

# A fast operation of nanometer-scale metallic memristors: highly transparent conductance channels in Ag<sub>2</sub>S devices†

Attila Geresdi, Miklós Csontos,\* Agnes Gubicza, András Halbritter and György Mihály

Cite this: *Nanoscale*, 2014, 6, 2613Received 24th October 2013  
Accepted 4th December 2013

DOI: 10.1039/c3nr05682a

www.rsc.org/nanoscale

The nonlinear transport properties of nanometer-scale junctions formed between an inert metallic tip and an Ag film covered by a thin Ag<sub>2</sub>S layer are investigated. Suitably prepared samples exhibit memristive behavior with technologically optimal ON and OFF state resistances yielding to resistive switching on the nanosecond time scale. Utilizing point contact Andreev reflection spectroscopy, we studied the nature of electron transport in the active volume of memristive junctions showing that both the ON and OFF states correspond to truly nanometer-scale, highly transparent metallic channels. Our results demonstrate the merits of Ag<sub>2</sub>S nanojunctions as nanometer-scale memory cells which can be switched by nanosecond voltage pulses.

The ongoing miniaturization beyond the limitations of current complementary metal-oxide-semiconductor (CMOS) technology is a major challenge in nanoscience.<sup>1–3</sup> Using individual atoms or molecules in nanoelectronic circuits has been a breakthrough towards the ultimate single atomic size limit.<sup>4,5</sup> The persisting technological difficulties in the reliable assembly of low resistance single molecule devices, however, still represent a major barrier to fast applications preferring low *RC* time constants with the capacitance of the environment. Alternatively, reversible solid state electrochemical reactions have been proposed to form tunable atomic scale junctions between metallic electrodes. The first results are extremely promising for the short term realization of highly integrated information storage applications.<sup>6–13</sup> The resistive state of a memory element, called *memristor*,<sup>6,8,9,14–16</sup> is altered by biasing the device above its writing threshold ( $V_{th}$ ). Readout is performed at lower signal levels which preserve the stored information.

Chalcogenide compounds have been put forward in the context of the “atomic switch”,<sup>7</sup> consisting of an inert metallic

electrode (Me) and an Ag layer capped with the solid state ionic conductor Ag<sub>2</sub>S. Upon positively biasing the Ag electrode with respect to Me, a metallic Ag propulsion is grown on the Ag<sub>2</sub>S–Me interface shunting the electrodes thus creating the non-volatile ON state of the device.<sup>17–21</sup> Real-time high resolution transmission electron microscopy (HRTEM) imaging<sup>22</sup> and first-principles band structure calculations<sup>23,24</sup> suggested that a structural phase transition in the Ag<sub>2</sub>S layer also plays a role in the resistive transition.<sup>25</sup>

Since the pioneering experiments reported on Ag–Ag<sub>2</sub>S–Pt devices<sup>7</sup> the development of memory cells based on memristive systems has achieved a remarkable progress. Besides providing an interesting model system for neural networks,<sup>26–28</sup> Ag<sub>2</sub>S based devices have been utilized as nanometer-scale non-volatile memory elements.<sup>7,21</sup> However, the best performing Ag<sub>2</sub>S devices<sup>7</sup> have only been operated up to ~10 MHz frequencies, presumably due to OFF state resistances typically  $\geq 100$  k $\Omega$ . These OFF state resistances require the monitoring of technically unfavorable low currents and give rise to larger *RC* time constants which are inconvenient for GHz applications.<sup>29</sup> In a tantalum oxide based system sub-nanosecond switching times were shown<sup>13</sup> in a significantly larger, lithographically defined structure whose operation relies on the reconfiguration of oxygen vacancies. Here we demonstrate devices which not only approach the atomic size limit but also the time scale of GHz operation.

