

Modifying the properties of 4f single-ion magnets  
by peripheral ligand functionalisation†Cite this: *Chem. Sci.*, 2014, 5, 1650Kasper S. Pedersen,<sup>a</sup> Liviu Ungur,<sup>b</sup> Marc Sigrist,<sup>ac</sup> Alexander Sundt,<sup>d</sup> Magnus Schau-Magnussen,<sup>a</sup> Veaceslav Vieru,<sup>b</sup> Hannu Mutka,<sup>c</sup> Stephane Rols,<sup>c</sup> Høgni Weihe,<sup>a</sup> Oliver Waldmann,<sup>d</sup> Liviu F. Chibotaru,<sup>b</sup> Jesper Bendix<sup>\*a</sup> and Jan Dreiser<sup>\*e</sup>

We study the ligand-field splittings and magnetic properties of three Er<sup>III</sup> single-ion magnets which differ in the peripheral ligand sphere but exhibit similar first coordination spheres by inelastic neutron scattering (INS) and SQUID magnetometry. The INS spectra of the three compounds are profoundly different pointing at a strong response of the magnetic behavior to minor structural changes, as they are e.g. encountered when depositing molecules on surfaces. The observation of several magnetic excitations within the  $J = 15/2$  ground multiplet together with single-crystal magnetic measurements allows for the extraction of the sign and magnitude of all symmetry-allowed Stevens parameters. The parameter values and the energy spectrum derived from INS are compared to the results of state-of-the-art *ab initio* CASSCF calculations. Temperature-dependent alternating current (ac) susceptibility measurements suggest that the magnetisation relaxation in the investigated temperature range of  $1.9 \text{ K} < T < 5 \text{ K}$  is dominated by quantum tunnelling of magnetisation and two-phonon Raman processes. The possibility of observing electron paramagnetic resonance transitions between the ground-state doublet states, which can be suppressed in perfectly axial single-ion magnets, renders the studied systems interesting as representations of quantum bits.

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## Introduction

The strong magnetic anisotropy of 4f ions is an essential basis for the properties of technologically applied magnetic materials. Magnetic anisotropy is determined by the local ligand (or crystal) field and a detailed understanding is of paramount importance to achieve control over it. Within the field of molecular magnetism, single 4f ions shielded from adjacent magnetic centres, typically by organic ligands, have gained interest due to the observation of intrinsic slow relaxation of the magnetisation with energy barriers for magnetisation reversal of as much as  $652 \text{ cm}^{-1}$ .<sup>1</sup> Such molecular systems are commonly referred to as mononuclear single-molecule magnets (SMMs) or single-ion magnets (SIMs). Recently, promising results for the incorporation of 4f SIMs in spintronics devices<sup>2</sup> have opened up questions

regarding the sensitivity of the magnetic properties towards small perturbations unavoidable in the anchoring or deposition of the molecules to e.g. a nanotube or a substrate. The strong response of the magnetic properties to the ligand field (LF) in traditional 4f magnets also holds true for lower-dimensional, molecular magnets like 4f SIMs.<sup>3</sup> This has initiated several approaches to estimate LF splittings by calculations from e.g. *ab initio*<sup>4</sup> or charge distributions.<sup>5</sup> *Ab initio* calculations on 3d clusters have shown the crucial role played by the second coordination sphere.<sup>6a</sup> Using single-crystal measurements, corroborated by *ab initio* calculations, Sessoli and co-workers demonstrated for 4f complexes exceedingly high sensitivity of the magnetic anisotropy towards the LF and that simple magneto-structural correlations used for transition element SMMs may fail completely for predicting e.g. the direction of the easy-axis of magnetisation in 4f SIMs.<sup>1b,6</sup> These results present a challenge for the design and control of nanostructured devices relying on 4f SIMs and point to the need for a detailed understanding of the sensitivity of the LF towards small changes of the first and second coordination sphere. Mimicking the environment experienced by a 4f SIM outside the native crystallographic environment is a complicated task. Because of the difficulties in acquiring structural information of metal ion complexes in solution or for surface-attached systems, such studies are best performed by structural modifications in the solid state. Nevertheless, even in the solid state the most detailed understanding of the magnetic properties of 4f systems is in most cases hampered by the lack of fundamental knowledge

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about the eigenvalues and eigenvectors of the single-centre ligand field operators. To obtain this information, the luminescence pertinent to most 4f ions has gained increasing popularity as a spectroscopic tool to estimate LF splittings in SIMs, but the successful parameter determination is significantly limited by the spectroscopic resolution, possible occurrence of “hot” transitions and non-radiative decay.<sup>1b,7,8</sup> Furthermore, the 4f–4f luminescence is not always accessible and can be screened by strong ligand-centred optical transitions as in the case of phthalocyaninate systems. In solid-state physics, information on the LF levels in 4f systems has traditionally been acquired by inelastic neutron scattering (INS) but, surprisingly, this technique has not been used to study any LF spectra of 4f SIMs.<sup>9</sup>

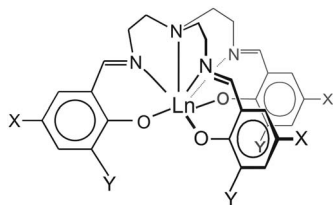
Here, a small class of structurally similar Er<sup>III</sup> SIMs has been investigated by INS spectroscopy and magnetometry. The compounds differ by peripheral ligand modifications and the presence or lack of crystallographic trigonal symmetry of the 4f centre. The parent complex; Er(trensal) (**1**) (H<sub>3</sub>trensal = 2,2',2''-tris(salicylideneimino)triethylamine), which is part of an isostructural series,<sup>10</sup> was studied by Riley and co-workers who parameterised the ligand field based on optical spectroscopy, but never reported on the magnetic properties.<sup>11</sup>

The trensal<sup>3-</sup> back-bone is relatively rigid when coordinating to lanthanide ions and can easily be functionalised.<sup>12</sup> An attractive feature of **1** is the presence of a crystallographic three-fold symmetry of the Er<sup>III</sup> ion (*P3c1* space group) and the concomitant reduced number of symmetry-allowed LF parameters. Further, high-resolution optical spectra are available from which all possible LF parameters were unraveled.<sup>11a</sup> Additionally, using the slightly modified ligand system 2,2',2''-tris(3-iodo-5-methylsalicylideneimino)triethylamine (3-I,5-Me-trensalH<sub>3</sub>) gives Er(3-I,5-Me-trensal) (**2**, cf. Scheme 1) again being trigonal (*P3* space group). On the contrary, chlorine-substituted 2,2',2''-tris(5-chlorosalicylideneimino)triethylamine (5-Cl-trensalH<sub>3</sub>) yields Er(5-Cl-trensal) (**3**), which crystallises in the monoclinic *P2<sub>1</sub>/c* space group with no axial, local symmetry. For these reasons, the presented compounds are ideal test beds for the systematic study of LF perturbations induced by minute modifications of the LF geometry and strength.

