The role of copper in the thermal conductivity of thermoelectric oxychalcogenides: do lone pairs matter?


Understanding the underlying mechanisms that suppress thermal conduction in solids is of paramount importance for the targeted design of materials for thermal management and thermoelectric energy conversion applications. Bismuth copper oxychalcogenides, BiOCuQ (Q = Se, Te), are highly crystalline thermoelectric materials with an unusually low lattice thermal conductivity of ~0.5 Wm⁻¹ K⁻¹, a value normally found in amorphous materials. Here we unveil the origin of the unusual thermal transport properties of these phases. First principles calculations of the vibrational properties combined with analysis of in-situ neutron diffraction data, demonstrate that weak bonding of copper atoms within the structure leads to an unexpected vibrational mode at low frequencies, which is likely to be a major contributor to the low thermal conductivity of these materials. In addition, we show that anharmonicity and the large Grüneisen parameter in these oxychalcogenides are mainly related to the low frequency copper vibrations, rather than to the Bi³⁺ lone pairs.

Introduction

Declining stocks of fossil fuels and concerns over CO₂ gas emissions pose a major challenge to society, which motivates the search for alternative energy generation technologies, including those for energy recovery. Thermoelectric power generation from waste heat offers one possible solution to the problem of increasing energy efficiency using existing energy-generation technologies. To enable widespread implementation of thermoelectric energy recovery, significant improvements in the performance of thermoelectric materials, which is related to a figure of merit, ZT = S²σ/κT/k (where S, σ and κ are the Seebeck coefficient, electrical and thermal conductivities respectively), are required. High ZT necessitates coexistence of high electrical conductivity and low thermal conductivity, an extremely unusual combination of properties. For this reason, understanding phonon transport in thermoelectric materials with ultralow thermal conductivities, such as Cu₁₂Sb₄S₁₃ and Cu₃SbSe₃, is attracting considerable interest. Bismuth copper oxychalcogenides, formulated as BiOCuQ or BiCuQO (Q = Se, Te) have been recently identified as promising thermoelectric materials. Aliovalent doping at the bismuth site can enhance their power factor and results in remarkable thermoelectric performances. High values of ZT have been obtained by doping with Sr³⁺, Ca²⁺ and Pb²⁺, while ZT = 1.4 has been achieved through texturing. The outstanding performance of these materials arises from their intrinsically low lattice thermal conductivity, whose origin remains unclear, despite recent studies of the phonon density of states of the oxyselenide. Previous experimental studies have suggested that low thermal conduction is due to the presence of the Bi³⁺ lone pair, which can reduce the lattice thermal conductivity due to bond anharmonicity. Here we will show, however, that anharmonicity is primarily linked to an unexpected copper vibrational mode; a feature that may lead to a new approach for the design of thermoelectric materials with enhanced ZT.

Experimental

Synthesis

Stoichiometric mixtures of Cu, Bi, Bi₂O₃ and a chalcogen (S, Se or Te) were ground prior to being sealed into evacuated fused...
silica tubes (~10⁻⁴ Torr). For the reaction producing BiOCuS, the sealed tube was first heated at 1 K min⁻¹ up to 673 K, and maintained at that temperature for 48 hours. After cooling the tube to room temperature, the sample was ground, pressed into a pellet and sealed into an evacuated silica tube for a second firing at 873 K for 48 hours. For BiOCuSe, the sealed tube was first heated at 623 K for 20 hours and then up to 773 K for 10 hours, using a 2 K min⁻¹ ramp rate. After grinding and rescaling the sample into a carbon-coated silica tube, it was heated at 873 K for 7 hours. For BiOCuTe, the tube was first heated up to 623 K with a ramp rate of 1 K min⁻¹. After 24 hours, the temperature was increased to 773 K, and the silica tube maintained at that temperature for 48 hours. After cooling, the sample was ground and sealed into an evacuated silica tube again for a final firing at 773 K for 72 hours. The quality of each sample was initially assessed using powder X-ray diffraction, which indicated that each sample consisted of highly-crystalline BiOCuQ (Q = S, Se, Te). For thermal transport measurements, the powders were hot pressed into well-densified pellets (~95% of theoretical density), under a N₂ flow. The following conditions were used: for BiOCuS, 823 K and 130 bar for 10 min; for BiOCuSe, 853 K and 50 bar for 30 min; for BiOCuTe, 773 K and 50 bar for 30 min.