We studied resistive switching in voltage biased nanojunctions created between inert metallic tips and 10–100 nm thick Ag<sub>2</sub>S surface layers deposited on Ag thin film samples by using an STM setup as illustrated schematically in the inset of Fig. 1(a). While longer sulfur deposition times resulted in semiconducting Ag<sub>2</sub>S layers in agreement with previous reports,<sup>18–20</sup> below an approximate thickness of 20 nm, metallic conductance with technologically optimal device resistances was found over the wide temperature range of 4.2–300 K (ref. 30) both in the ON and OFF states. These characteristics enabled fast operation by nanosecond voltage pulses. Here we focus on switching phenomena observed in all-metallic junctions. We

Department of Physics, Budapest University of Technology and Economics and Condensed Matter Research Group of the Hungarian Academy of Sciences, Budafoki ut 8, 1111 Budapest, Hungary. E-mail: csontos@dept.phy.bme.hu

† Electronic supplementary information (ESI) available: Sample preparation and structural characterization as well as involved experimental and analysis details. See DOI: 10.1039/c3nr05682a



used a superconducting Nb tip to study the nonlinear differential conductance on the voltage scale of the superconducting gap,  $eV \leq \Delta = 1.4 \text{ meV} \ll V_{\text{th}}$ . By utilizing the theory of charge conversion at the interface of a normal metal and a superconductor<sup>31–34</sup> we quantitatively evaluate the reconfiguration of the conducting channels in the nanojunction and thus demonstrate that resistive switching takes place in highly transparent devices with an effective junction area of 2–5 nm in diameter.

Numerous nanoscale contacts with reproducible  $I$ - $V$  characteristics were created by gently touching the sample surface with a mechanically sharpened PtIr or Nb tip. The ON and OFF state resistances  $R_{\text{ON}}$  and  $R_{\text{OFF}}$  were probed in a narrow voltage window of  $\pm 50 \text{ mV} \ll V_{\text{th}}$ . For more experimental details see the ESI† and ref. 30 and 35.

A typical room temperature  $I$ - $V$  trace is shown in Fig. 1(a) for a PtIr tip. A linear current-voltage dependence was observed when an increasing positive voltage on the Ag electrode in the high resistance ( $R_{\text{OFF}} \approx 0.5 \text{ k}\Omega$ ) state was applied. At  $V_{\text{th}} \approx 300 \text{ mV}$  the junction switches to its low resistance ON state ( $R_{\text{ON}} \approx 0.1 \text{ k}\Omega$ ). At a subsequent decrease of the bias a linear dependence is observed until the negative threshold voltage is reached where the OFF state is restored. The slight reverse of the onset of OFF to ON switching arises due to the compensation for the finite 50–200  $\Omega$  serial resistance of the voltage biasing circuit. By limiting the current in the ON state, this serial resistance also plays a role in maintaining the stability of the junctions. The observed switching scheme is ideal for memory applications, as the device can be switched between the two states at a reasonably high bias while  $R_{\text{ON}}$  and  $R_{\text{OFF}}$  are optimal for fast readout at low bias.

A similar  $I$ - $V$  trace is obtained at 4.2 K as shown in Fig. 1(b) testifying to the metallic nature of the system. The linear ON and OFF traces and the wide range of operation temperatures contradict the thermally activated ionic migration in bulk  $\text{Ag}_2\text{S}$ .<sup>29</sup> The observed metallic behavior is consistent with first-principles calculations applicable to a thin surface layer of  $\text{Ag}_2\text{S}$  deposited on bulk Ag which is expected to undergo a structural phase transition called spontaneous metallization.<sup>23</sup> Next, we investigated this metallized state experimentally by replacing the PtIr tip by superconducting Nb and utilizing point contact Andreev reflection (PCAR) spectroscopy.<sup>31–34</sup>

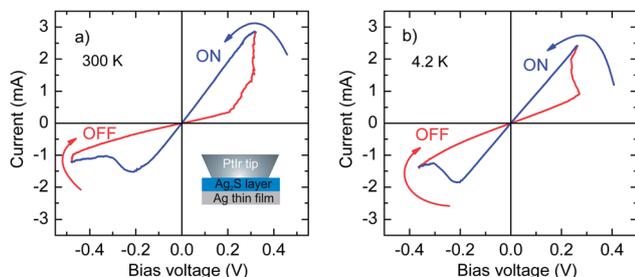


Fig. 1 Typical  $I$ - $V$  characteristics of  $\text{Ag}_2\text{S}$  based memristive junctions at room temperature (a) and at 4.2 K (b). Positive voltage corresponds to the positive biasing of the Ag layer with respect to the PtIr electrode. The arrows indicate the direction of the bias sweep. The inset shows the schematic cross section of the sample.