## Experimental section

### Synthesis

All starting materials were purchased from commercial sources and used without further purification. The large-scale synthesis



**Scheme 1** Pictorial representation of **1–3**: **1** X = Y = H; **2**: X = CH<sub>3</sub>, Y = I; **3**: X = Cl, Y = H. The three-fold rotation axis in **1** and **2** lies along the axial, tertiary amine N–Ln bond.

of **1** was first performed by the procedure published by Bernhardt *et al.*<sup>10a</sup> However, this procedure often gave powder samples containing significant amounts of an unidentifiable phase. For this reason, we employed another, modified literature procedure of Kaneshato and Yokoyama which also proved useful to obtain large single crystals suitable for single-crystal SQUID magnetometry.<sup>13</sup> In addition, all employed samples were rigorously characterised by single-crystal or powder X-ray diffraction and elemental analysis. Er(CF<sub>3</sub>SO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O (0.50 g, 0.64 mmol) and tris(2-aminoethyl)amine (“tren”, 0.19 g, 1.3 mmol) were refluxed in acetonitrile (15 ml) for 15 min. The solution was cooled and placed at the bottom of a 35 ml glass tube (Ø ≈ 8 mm) and layered with acetonitrile (*ca.* 20 ml) and salicylaldehyde (0.24 g, 2.1 mmol). Large pencil-shaped crystals developed over a week. For **2**, 3-iodo-5-methylsalicylaldehyde was synthesized as described in literature.<sup>14</sup> For 3-I,5-Me-trensal, to 3-iodo-5-methylsalicylaldehyde (7.5 g, 29 mmol) dissolved in boiling methanol (100 ml) was added tris(2-aminoethyl)amine (1.5 g, 10 mmol). After cooling to room temperature, the crystalline, yellow product was isolated by filtration and washed with methanol. Yield: 7.8 g (88%). Calc. (found) for C<sub>30</sub>H<sub>33</sub>I<sub>3</sub>N<sub>4</sub>O<sub>3</sub> (%): C, 41.02 (41.04); H, 3.79 (3.48); N, 6.38 (6.33). Subsequently, Er(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O (220 mg, 0.50 mmol) was dissolved in methanol (20 ml) and added to a boiling methanol (150 ml) solution of 3-I-5-Me-trensal (0.40 g, 0.46 mmol) and triethylamine (0.14 g, 1.4 mmol). The boiling was continued for 5 min, the mixture was cooled to RT and **2** was isolated by filtration and washed with methanol. Yield: 0.44 g (88%). Polycrystalline samples of **3** were synthesised as reported for the Gd analogue and proven to be isostructural from X-ray powder diffraction. The structural data for the Gd analogue with Gd replaced by Er was employed in the *ab initio* calculations for **3**.<sup>15</sup> Recrystallisation from boiling methanol afforded single crystals suitable for structure determination (cf. Table S1 and Fig. S3†). This phase is found to be different from the powdered sample used in the INS and magnetic measurements and no further studies were performed on this phase. The diamagnetic Y analogues (**1'–3'**) were synthesized similarly and shown by X-ray powder diffraction to be isostructural to the Er systems (cf. Fig. S4–6†). Er-doped **1'** samples were obtained similarly. Elemental analysis results (%): Calc. (found) for **1**: C, 52.07 (52.01); H, 4.37 (4.01); N, 9.00 (8.98). Calc. (found) for **2**: C, 34.56 (34.86); H, 2.90 (2.60); N, 5.37 (5.38). Calc. (found) for **3** (C<sub>27</sub>H<sub>24</sub>Cl<sub>3</sub>ErN<sub>4</sub>O<sub>3</sub>): C, 44.66 (44.53); H, 3.33 (2.95); N, 7.72 (7.65). Calc. (found) for **1'** (C<sub>27</sub>H<sub>27</sub>N<sub>4</sub>O<sub>3</sub>Y): C, 59.56 (58.86); H, 5.00 (4.95); N, 10.29 (10.52). Calc. (found) for **2'**: C, 37.37 (37.47); H, 3.14 (2.79); N, 5.81 (5.77). Calc. (found) for **3'**: C, 50.06 (49.90); H, 3.73 (3.40); N, 8.64 (8.58).

### X-Ray crystallography

Single-crystal X-ray diffraction studies were performed at 122(1) K on a Nonius Kappa CCD area-detector diffractometer (equipped with an Oxford Cryostreams low-temperature device, using graphite-monochromated Mo-K<sub>α</sub> radiation (λ = 0.71073 Å) or on a Bruker D8 VENTURE diffractometer equipped with Mo-K<sub>α</sub> high-brilliance IμS (micro-source) radiation (λ = 0.71073 Å), a



multilayer X-ray mirror and a PHOTON 100 CMOS detector, and an Oxford Cryosystems low-temperature device. The instrument was controlled with the APEX2 software package. The structures were solved using direct methods (SHELXS97) and refined using the OLEX2 programme.<sup>16</sup> All non-hydrogen atoms were refined anisotropically, whereas H-atoms were isotropic and constrained. Crystal structure and refinement data for **1** (122 K), **2** (122 K) and **3** (122 K, recrystallised phase) are summarised in Table S1.† Powder X-ray crystallographic data were collected on a Ge Bruker D8 Advance Powder diffractometer operating in  $2\theta$ - $\theta$  configuration using  $\text{Co-K}\alpha$  ( $\lambda = 1.7902 \text{ \AA}$ ) radiation.

## Elemental analysis

Elemental analysis for C, H and N was performed with a CE Instrument: FLASH 1112 series EA, at the microanalysis laboratory, University of Copenhagen.

## Magnetic measurements

**SQUID magnetometer and PPMS.** The magnetic data shown in all figures except S51 and S52 were acquired on a Quantum-Design MPMS-XL SQUID magnetometer and a Quantum Design physical property measurement system (PPMS). Magnetisation data were obtained with selected fields from 0 to 5 T and ac data obtained with frequencies 1–1500 Hz (MPMS) and up to 10 kHz (PPMS) with an ac field amplitude of 3.0–3.8 Oe (MPMS) and 10 Oe (PPMS) with or without applications of static field. Single crystals were checked by single-crystal X-ray diffraction for phase purity before each measurement. Single-crystal magnetic data were obtained with a horizontal rotator setup. Prior to all measurements the response of the rotator was determined and subtracted from the response in the real experiment. Polycrystalline samples were immobilised in hexadec-1-ene in polycarbonate capsules. The diamagnetic contribution to the sample moment from the sample holder and sample was corrected through background measurements and Pascal constants, respectively. Paramagnetic relaxation times were obtained from  $\chi''(\nu_{\text{ac}})$  data as  $\tau(T) = [2\pi\nu_{\text{max}}(T)]^{-1}$ .

**Hall magnetometer.** The field dependence of the isothermal magnetisation of a single crystal of **1** of a mass of 0.26 mg was measured with a home-built micro-Hall magnetometer at temperatures between 1.4 and 8.4 K. The sweeping rate of the external magnetic field was varied in the range from 150 to 2300 mT s<sup>-1</sup>.

## Inelastic neutron scattering

INS spectra were obtained on the time-of-flight spectrometers IN4 and IN5 located at Institut Laue-Langevin, Grenoble, France. Several grams of non-deuterated samples were measured in double-wall Al/Mg sample cans or as wrapped in Al foil at selected temperatures and incoming neutron wavelengths. The Y<sup>III</sup> analogues **1'**, **2'** and **3'** were systematically measured with the same settings to facilitate a precise assignment of the phonon spectra. The data were reduced and analysed using the Large Array Manipulation Program (LAMP).<sup>17</sup> Magnetic (de)excitations were localised by their characteristic  $|Q|$  (linear momentum transfer) dependence which follows the

magnetic form factor, their temperature variation and the comparison with the spectra of the diamagnetic Y<sup>III</sup> analogues.