**Powder neutron diffraction**

Neutron diffraction data for powdered samples of BiOCuQ were collected on the POLARIS diffractometer at the ISIS facility, Rutherford Appleton Laboratory. Approximately 3 g of each powdered sample was contained in an evacuated and sealed high-purity boron-free silica ampoule. The ampoule was contained in a vanadium can held inside a furnace evacuated to a pressure <10⁻⁴ Torr. Time-of-flight data were collected on heating from ambient temperature to 673 K, at the following temperatures: 293, 373, 473, 573, 623 and 673 K. Each dataset was collected for ~300 μA h proton beam current on the ISIS neutron target (approximately 2 hours duration). Data reduction and file output was carried out using the Mantid¹³ software package. Normalised diffraction patterns from the POLARIS backscattering, 90° and low angle detector banks (average 2θ = 146.7°, 92.6° and 52.2° respectively) were used simultaneously in Rietveld refinements carried out using the GSAS package.¹⁴ Final observed, calculated and difference profiles, together with final refined parameters, and selected bond distances and angles can be found in the ESI. The following expression, which corresponds to the Einstein model of a harmonic oscillator,¹⁵ has been used to determine the Einstein temperature of the copper atoms:

\[ U_{iso} = \frac{h^2 T}{(m \theta_E^2 4 \pi^2)} \]

where \( h \) and \( k_B \) are the Plank and Boltzmann constants, \( m \) is the mass and \( \theta_E \) is the Einstein temperature of the oscillator. The Einstein temperature can be determined from the slope of \( U_{iso}(T) \). The vibrational frequency is related to \( \theta_E \) by \( \nu = \frac{h}{k_B} \theta_E \). The Debye temperature of the stiff network formed by Bi, O and Q can be determined using the following expression:¹⁵

\[ U_{iso} = \left(3 h^2/m \theta_D^2 4 \pi^2\right)^2 \]

where \( \theta_D \) is the Debye temperature. The average velocity of sound in the Debye model is given by:

\[ v_s = \frac{[\theta_D k_B 2 \pi/k]}{\left[6 \pi^2 n \right]^{1/3}} \]

and evaluating the fractional contribution of the \( j \)th atom in the unit cell to the \( n \)th mode at a sampling point \( q \). Total energy response to atomic displacements was determined by shifting the ions along inequivalent tetragonal directions and computing the energy of the new configuration. The maps explore a range of displacements up to 0.4 Å.

**Results and discussion**

Three sets of experimental observations are especially relevant: (1) the total thermal conductivity of BiOCuQ (Fig. 1) is dominated by the lattice component, shows the \( 1/T \) dependence expected for crystalline solids at temperatures above the Debye temperature, and falls from the oxysulfide to the oxetalluride, following the normal trend of decreasing thermal conductivity with increasing atomic mass. (2) Measurements of the elastic properties of BiOCuSe indicate that this material has a low Young’s modulus (76.5 GPa at 300 K) suggesting weak bonding, and a high Gruneisen parameter of ~1.5,¹² which is indicative of bond anharmonicity. (3) Powder neutron diffraction data on...
BiOCuQ (ESI†), collected as a function of temperature, indicate that the BiOCuQ structure (Fig. 1) is retained over this temperature range. These data also reveal striking differences in the magnitude of the atomic displacement parameter (ADP). In particular, the ADP for copper, which does not show any significant anisotropy, is considerably larger than those of the remaining crystallographic sites. Although all ADPs increase linearly with temperature, the copper ADP increases at a faster rate (Fig. 1c). In sharp contrast to Cu$_{3}$Se or Cu$_{3}$SbSe$_{3}$, in which mobile copper atoms are not constrained to their equilibrium positions within the structure and can move to neighbouring sites,2,20,21 examination of the difference Fourier maps for BiOCuQ at high temperatures indicate that copper atoms remain in their crystallographic sites, and there is no evidence of structural disorder. This is confirmed by measurements of ionic conductivity in BiOCuSe, which has been found to be negligible,22 and indicates that the unusual thermal transport behaviour of BiOCuQ cannot be attributed to a liquid or part-liquid like state of the copper sublattice. This is highly significant for thermoelectric applications, given the detrimental effect of ion migration that irreversibly damages the legs of thermoelectric devices.23

