar, *J. Phys. Chem. A*, 2007, **111**, 4632; (c) J. Zheng, Y. Zhao and D. G. Truhlar, *J. Chem. Theory Comput.*, 2009, **5**, 808.
- 25 P. D. Dau, J. Su, H.-T. Liu, D.-L. Huang, J. Li and L.-S. Wang, *J. Chem. Phys.*, 2012, **137**, 064315.



- 26 (a) J. W. Lauher and R. J. Hoffmann, *J. Am. Chem. Soc.*, 1976, **98**, 1729; (b) B. E. Bursten, L. F. Rhodes and R. J. Strittmatter, *J. Am. Chem. Soc.*, 1989, **111**, 2756; (c) B. E. Bursten, L. F. Rhodes and R. J. Strittmatter, *J. Am. Chem. Soc.*, 1989, **111**, 2758; (d) R. J. Strittmatter and B. E. Bursten, *J. Am. Chem. Soc.*, 1991, **113**, 552; (e) W. W. Lukens Jr and R. A. Andersen, *Organometallics*, 1995, **14**, 3435; (f) R. G. Denning, J. Harmer, J. C. Green and M. Irwin, *J. Am. Chem. Soc.*, 2011, **133**, 20644; (g) M. BenYahia, L. Belkhir and A. Boucekkine, *J. Mol. Struct.: THEOCHEM*, 2006, **777**, 61; (h) B. E. Bursten and R. J. Strittmatter, *Angew. Chem., Int. Ed.*, 1991, **30**, 1069; (i) L. Maron, O. Eisenstein and R. A. Andersen, *Organometallics*, 2009, **28**, 3629; (j) I. Kirker and N. Kaltsoyannis, *Dalton Trans.*, 2011, **40**, 124.
- 27 R. S. Mulliken, *J. Chem. Phys.*, 1955, **23**, 1833.
- 28 (a) M. Coreno, M. de Simone, J. C. Green, N. Kaltsoyannis, R. Coates, C. Hunston, N. Narband and A. Sella, *Dalton Trans.*, 2014, **43**, 5134; (b) Q.-Y. Wu, J.-H. Lan, C.-Z. Wang, Z.-P. Cheng, Z.-F. Chai, J. K. Gibson and W.-Q. Shi, *Dalton Trans.*, 2016, **45**, 3102; (c) S. A. Kozimor, P. Yang, E. R. Batista, K. S. Boland, C. J. Burns, D. L. Clark, S. D. Conradson, R. L. Martin, M. P. Wilkerson and L. E. Wolfsberg, *J. Am. Chem. Soc.*, 2009, **131**, 12125.
- 29 R. L. Martin and D. A. Shirley, Many electron theory of electron emission, in *Electron Spectroscopy: Theory, Techniques and Applications*, ed. C. R. Brundle and A. D. Baker, Academic Press, London, 1977, vol. 1, pp. 75–117.
- 30 M. Segala and D. P. Chong, *J. Electron Spectrosc. Relat. Phenom.*, 2010, **182**, 141.
- 31 T. Saue and L. Visscher, Relativistic All-Electron Approaches to the Study of f Element Chemistry. in *Computational Methods in Lanthanide and Actinide Chemistry*, ed. M. Dolg, Wiley, Chichester, UK, 2015, ch. 3, pp. 55–87.
- 32 (a) M. Keyes and W. B. Tolman, *Adv. Catal. Processes*, 1997, **2**, 189; (b) A. S. Borovik, *Comments Inorg. Chem.*, 2002, **20**, 45; (c) S. E. Gibson and M. P. Castaldi, *Chem. Commun.*, 2006, 3045; (d) C. C. Cummins, *Prog. Inorg. Chem.*, 1998, **47**, 685; (e) K. Meyer and S. C. Bart, *Adv. Inorg. Chem.*, 2008, **60**, 1; (f) P. T. Wolczanski, *Chem. Commun.*, 2009, 740; (g) J. P. Krogman and C. M. Thomas, *Chem. Commun.*, 2014, **50**, 5115; (h) P. Mayank and L. Que Jr, *Acc. Chem. Res.*, 2015, **48**, 2443.