## Electron paramagnetic resonance

EPR spectra were acquired on a Bruker Elexsys E500 equipped with a Bruker ER 4116 DM dual mode cavity, an EIP 538B frequency counter and a ER035M NMR Gauss-meter. The spectra were simulated or fitted using home-written software.<sup>18</sup> Large single crystals of Er<sup>III</sup> doped into **1'** were obtained as described for **1** but employing  $\text{Y}(\text{CF}_3\text{SO}_3)_3 \cdot 9\text{H}_2\text{O}$  with a 5 mol% presence of  $\text{Er}(\text{CF}_3\text{SO}_3)_3 \cdot 9\text{H}_2\text{O}$ .

## Modelling

In order to compare the LF obtained from the published optical spectra<sup>11a</sup> as well as that from *ab initio* calculations we will use sets of extended Stevens operator coefficients or, in short, Stevens parameters. This involves a reduction of the quantum-mechanical basis set to the states of the Er<sup>III</sup> ground-state multiplet  $^4\text{I}_{15/2}$ . Such a procedure is justified because the temperature range used in our measurements lies far below the energies of the first excited  $^4\text{I}_{13/2}$  multiplet ( $\sim 6000 \text{ cm}^{-1}$ ).

**Zero-field splitting Hamiltonian.** In the following we will refer to the zero-field splitting (zfs) Hamiltonian as the extended Stevens operators parameterisation with only the ground-state multiplet as basis (*vide infra*). In contrast, the LF Hamiltonian operates on the full space of the  $4f^{11}$  configuration with all possible multiplets of Er<sup>III</sup>.

The following procedure was applied to convert the published LF parameterisation into Stevens parameters: The energy spectrum of **1** was calculated using the published LF model and best-fit parameters<sup>11a</sup> by using the same software written by M. F. Reid.<sup>19</sup> The eigenvalues and eigenvectors in the  $|L, S, J, m\rangle$  basis using 284 states at energies of up to  $\sim 50000 \text{ cm}^{-1}$  were exported into a MATLAB program.<sup>20</sup> The LF Hamiltonian  $H_{\text{LF}}$  in matrix form was obtained by back transformation  $H_{\text{LF}} = A H_{\text{diag}} A^\dagger$  with  $A$  the unitary matrix containing the eigenvectors as columns and  $H_{\text{diag}}$  the diagonal matrix of the corresponding eigenvalues. To obtain the zfs Hamiltonian only the coefficients in the eigenvectors  $v_i$  referring to the ground-state multiplet  $^4\text{I}_{15/2}$  were considered and the resulting new eigenvectors were renormalized to  $\text{norm}(v_i) = 1$ .

As mentioned later in the text least-squares fits were performed by minimising the sum of weighted and squared deviations  $\kappa^2 = \sum_i (y_{\text{obs},i} - y_{\text{calc},i})^2 \times w_i$  with  $y_{\text{obs},i}$  and  $y_{\text{calc},i}$  the experimental and calculated values of the  $\chi T(T)$  product, field-dependent magnetisation and of the energies of the INS transitions. The weight  $w_i$  results from the inverse square of the estimated experimental errors. In the case of **1** also the published energy spectrum was included as observations and a smaller weight ( $w_i = 0.03$  vs. 0.1) was given to the high-energy part of the spectrum which could not be observed by INS.

**Effective spin-1/2 Hamiltonian.** In this common simplified formalism given by  $\hat{H}_{\text{eff}} = \mu_0 \mu_B (g_{\parallel} \hat{\tau}_z H_z + g_{\perp} \hat{\tau}_x H_x + g_{\perp} \hat{\tau}_y H_y)$  the zfs is omitted and the magnetic anisotropy is now present in the  $g$ -factor.  $\hat{\tau}$  has the properties of a spin-1/2 angular momentum mapped on to the lowest Kramers doublet. The coordinate



frame is chosen to be identical with that of the zfs Hamiltonian eqn (1) (*vide infra*), i.e., the z-axis coincides with the molecular  $C_3$  axis.

All spin-Hamiltonian calculations shown in this work are based on full diagonalisation. The matrices of the extended Stevens operators were generated with the help of the *stev* function from the EasySpin package written by Stefan Stoll.<sup>21</sup> The powder average of the magnetisation for large fields was generated using a 110-point Lebedev–Laikov grid.<sup>22</sup>

**Ab initio calculations.** The *ab initio* calculations were carried out using the Molcas 7.8 package program.<sup>23</sup> The calculations for **1** were performed using the measured crystal structures at 122 and 293 K. The differences in the results are negligible (see ESI†). For **2** the measured crystal structure at 122 K and for **3** the structure of the Gd analogue with Gd replaced by Er were used. All elements were described using standard basis sets from the ANO-RCC library available in Molcas. The TZP basis set was employed for Er and first coordination sphere atoms, DZP for the atoms involved in the aromatic rings around the central atom and DZ for the other atoms. Contractions of the employed basis sets are given in Table S2.† In order to save disk space, the Cholesky decomposition of bielectronic integrals was employed with a threshold of  $0.5 \times 10^{-7}$ . The spin-free wave functions and corresponding energies were calculated within the Complete Active Space Self-Consistent Field (CASSCF) method.<sup>24</sup> The active space of the CASSCF method included 11 electrons of  $\text{Er}^{\text{III}}$  spanning seven 4f-type orbitals. The spin-orbit interaction was considered within the Douglas–Kroll–Hess Hamiltonian in the mean-field approximation. The spin-orbit coupling was taken into account within the restricted active space state interaction (RASSI) method,<sup>25</sup> by mixing all spin-quartet states (35) and all spin-doublet states (112). On the basis of the resulting spin-orbital multiplets, the SINGLE\_ANISO<sup>26</sup> program was used to compute the magnetic properties and the LF parameters. Further, we tested the effect of the crystal environment (Madelung potential) on the low-lying energy states. To this end, the Madelung potential was approximated by five layers of point charges. Every point charge was placed at the Cartesian position of each atom belonging to the crystal environment. The charge value assigned to each atom was the calculated Mulliken charge of the corresponding atom in the ground state, taken from the previous CASSCF calculation on the individual molecule. The reason for describing the Madelung potential in this way is to give a realistic charge distribution in the crystal environment formed by neutral molecules.

## Results and discussion

The condensation products of tris(2-aminoethyl)amine (“tren”) with aromatic aromatic aldehydes have been extensively employed as ligands for single lanthanide ions<sup>12</sup> and bicompartamental systems.<sup>27</sup> However, only few of the systems possess the high symmetry that is beneficial for the present type of study. For the unsubstituted, pristine **1**, the crystal structure at room temperature was reported by Kanesato and Yokoyama and our redetermination of the structure at 122 K revealed only small differences with respect to the 300 K structure. The

structure is shown in Scheme 1 and Fig. 1 (Fig. S1†). **1** crystallises in the  $P\bar{3}c1$  space group and possesses crystallographic three-fold ( $C_3$  point group) symmetry of the seven-coordinate  $\text{Er}^{\text{III}}$  ion. For the novel **2** and **3**, the space groups are  $P\bar{3}$  and  $P2_1/c$ , respectively, and the Er sites have local  $C_3$  and  $C_1$  symmetries. The bond lengths and angles of the first coordination sphere are almost identical for **1–3** as visible in Fig. 1 (Fig. S1–S3†). The root-mean-square deviations of the central  $\text{Er}^{\text{III}}$  ion and the first coordination sphere are 0.082 Å (**1** vs. **2**) and 0.17 Å (**1** vs. **3**), respectively, illustrating the strong similarity of the first coordination spheres. Importantly, the Er– $\text{N}_{\text{amine}}$  distances of 2.67–2.70 Å the average Er– $\text{N}_{\text{imine}}$  of 2.46–2.47 Å, and the Er– $\text{O}_{\text{phenolate}}$  of 2.18–2.19 Å are all in the typical range of Ln–N and Ln–O distances. Hence the results of the present study can be considered representative for these ligators.