In order to elucidate the origin of the low thermal conductivity, phonon vibrations for BiOCuQ (Fig. 2) were computed using first principles calculations. The main feature, common to all three compositions, is the presence of a very low frequency manifold of optic modes centred at ca. 60 cm$^{-1}$ (Fig. 3 and ESI†). Low frequency optic modes are known to exist in low-thermal-conductivity materials containing heavy ions.24,25 However, in BiOCuQ those vibrations have a large and unexpected copper contribution to the phonon DOS, which outweighs that of the heavier Bi atom, as it is six times larger than that of Bi in BiOCuS, three times larger in BiOCuSe, and twice in BiOCuTe (ESI†). The position of this low frequency optic manifold is in excellent agreement with the neutron data analysis. Assuming that copper behaves as a localised Einstein harmonic oscillator, the Einstein temperature, $\theta_E$, can be determined from the slope of a plot of $U_{iso}$ vs. temperature (Table 1). The determined $\theta_E$ corresponds to a frequency of ca. 60 cm$^{-1}$, in very good agreement with our computed values. By assuming that Bi, O and Q atoms constitute a stiff network, the average mass and $U_{iso}$ for those atoms can be used to estimate the Debye temperature, $\theta_D$ (Table 1). These estimates are in excellent agreement with the values computed within the quasi-harmonic approximation, and also agree with the limited literature data available ($\theta_D = 245$ K for BiOCuSe).26 The average velocity of sound estimated from $\theta_D$ is also in good agreement with measurements ($2170$ ms$^{-1}$ for BiOCuSe$^{11}$).

In order to evaluate the role of copper in the reduction of the lattice thermal conductivity, we computed the mode-resolved Grüneisen parameter and projected it onto the atoms (Table 2).

<table>
<thead>
<tr>
<th>Material</th>
<th>$\theta_E$/K</th>
<th>$v_{Cu}$/cm$^{-1}$</th>
<th>$\theta_D$/K</th>
<th>$v_s$/ms$^{-1}$</th>
</tr>
</thead>
<tbody>
<tr>
<td>BiOCuS</td>
<td>93</td>
<td>65</td>
<td>230 (231)</td>
<td>2100</td>
</tr>
<tr>
<td>BiOCuSe</td>
<td>95</td>
<td>66</td>
<td>210 (211)</td>
<td>2000</td>
</tr>
<tr>
<td>BiOCuTe</td>
<td>89</td>
<td>62</td>
<td>170 (196)</td>
<td>1700</td>
</tr>
</tbody>
</table>

Table 1: Einstein ($\theta_E$) and Debye ($\theta_D$) temperatures, together with the average velocity of sound ($v_s$) and the frequency of the localised copper mode ($v_{Cu}$), determined from analysis of neutron diffraction data. Values in parenthesis are $\theta_D$ computed from first-principle calculations.
Contrary to previous work which attributed the large Grüneisen parameter to the presence of the Bi$^{3+}$ lone pair,\textsuperscript{12} our results clearly indicate that the major contributors are copper and chalcogen, while the bismuth contribution is always small. The largest Grüneisen parameters at low frequency are found for BiOCuS and BiOCuSe (Fig. 3b), but the copper contribution for BiOCuTe is still comparatively high. This finding supports the existence of a copper rattling mode in BiOCuQ. At the root of this rattling mode is weak Cu–Q bonding, as evidenced by a number of findings: (1) the Cu–Q distances in BiOCuQ are longer than those reported for other compounds containing Cu$^+$ in a tetrahedral coordination. For instance, the Cu–Se distance in CuGaSe$_2$ is 2.385(1) Å,\textsuperscript{27} significantly shorter than the value of 2.5143(3) Å found for BiOCuSe (ESI†). This is consistent with the observation that empirical bond valence sums for BiOCuQ are too small and appear to underestimate the oxidation state of copper, hence suggesting that bonds are weaker. (2) Although BiOCuQ are layered compounds without cages, the $\theta_b$ of ca. 90 K for copper (Table 1) is comparable to that previously found (79 K) for the rattler atom located in the oversized cages of the skutterudite LaFe$_4$Sb$_{12.15}$ (3) Calculations that previously found (79 K) for the rattler atom located in the oversized cages of the skutterudite LaFe$_4$Sb$_{12.15}$ (3) Grüneisen parameters projected on the atoms. Values are computed using a finite difference formula using the phonon frequencies determined from the experimental lattice parameters at 293 K and at 673 K. Last column gives the total Grüneisen parameter per unit cell (2 formula units)