- 33 (a) S. D. Stults, R. A. Andersen and A. Zalkin, *Organometallics*, 1990, **9**, 115; (b) S. G. Minasian, J. L. Krinsky, J. D. Rinehart, R. Copping, T. Tyliszczak, M. Janousch, D. K. Shuh and J. Arnold, *J. Am. Chem. Soc.*, 2009, **131**, 13767; (c) J. K. Peterson, M. R. MacDonald, J. W. Ziller and W. J. Evans, *Organometallics*, 2013, **32**, 2625; (d) J. L. Krinsky, S. G. Minasian and J. Arnold, *Inorg. Chem.*, 2011, **50**, 345.
- 34 W. J. Evans, J. W. Grate, H. W. Choi, I. Bloom, W. E. Hunter and J. L. Atwood, *J. Am. Chem. Soc.*, 1985, **107**, 941.
- 35 F. T. Edelmann, *Synth. Methods Organomet. Inorg. Chem.*, 1997, **6**, 37.
- 36 W. J. Evans, D. K. Drummond, H. Zhang and J. L. Atwood, *Inorg. Chem.*, 1988, **27**, 575.
- 37 W. J. Evans and N. T. Allen, *J. Am. Chem. Soc.*, 2000, **122**, 2118.
- 38 K. Izod, S. T. Liddle and W. Clegg, *Inorg. Chem.*, 2004, **43**, 214.
- 39 S. Calvin, *XAFS for everyone*, CRC Press, 2013.
- 40 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- 41 ADF 2014.11, <http://www.scm.com>.
- 42 C. F. Guerra, J. G. Snijders, G. te Velde and E. J. Baerends, *Theor. Chem. Acc.*, 1998, **99**, 391.
- 43 G. T. Velde, F. M. Bickelhaupt, E. J. Baerends, C. F. Guerra, S. J. A. van Gisbergen, J. G. Snijders and T. J. Ziegler, *Comput. Chem.*, 2001, **22**, 931.
- 44 E. van Lenthe and E. J. Baerends, *Comput. Chem.*, 2003, **24**, 1142.
- 45 A. D. Becke, *Phys. Rev. A*, 1988, **38**, 3098.
- 46 C. Lee, W. Yang and R. G. Parr, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1988, **37**, 785.
- 47 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 5648.
- 48 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 1372.
- 49 E. Ronca, C. Angeli, L. Belpassi, F. De Angelis, F. Tarantelli and M. Pastore, *J. Chem. Theory Comput.*, 2014, **10**, 4014.
- 50 P. J. LeStrange, P. D. Nguyen and X. Li, *J. Chem. Theory Comput.*, 2015, **11**, 2994.
- 51 E. van Lenthe, E. J. Baerends and J. G. Snijders, *J. Chem. Phys.*, 1993, **99**, 4597.
- 52 H. J. Werner, *Mol. Phys.*, 1996, **89**, 645.
- 53 P. Celani and H. J. Werner, *J. Chem. Phys.*, 2000, **112**, 5546.
- 54 H. J. Werner, P. J. Knowles, G. Knizia, F. R. Manby and M. Schütz, Molpro: a general-purpose quantum chemistry program package, *Wiley Interdiscip. Rev.: Comput. Mol. Sci.*, 2012, **2**, 242.
- 55 H.-J. Werner, P. J. Knowles, G. Knizia, F. R. Manby and M. Schütz, and others, *MOLPRO, version 2015.1*, a package of *ab initio* programs, see <http://www.molpro.net>.
- 56 T. H. Dunning Jr, *J. Chem. Phys.*, 1989, **90**, 1007.
- 57 M. Dolg, H. Stoll and H. Preuss, *J. Chem. Phys.*, 1989, **1730**, 90.
- 58 X. Cao and M. Dolg, *J. Mol. Struct.: THEOCHEM*, 2002, **581**, 139.
- 59 G. Ghigo, B. O. Roos and P. A. Malmqvist, *Chem. Phys. Lett.*, 2004, **396**, 142.