The ground state of  $\text{Er}^{\text{III}}$  is well described by a  $^4I_{15/2}$  Russell–Saunders term. In the absence of any symmetry the enormous number of 27 LF parameters is allowed to be nonzero and the precise determination of a unique parameter set is certainly impossible. In the presence of three-fold rotational symmetry ( $C_3$ ), the number of symmetry-allowed terms drops dramatically to 9. However, magnetisation and susceptibility data on most 4f systems, especially on polycrystalline samples, are not very distinct, hence the unambiguous determination of 9 parameter values still remains a challenge. For compound **1**, high-resolution optical spectra are available from which all possible LF parameters were unraveled by Riley and co-workers.<sup>11a</sup> The LF parameters were fitted to several multiplets, however, here we are concerned with the magnetic properties which are largely determined by only the ground multiplet. Hence all excited multiplets can be safely neglected. This can be understood by considering that the first excited  $^4I_{13/2}$  multiplet is separated by more than  $6000 \text{ cm}^{-1}$  from the ground state while the full span

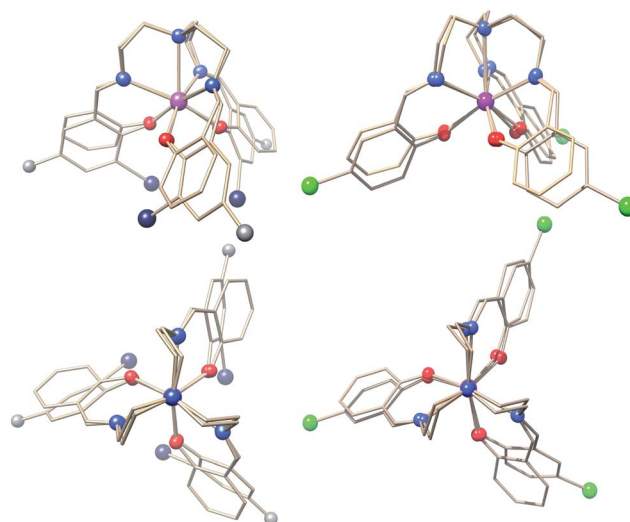


Fig. 1 Structural overlays of **1** with **2** (left figures) and **3** (powder phase, right figures). The projections at the bottom are along the three-fold rotation axis of **1**. Colour code: Er, purple; I, dark blue; Cl, green; O, red; N, blue; C, grey. With the exception of the substituents, the remaining parts of the trensal<sup>3–</sup> ligands are shown as wireframe for clarity.





of the LF-split  $^4I_{15/2}$  ground multiplet is around  $600\text{ cm}^{-1}$ .<sup>11a</sup> The limited resolution of the optical spectroscopy and approximations that result from the translation of LF into zero-field splitting (zfs) parameters lead to only a rough prediction of the magnetic properties which are extremely sensitive to small changes in the LF. Therefore, an optimised set of zfs parameters for the ground-state multiplet of **1** is sought by fitting the zfs Hamiltonian simultaneously to the neutron and optical spectroscopic data as well as to the temperature and field dependence of the magnetisation.

In order to compare the *ab initio* calculated LF with that from optical spectroscopy both were translated into an extended Stevens operator description working on the  $^4I_{15/2}$  ground multiplet

$$\hat{H}_{\text{zfs}} = \sum_{k,-k \leq q \leq k} B_k^q \hat{O}_k^q \quad (1)$$

Magnetic field was taken into account in the phenomenological LF model by the effective Zeeman Hamiltonian  $\hat{H}_z = g_{\text{eff}}\mu_0\mu_{\text{B}}\hat{\mathbf{J}} \cdot \mathbf{H}$  while an exact microscopic expression has been used in *ab initio* calculations.<sup>4</sup> From  $\hat{H} = \hat{H}_z + \hat{H}_{\text{zfs}}$  the magnetic properties and the INS spectra were calculated. Further, least-squares fits to the magnetic data and the neutron spectra were performed to find optimised parameter sets. In the fits, the  $B_4^{-3}$  parameters were fixed to zero which corresponds to a fixing of the coordinate frame hence it can be done without loss of generality. To allow for a comparison, the *ab initio* calculated parameter sets and that obtained from optical spectroscopy were transformed by rotations of the coordinate frame in order to give  $B_4^{-3} = 0$ .

### Static properties of **1**

The INS excitation spectra for **1** and **1'** obtained at 1.5 K are shown in Fig. 2 and S7–12.† For **1**, the spectrum is dominated by two prominent transitions labelled as P1 and M1 and located at 31 and  $53\text{ cm}^{-1}$ , respectively. The peaks are intense and nicely visible although the experiments were performed on non-deuterated samples. The phonon background can be estimated by downscaling a high-temperature spectrum according to the Bose factor<sup>28</sup>  $[1 - \exp(-\hbar\omega/k_{\text{B}}T)]^{-1}$  as shown in Fig. 2. Phononic and magnetic peaks are distinguished by studying the dependence of their intensity on temperature (Fig. S9–11†), on linear momentum transfer  $Q$  (Fig. S7 and 8†), and by comparison with the estimated phonon background and with the purely phononic spectrum of the nonmagnetic **1'**. Accordingly, phononic and magnetic features are labelled as  $P_n$  and  $M_n$ , respectively. The temperature dependence of M1–3 indicates that the associated transitions are excitations from the ground state to excited states (“cold transitions”). Depending on the nature of the phonon modes, their energy is a function of the mass of the involved 4f ion, thereby explaining the small difference between the spectra of **1** and **1'** around the feature P1. The positions of M1, M2 and M3 of 53, 102 and  $111\text{ cm}^{-1}$  are in perfect agreement with the energies reported by Flanagan *et al.* of 54, 102 and  $110\text{ cm}^{-1}$ .<sup>11a</sup> Given the precise knowledge of the energy spectrum it is interesting to perform a correlation with the magnetic properties. Also this provides one of the rare opportunities to further examine the performance of the LF

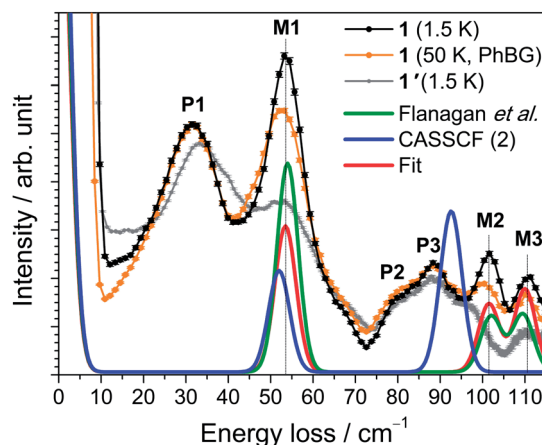


Fig. 2 INS spectra of **1** (black trace) and **1'** (grey) acquired with  $\lambda_i = 2.2\text{ Å}$  at  $T = 1.5\text{ K}$ . The intensity was summed over the complete available  $Q$  range. The calculated phonon background from the 50 K spectrum (see main text) is shown as an orange curve (PhBG). The lower red, blue and green traces are simulations employing different sets of Stevens parameters as described in the text.

parameters reported by Flanagan *et al.* as well as state-of-the-art CASSCF/RASSI-SO calculations. The *ab initio* calculated crystal field components of the  $J = 15/2$  multiplet in **1** taking into account the structure obtained at  $T = 293\text{ K}$  are given in Table 1. The last column indicates the multiplet energies extracted from luminescence spectra. The agreement with *ab initio* calculated energies is remarkably good for the lowest four Kramers doublets (*cf.* Fig. 2). Also the calculated energies arising from higher multiplets ( $J = 13/2, 11/2$ ) compare very well (*cf.* Table 1). Further we have employed the experimental structure determined at low temperature ( $T = 122\text{ K}$ ) for similar *ab initio* calculations. The resulting energy spectrum (Table S3†) and extended Stevens parameters (Table S4†) are similar to those obtained for the room-temperature structure.