<table>
<thead>
<tr>
<th>ndsp</th>
<th>La/Bi</th>
<th>O</th>
<th>Cu</th>
<th>Q</th>
<th>TOT</th>
</tr>
</thead>
<tbody>
<tr>
<td>LaOCuS</td>
<td>0.13</td>
<td>0.16</td>
<td>0.37</td>
<td>0.34</td>
<td>2.00</td>
</tr>
<tr>
<td>LaOCuSe</td>
<td>0.15</td>
<td>0.17</td>
<td>0.40</td>
<td>0.27</td>
<td>1.98</td>
</tr>
<tr>
<td>BiOCuS</td>
<td>0.18</td>
<td>0.22</td>
<td>0.43</td>
<td>0.48</td>
<td>2.62</td>
</tr>
<tr>
<td>BiOCuSe</td>
<td>0.19</td>
<td>0.22</td>
<td>0.45</td>
<td>0.31</td>
<td>2.34</td>
</tr>
<tr>
<td>BiOCuTe</td>
<td>0.19</td>
<td>0.25</td>
<td>0.31</td>
<td>0.25</td>
<td>2.00</td>
</tr>
</tbody>
</table>

from that of copper, we have also analysed the vibrational properties of LaOCuQ. Our results show that the low frequency region is still dominated by copper modes as in BiOCuQ, and that the lanthanum vibrational modes are shifted to higher frequencies and overlap less with the copper modes (ESI†), when compared to BiOCuQ. This indicates that the change in atomic mass is at the origin of the variation in thermal conductivity associated with the Bi/La substitution. Calculations of the deformation energy indicate that both Bi–O and La–O bonds are reasonably stiff when compared to Cu–Q bonds (Fig. 2b), and the mode resolved Grüneisen parameter indicates that in all cases Cu and the chalcogen are the major contributors to anharmonicity.

The effects associated with the bismuth site may be, however, indirect.\textsuperscript{2} If the bismuth lone pair is indeed stereochemically active, the Coulomb interaction could couple the copper rattling with the bismuth displacement. The direct observation of a lone pair in the electronic structure is dubious,\textsuperscript{33} and this concept often translates into hybridization between O 2p and Bi 6s states, which has been observed in the projected density of states (PDOS).\textsuperscript{32} In addition, the PDOS of BiOCuS indicates cross gap hybridisation involving Bi 6p and O 2p as well as Cu and S p states (the second being larger). By contrast, in LaCuOS cross gap hybridization associated with La and O is negligible whereas it remains large for the Cu–S layer. A deviation of the Born effective charges from their nominal value is also a characteristic signature of stereochemically active ions. Those effects may lead to vibronic interactions, which involve the coupling of electronic and vibrational effects, and justify the increased anharmonicity, which will be associated with cross gap hybridization. Calculated Born effective charges (ESI†), show larger deviations from nominal values in BiCuOS with respect to LaCuOS, confirming that the lone pair in BiOCuQ is stereochemically active.

To further assess the possible Bi–Cu and La–Cu interplay in the anharmonicity, we studied the total energy response to atomic displacements. These maps (Fig. 4 and ESI†) represent the energy landscape associated with the simultaneous displacement of Bi/La and Cu ions. When cooperative distortions are favored the...
total energy change is small or negative, but as shown in Fig. 4, the Bi/La displacements are independent from the Cu displacement. These maps also provide further evidence that the energy landscape for the copper vibrations is very flat. The results presented here are in sharp contrast with the recent report of Cu–Sb bond length fluctuations in the tetrahedrite Cu12Sb4S13, in which low lattice thermal conductivity has been linked to an anharmonic rattling mode of copper, which is assisted by the lone pairs in the antimony atoms. This clear difference between Cu12Sb4S13 and BiOCuQ might be related to the layered structure of BiOCuQ, in which Bi and Cu are segregated into different layers (Fig. 1a).

Conclusions

Using a combination of theoretical calculations and neutron diffraction experiments, we have shown that the presence of a localised low-energy vibrational mode on the copper, rather than the presence of the Bi3+ lone pair, is the fundamental mechanism leading to low thermal conductivity in these oxychalcogenides. This localised mode, which occurs in layered materials containing no cages, is a consequence of weak copper bonding. These oxychalcogenides could be described as incipient ionic conductors, in which copper mobility may be suppressed due to the absence of a suitable diffusion path. For decades, the search for novel thermoelectrics has been based on rattling mechanisms associated with heavy elements in cages (e.g. skutterudites, clathrates). We have provided compelling qualitative and quantitative evidence that layered compounds, with no cages, may exhibit a similar rattling behavior, arising from suppressed ionic conduction. This mechanism for the reduction of lattice thermal conductivity, which has been identified here for the first time, offers an alternative approach for the design of thermoelectric materials with $ZT > 1$.

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Notes and references


13 Details of the Mantid open source framework supporting high-performance computing to manipulate and analyse neutron and muon scattering data may be found at http://dx.doi.org/10.5286/SOFTWARE/MANTID.