The magnetic properties of **1** measured as field-dependent magnetisation  $M(H)$  and product of magnetic susceptibility and temperature ( $\chi T$ ) on a polycrystalline sample and on a single crystal parallel and perpendicular to the crystallographic three-fold axis are shown in Fig. 3 and 4. The temperature dependence of the magnetisation and the corresponding fits using Hamiltonian eqn (1) are shown in Fig. S25.†

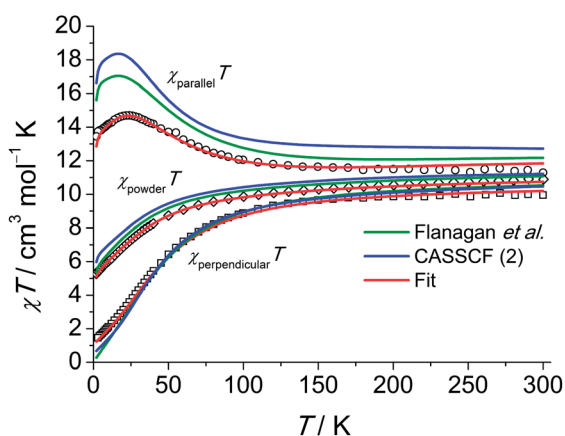
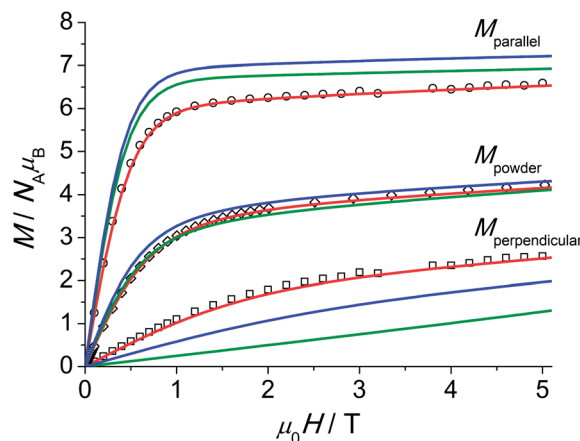
The single-crystal magnetisation data are clearly proving the presence of strong magnetic anisotropy. Further, at 300 K the  $\chi T$  exhibits still a distinct anisotropy confirming the overall span of the LF split ground multiplet being comparable to the thermal energy at room temperature. The X-band EPR spectra of an  $\text{Er}^{\text{III}}$ -doped **1'** (*ca.* 5 mol%) single-crystal provide  $g_{\parallel} = 11.8$  and  $g_{\perp} = 3.53$  (*cf.* Fig. S28†) in excellent agreement with the low-temperature part of the  $\chi T$  data. Furthermore, the observation of an EPR signal from the ground-state doublet directly confirms that the ground-state Kramers doublet transforms like the  $\Gamma_{4,5}$  irreducible representation<sup>29</sup> as stated in ref. 11a. The calculation of the magnetisations parallel and perpendicular to the three-fold axis using the effective spin-1/2 Hamiltonian with the  $g$ -values from EPR slightly underestimates the magnetisation compared to the experimental data (*cf.* Fig. S26†). This is particularly severe at



**Table 1** *Ab initio* calculated and observed energies of low-lying Kramers doublets in **1** (293 K structure) in units of  $\text{cm}^{-1}$ 

Free-ion multiplet	One molecule CASSCF (1)	One molecule embedded in 5 layers of point charges CASSCF (2)	Observed (ref. 11a)
$^4\text{I}_{15/2}$	0	0	0
	64	52	54
	99	91	102
	103	93	110
	198	203	299
	421	410	568
	459	448	610
	484	472	642
$^4\text{I}_{13/2}$	6652	6650	6594
	6687	6679	6612
	6692	6683	6621
	6705	6705	6690
	6903	6892	6909
	6904	6893	6928
	6917	6906	6939
$^4\text{I}_{11/2}$	10 716	10 711	10 291
	10 724	10 722	10 301
	10 738	10 732	10 316
	10 863	10 854	10 444
	10 869	10 859	10 449
	10 883	10 872	10 510

larger field due the admixture with excited states leading to a change in the wavefunctions (second-order effects). The fitted Stevens parameter set for **1** using the zfs Hamiltonian eqn (1) yields nearly perfect agreement with all magnetic and spectroscopic data. The effective  $g$ -factors of the lowest Kramers doublet derived from the fitted Stevens parameters are  $g_{\parallel} = 11.9$  and  $g_{\perp} = 3.36$  which coincide almost perfectly with the values found from EPR. The *ab initio* calculated main values are  $g_{\parallel} = 13.68$  and  $g_{\perp} = 2.28$ . The  $g$ -factor anisotropy is somewhat overestimated, which can be due to the following reason: the first excited Kramers

**Fig. 3** Experimental  $\chi T$  ( $\chi = M/H$ ,  $H = 2000$  Oe) products for a single crystal of **1** along (○) and perpendicular (□) to the  $C_3$  axis and for a polycrystalline sample of **1** (◇,  $H = 1000$  Oe). For the polycrystalline sample the  $\chi T$  was calculated as  $(\chi_{xx}T + \chi_{yy}T + \chi_{zz}T)/3$ . Calculations using eqn (1) are shown as solid lines as indicated in the plot.**Fig. 4** Experimental magnetisation at  $T = 2.0$  K for **1** along (○) and perpendicular (□) to the  $C_3$  axis and for a polycrystalline sample (◇). The powder average of the magnetisation ( $M$ ) was obtained by averaging over a spherical surface. The colour labelling of the simulations and fit is identical to that employed in Fig. 3.

doublet possesses opposite magnetic anisotropy with respect to the ground one, being of easy plane type ( $g_{\parallel} = 2.87$  and  $g_{\perp} = 7.62$ ). This means that small modifications of the LF will strongly influence the magnetic anisotropy of the ground multiplet given the proximity of the first excited Kramers doublet. The calculations employing the low-temperature structure show a similar trend (Table S5†).

An overview of the Stevens parameters obtained for **1** by *ab initio* calculations, from optical spectroscopy and from fits to the magnetic and INS data is given in Table 2. There is good agreement between the different parameter sets. Parameters forbidden in  $C_3$  symmetry come out to be non-zero from *ab initio* calculations because of limited numerical accuracy due to employment of Cholesky decomposition of the bielectronic integrals,<sup>23</sup> however, they are small compared to most of the allowed parameters of the same rank  $k$ . Since the comparison of sets of Stevens parameters is rather abstract it is of great help to examine the performance of the *ab initio* calculations and LF parameters from optical spectroscopy in reproducing the magnetic and INS data. Notably, as visible from the magnetisation and susceptibility data, the parameter sets obtained from optical spectra and *ab initio* calculations suggest slightly stronger anisotropy than what is consistently obtained from single-crystal magnetic data and EPR measurements.

### Static properties of **2**

The INS spectra of **2** and the dc magnetic data are shown in Fig. 5 and 6, respectively. The detailed temperature dependence of the INS spectra of **2** and **2'** is given in Figs. S13–17.† The neutron spectra of **2** (Fig. 5) are remarkably different from those of **1**. The magnetic and phononic features were disentangled using the same criteria and procedures as for **1**. In the  $2.8 \text{ \AA}$  spectra, a prominent peak, M2, is observed at  $70 \text{ cm}^{-1}$  and a weaker feature M1 at  $37 \text{ cm}^{-1}$ . In the  $1.4 \text{ \AA}$  spectrum the magnetic features are not very strong. An additional magnetic feature M3 at an energy loss of  $140 \text{ cm}^{-1}$  can be identified (cf. Fig. S13†).

Least-squares fits of all  $C_3$ -allowed Stevens parameters to the magnetic data and the INS spectra of **2** were performed as described for compound **1**. The best-fit parameters and the corresponding simulated curves are shown in Table 2 and in Fig. 5 and 6. By using random initial values we were able to obtain three parameter sets only slightly differing in  $\kappa^2$  (cf. Modelling Section) that reproduce all available data very well. A comparison of the three sets yields that the 'axial' parameters  $B_2^0$ ,  $B_4^0$  and  $B_6^0$  are quite robust, while the 'non-axial' parameters are fluctuating. The *ab initio* calculated magnetic properties and INS spectra, which were based on the 122 K structural data, are presented in Figs. S32–S34.† While the calculated magnetic properties are only slightly deviating from the experimental data, the INS spectra do not match well. Given that the *ab initio* predictions for the similar compounds **1** and **2** are expected to be of equal accuracy, the likely explanation for the larger discrepancy in **2** is that the employed 122 K structural data differs significantly from the geometry at 1.5 to 40 K, where the INS experiments were performed. In such a situation INS and optical spectroscopic data are indispensable for an accurate description of the electronic spectra of lanthanide SIMs.

### Static properties of **3**

For **3**, a representative INS spectrum at 2.2 Å is shown in Fig. 7. More INS data of **3** and **3'** are shown in Fig. S18–S22.† The comparison of the spectra of **3** and **3'** reveals magnetic

excitations at 29 cm<sup>−1</sup> (M1), 76 cm<sup>−1</sup> (M2) and 106 cm<sup>−1</sup> (M3). An inspection of the temperature dependence indicates that M1–3 are cold transitions. Comparing the 1.1 Å spectra indicate no further magnetic excitations up to energy transfers of ca. 500 cm<sup>−1</sup>. However, in the temperature dependence of the 2.8 Å spectra, a magnetic excitation (m4) starting out from an excited state ("hot" transition) was located at 48 cm<sup>−1</sup>. The parameters derived from *ab initio* calculations, done for the room-temperature structure of Gd(5-Cl-trensal)<sup>15</sup> in which Gd has been replaced by Er, do not satisfactorily reproduce the experimental data (cf. Tables S9–S11 and Fig. S35–S37†). Structural changes due to the exchanged central ion, or changes not detectable in powder XRD and/or due to the low temperatures in the experiments are obvious reasons for this discrepancy. A fit to the magnetic data and INS spectra using the *ab initio* derived parameters as initial values stops in a local minimum. To extract LF parameters within our phenomenological model, fits using random initial values were performed. The best results are plotted as solid lines in Fig. 6 and 7. While there are several parameter sets which lead to an excellent reproduction of the  $M(H)$  and  $\chi T$ , the INS plays a decisive role here. Only one of the parameter sets reproduces well also the warm INS transition m4 (cf. Fig. S20†). In these fits, only parameters allowed in  $C_3$  symmetry were considered, which is certainly a strong approximation in view that this compound actually possesses lower symmetry. However the introduction of more fit parameters would not lead to a significant improvement of the fits since a

Table 2 Stevens coefficients  $B_k^q$  for compounds **1**–**3** in units of cm<sup>−1</sup>. CASSCF calculations are based on 122 K structural data

$k, q$	Compound <b>1</b>			Compound <b>2</b>		Compound <b>3</b>
	<i>ab initio</i> CASSCF (2)	Flanagan <i>et al.</i> <sup>11a</sup>	Best fit	<i>ab initio</i> CASSCF (2)	Best fit	Best fit
2, −2	$3.780 \times 10^{-3}$			$3.244 \times 10^{-3}$		
2, −1	$-5.321 \times 10^{-3}$			$6.266 \times 10^{-3}$		
2, 0	−0.8776	−0.975	−1.07(2)	$-2.824 \times 10^{-1}$	$-1.9(6) \times 10^{-1}$	1.22(2)
2, 1	$5.058 \times 10^{-3}$			$-3.124 \times 10^{-3}$		
2, 2	$3.439 \times 10^{-4}$			$1.775 \times 10^{-3}$		
4, −4	$1.274 \times 10^{-4}$			$-1.964 \times 10^{-4}$		
4, −3	0	0	0	0	0	0
4, −2	$1.273 \times 10^{-4}$			$-1.894 \times 10^{-4}$		
4, −1	$-7.482 \times 10^{-5}$			$1.359 \times 10^{-4}$		
4, 0	$-1.010 \times 10^{-3}$	$-0.260 \times 10^{-3}$	$-0.2(2) \times 10^{-3}$	$-1.461 \times 10^{-3}$	$-6.0(6) \times 10^{-3}$	$-3.9(5) \times 10^{-3}$
4, 1	$6.359 \times 10^{-5}$			$-2.101 \times 10^{-5}$		
4, 2	$8.629 \times 10^{-5}$			$-3.777 \times 10^{-5}$		
4, 3	0.1770	0.287	0.255(5)	$1.707 \times 10^{-1}$	$1.1(4) \times 10^{-1}$	$1(1) \times 10^{-2}$
4, 4	$-1.121 \times 10^{-4}$			$1.348 \times 10^{-5}$		
6, −6	$1.555 \times 10^{-5}$	$5.89 \times 10^{-4}$	$3.5(5) \times 10^{-4}$	$4.235 \times 10^{-5}$	$1.6(7) \times 10^{-4}$	$1.2(1) \times 10^{-4}$
6, −5	$1.335 \times 10^{-5}$			$-4.616 \times 10^{-6}$		
6, −4	$-4.404 \times 10^{-7}$			$5.479 \times 10^{-7}$		
6, −3	$-1.165 \times 10^{-4}$	$-2.45 \times 10^{-4}$	$-1(2) \times 10^{-4}$	$-2.341 \times 10^{-4}$	$1.3(2) \times 10^{-3}$	$7(7) \times 10^{-5}$
6, −2	$-2.360 \times 10^{-6}$			$1.853 \times 10^{-6}$		
6, −1	$3.101 \times 10^{-6}$			$-6.209 \times 10^{-6}$		
6, 0	$9.685 \times 10^{-5}$	$1.23 \times 10^{-4}$	$1.25(1) \times 10^{-4}$	$8.896 \times 10^{-5}$	$9.5(6) \times 10^{-5}$	$-1.07(3) \times 10^{-4}$
6, 1	$-2.971 \times 10^{-6}$			$1.485 \times 10^{-6}$		
6, 2	$-1.892 \times 10^{-6}$			$-1.123 \times 10^{-6}$		
6, 3	$-0.8741 \times 10^{-3}$	$-0.943 \times 10^{-3}$	$-1.27(2) \times 10^{-3}$	$-5.176 \times 10^{-4}$	$-1(5) \times 10^{-5}$	$3.13(8) \times 10^{-6}$
6, 4	$8.189 \times 10^{-7}$			$-8.547 \times 10^{-7}$		
6, 5	$7.960 \times 10^{-6}$			$4.747 \times 10^{-6}$		
6, 6	$0.8663 \times 10^{-3}$	$1.03 \times 10^{-3}$	$1.27(2) \times 10^{-3}$	$7.316 \times 10^{-4}$	$4.2(8) \times 10^{-4}$	$1.4(1) \times 10^{-5}$
$g_{\text{Er}}$	6/5	6/5	1.18(1)	6/5	1.18(1)	1.18(1)



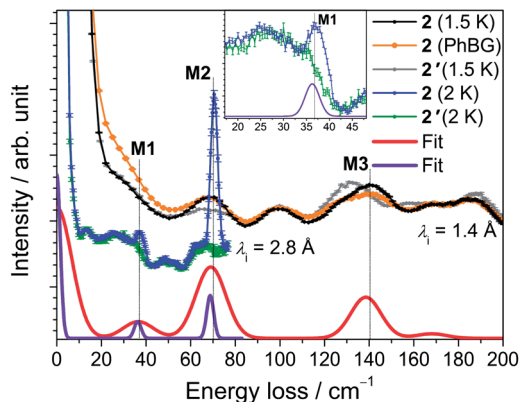


Fig. 5 INS spectra of **2** (black and blue traces) and **2'** (grey and green traces) obtained with incident neutron wavelengths of 1.4 Å (at  $T = 1.5$  K) and 2.8 Å (at  $T = 2$  K). The curves are offset to improve clarity. The calculated phonon background from the 40 K spectrum is shown as a solid orange line. The red and purple lines are simulations with resolutions comparable to the 1.4 and 2.8 Å data set, respectively, as described in the text. Inset: Zoom of the low-energy region of the 2.8 Å spectra around peak M1.

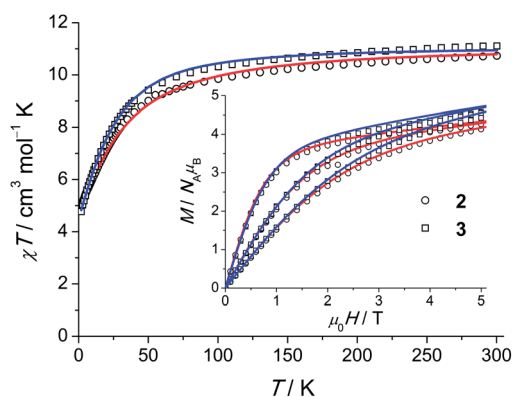


Fig. 6  $\chi T$  ( $\chi = M/H$ ,  $H = 1000$  Oe) products for polycrystalline **2** and **3**. Open symbols correspond to the experimental data, solid lines are calculated curves obtained from the model of eqn (1). Field-dependent magnetisation  $M(H)$  at 2, 4 and 6 K is shown in the inset.

good fit can already be achieved with the current parameter set. In such a situation, in order to find values of all 27 LF parameters *ab initio* calculations are indispensable. Our study shows that the ligand field spectrum of lanthanides is highly sensitive to the structural details. These may not be correctly captured by the structures of isostructural compounds as emphasized by the example of compound **3**.

### Dynamic magnetic properties of 1–3

The temperature dependencies of the magnetisation relaxation times of **1** (polycrystalline and as oriented single-crystal), Er-doped **1'** and **3** measured by ac SQUID magnetometry are shown in Fig. 8. In the absence of a static field none of the compounds **1**–**3** showed a peak in the out-of-phase component ( $\chi''$ ) of the ac susceptibility (Fig. S42, S47 and 48†). In small dc fields, clear maxima in  $\chi''(\nu_{ac})$  are observed for **1** and **3**, whereas

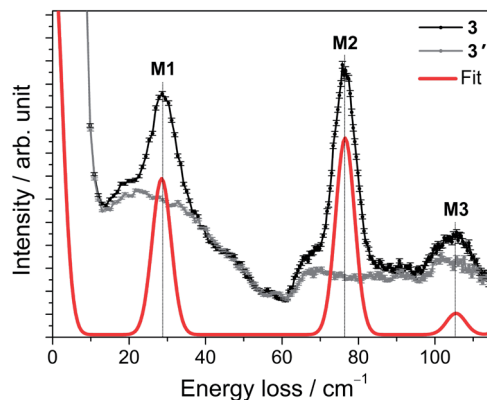


Fig. 7 INS spectra of **3** (black trace) and **3'** (grey trace) acquired with an incident neutron wavelength of 2.2 Å at  $T = 1.5$  K. The red line is the simulation using the best-fit parameters as described in the text.

only onsets of peaks are observed in **2** with  $\nu_{ac} \leq 1.5$  kHz at 1.8 K (cf. Fig. S47†). The magnetisation curves of an aligned (magnetic field applied parallel to the  $C_3$  axis) single crystal were also measured using a micro-Hall sensor at temperatures down to 1.4 K and magnetic field sweep rates up to  $2.3 \text{ T s}^{-1}$ . However, magnetic hysteresis was not observed (cf. Fig. S51 and 52†). A polycrystalline sample of 10% Er<sup>III</sup> doped into **1'** shows almost the same temperature dependence as pristine **1**. The field dependence (cf. Fig. S50†), however, hints at a weak effect of intermolecular interactions, which are suppressed in the diluted sample. For this reason we focus here on the latter compound, in order to investigate the mechanism that dominates the magnetisation relaxation.

With the frequently made assumption of an Orbach process and a temperature-independent quantum tunnelling of magnetisation (QTM) as dominant relaxation processes, a good fit can be obtained yielding a barrier for magnetisation reversal of  $\Delta_{\text{eff}} = 20(1) \text{ cm}^{-1}$ . The size of this barrier is, however, not compatible with the observed excited-state separation of  $54 \text{ cm}^{-1}$  and an Orbach process can be ruled out as a dominating

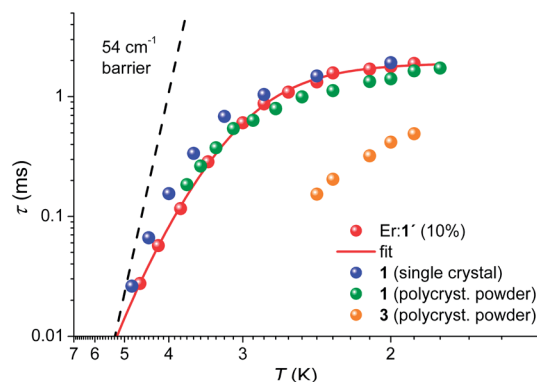


Fig. 8 Arrhenius plot for **1** as a single crystal measured with  $H_{dc} = 900$  Oe applied along the  $C_3$  axis, as polycrystalline material ( $H_{dc} = 900$  Oe), and as Er doped into **1'** (Er:1',  $H_{dc} = 1100$  Oe). For **3**, the data were acquired on a polycrystalline sample only ( $H_{dc} = 700$  Oe). The solid red line represents the best fit to the Er:1' data using the model of eqn (2). The dashed solid line is the Arrhenius prediction  $\tau(T) = 9\text{ps} \times \exp[54 \text{ cm}^{-1}/(k_B T)]$ .



relaxation mechanism in the studied temperature range of 1.9 K <  $T$  < 5 K. In view of this narrow temperature range and the gradually increasing slope towards higher temperatures it might however be possible that the Orbach process becomes the leading mechanism at elevated temperatures. It has been proposed theoretically and experimentally that first and second-order Raman processes may be significant.<sup>30</sup> Indeed we find that the relatively simple model for the temperature-dependent magnetisation relaxation times, eqn (2):

$$\tau^{-1} = \tau_{\text{QTM}}^{-1} + CT^n \quad (2)$$

gives an excellent fit to the experimental data on the Er-doped **1'**. The first term takes into account temperature-independent QTM while the second term represents a two-phonon Raman process. The best-fit parameters extracted using eqn (2) are  $C = 0.17(3) \text{ s}^{-1} \text{ K}^{-n}$ ,  $n = 8.0(1)$  and  $\tau_{\text{QTM}} = 1.93(5) \text{ ms}$  and the corresponding calculated curve is shown in Fig. 8. For a two-phonon Raman process an exponent of  $n = 9$  was predicted for Kramers ions<sup>30a</sup> but  $n$  may vary depending on the exact energies of the lowest Kramers doublets.<sup>30b</sup> Experimentally the  $T^n$  power law was indeed found with an exponent deviating from  $n = 9$  in most cases.<sup>30c,d,g</sup> From this viewpoint the value of the exponent found here is in reasonable agreement with the expectations for two-phonon Raman processes. Also, the values for  $C$  and  $\tau_{\text{QTM}}$  are in reasonable ranges.<sup>30d</sup> Further confirmation for the importance of a two-phonon Raman process comes from fits for Er-doped **1'** using a complete model (*cf.* Fig. S50†).<sup>30</sup> Here, all four processes (QTM, Orbach with  $\Delta_{\text{eff}} = 54 \text{ cm}^{-1}$ , direct and Raman) were included and the fit took into account simultaneously both temperature and field dependent relaxation time data. A fit assuming only an Orbach mechanism, the direct process, and QTM fails to reproduce the data. For both compounds a more or less pronounced increase of the relaxation times with field is visible reaching a maximum at fields of *ca.* 1 kOe, followed by a strong decay upon further increase of the field. Modelling these data is challenging because of the complexity induced by the simultaneous presence of more than one relaxation process. Similar to other studies (*e.g.* ref. 30e and f) the data on Er-doped **1'** can be understood by the quenching of tunneling in the low-field regime and the direct process becoming dominant at high fields. The complete fits reproduce the data very well, and the derived parameters for the four relaxation mechanisms demonstrate that both the Orbach and direct processes are small as compared to the QTM and Raman processes (for details see ESI†). The hyperfine field of  $B_{\text{hyp,ac}} = 17 \text{ mT}$  deduced from the  $B_2$  parameter describing the field dependence at low fields is consistent with  $B_{\text{hyp,EPR}} = 26 \text{ mT}$  extracted from EPR measurements. The observations hence illustrate nicely the presence of a non-Orbach process in a 4f SIM at low temperatures corroborated by the spectroscopic proof. Notably, this behavior parallels Na [Dy(DOTA)]·4H<sub>2</sub>O in which the first excited state was estimated from room-temperature luminescence spectra.<sup>1b</sup>

When comparing the magnetic and spectroscopic properties of compounds **1–3** studied in this work, it becomes clear that they are vastly different. This is also reflected in the results of the *ab initio* calculations. Interestingly, the introduction of

substituent groups far from the first coordination sphere leads to drastic modifications of the low-lying energy spectrum, hence the changes are not at all just a small perturbation to the system. The strong effect of the second coordination sphere on the anisotropy of the metal ion in a Co<sup>II</sup><sub>2</sub>(calix-8-arene)<sub>2</sub> complex was found by *ab initio* calculations to originate from the significantly stronger chemical bond within the ligand group than between the ligand atoms and the metal ion.<sup>6a</sup> Accordingly, in lanthanide complexes the sensitivity of the magnetic anisotropy to modifications in the second (and more distant) coordination sphere can be expected to be even higher. As depicted in Fig. 1 the first coordination sphere of the series **1–3** is largely similar. Therefore, the differences in the properties are likely to arise from two contributions. One comes from the difference in the electron withdrawing or donating effects imposed by the substituents and another from the subtle structural distortions of the first and second coordination sphere. The latter contributions are likely to be associated with changes in  $\pi$  interactions that are expected to be strongly dependent on the tilting of the aromatic rings. As an evidence for this picture, changes in the first coordination sphere have induced strong changes in the magnetic behaviour for Tb<sup>III</sup>-phthalocyaninate systems, where a contraction of the ligand sphere lead to an increase of the barrier for magnetisation reversal.<sup>31</sup> A few recent studies further suggest that subtle changes in the far range can also have strong effects on lanthanide anisotropy.<sup>32</sup> A strong response of the <sup>57</sup>Fe Mössbauer spectra was detected in a Fe<sub>2</sub>Dy<sub>2</sub> SMM upon peripheral ligand substitution suggesting a concomitant change of the lanthanide anisotropy and 3d–4f exchange coupling.<sup>33</sup> Moreover, similar effects were proposed on the basis of an *ab initio* study on the SIM Na[Dy(DOTA)]·4H<sub>2</sub>O.<sup>1b</sup> These recent studies already shed some light onto this topic but it was so far difficult to imagine that the spectroscopic changes were so profound.

These results should have an impact on the current efforts of studying surface-deposited SMMs and SIMs and the grafting of such species on to carbon nanotubes and other nanostructured materials. It certainly cannot be taken for granted that the magnetic properties remain unchanged after the grafting or surface deposition process even if the molecular stoichiometry is retained and the molecules seem to be intact. Our study suggests that even if the first coordination sphere remains unchanged after the surface deposition, the interaction of the ligand with the surface may have similar effects as the ligand functionalisation studied here. However, it should also be noted that the present results can be seen as an opportunity to use the surface in order to modify the magnetic properties of the deposited SMMs or SIMs. Currently, this still seems to be out of reach mainly because the understanding of the observed effects is underdeveloped. Controlling and harnessing such interactions may open up paths to achieving unprecedented spintronic applications.

## Conclusions

In summary, an extensive study of the magnetic and low-energy spectroscopic properties of a small family of Er<sup>III</sup> SIMs has been



performed. The measured data are compared to results from previously published optical spectroscopy as well as to state-of-the-art *ab initio* CASSCF/RASSI-SO calculations. The peripheral ligand functionalisation leads to largely unexpected drastic changes in all properties. These results are important in view of the on-going grafting of SMMs and SIMs on to surfaces and nano-objects suggesting that changes far away from the first coordination sphere can lead to vastly different magnetic properties. If control over these effects can be achieved, it may open a path to obtaining superior magnetic properties by using advantageous combinations of SIMs and surfaces.

Moreover, strong evidence is delivered that the Orbach mechanism is of minor importance for the relaxation of magnetisation in these systems at temperatures  $1.9\text{ K} < T < 5\text{ K}$ . In contrast, it appears that the relaxation is dominated by tunnelling of magnetisation at the lowest temperatures and that two-phonon Raman processes are dominating at temperatures close to 4 K.

## Note added after submission

We became aware of an accepted manuscript by E. Lucaccini *et al.* about Er(trensal) and Dy(trensal) complexes.<sup>34</sup>

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