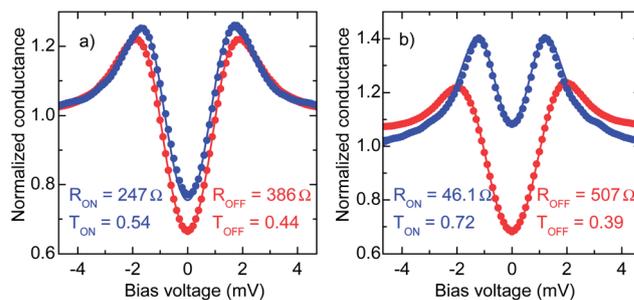


Fig. 2 ON and OFF state finite bias differential conductances obtained from the measured  $I$ - $V$  traces by numerical differentiation with respect to the bias voltage in two representative  $\text{Ag-Ag}_2\text{S-Nb}$  junctions at 4.2 K. Each curve is normalized to its high bias ( $eV \gg \Delta$ ) value  $G_{\text{ON,OFF}} = (R_{\text{ON,OFF}})^{-1}$ . Continuous lines are fits against the BTK theory resulting in the indicated transmissions.

This method is inspired by the fact that linear conductance measurements alone cannot distinguish between fundamentally different types of junctions exhibiting the same conductance. The conductance of a nanoscale device is given as  $G = 2e^2/hMT$ , where  $T$  is the average electron transmission probability across the device and  $M$  is the number of open conductance channels.<sup>36,37</sup> The latter is approximated as  $M \approx (\pi d/2\lambda_{\text{F}})^2$ , where  $d$  is the device diameter and  $\lambda_{\text{F}}$  is the Fermi wavelength which is  $\approx 0.4 \text{ nm}$  in bulk Ag. This simplified picture shows that a  $\text{k}\Omega$  device resistance may as well correspond to a large area tunnel junction ( $d \gg \lambda_{\text{F}}$ ) with very small transmission probability ( $T \ll 1$ ) or, alternatively, to a truly nanometer-scale junction with only a few well transmitting ( $T \approx 1$ ) conductance channels. PCAR measurements performed on the voltage scale of  $\Delta$  can distinguish between these extremities *via* the nonlinear transport properties of a point contact connecting a normal metal (N) and a superconductor (S).<sup>31</sup> The transmission can be evaluated by fitting the voltage dependence of differential conductance with the model of Blonder, Tinkham and Klapwijk (BTK).<sup>32,33</sup>

Fig. 2 shows the finite bias differential conductance of the ON and OFF states in two representative junctions exhibiting a minor change [Fig. 2(a)] and an order of magnitude larger change [Fig. 2(b)] in the normal state differential conductance ( $G_{\text{N}}$ ) upon switching. The typical switching threshold voltages of  $V_{\text{th}} \approx 300\text{--}800 \text{ mV}$  are well separated from the N-S features taking place at the voltage scale of  $\Delta$ . The effective transmissions were determined by fitting the differential conductance curves against the modified version of the BTK theory, incorporating a  $\Gamma$  broadening parameter, as defined in ref. 38, which is also necessary to account for contact diameters comparable to the electronic mean free path.<sup>39</sup> With this extension the spin degenerate model contains five parameters: the temperature,  $\Delta$ ,  $G_{\text{N}}$ ,  $\Gamma$  and  $T$ . A numerically reliable and physically meaningful fitting procedure, however, requires the reduction of an extended parameter space to a minimum set of key parameters. After taking extra care of the proper thermal anchoring of the device leads in our  $^4\text{He}$  cryostat we fixed the temperature value to 4.2 K. Finite bias tunneling measurements ( $R_{\text{contact}} > 100 \text{ k}\Omega$ ) carried out on  $\text{Ag-Ag}_2\text{S-Nb}$  samples and on clean Au foils revealed that the bulk value of  $\Delta = 1.4 \text{ meV}$  was



also a good approximation in our point contact arrangement. In these data,  $\Gamma$  stayed below 5% of  $\Delta$  quantifying the voltage noise of our setup. After evaluating  $G_N$  from the high bias linear slopes of the raw  $I$ - $V$  traces, the BTK fittings were run with two free parameters  $T$  and  $\Gamma$ .

In device 1 [Fig. 2(a)]  $R_{ON} = 247 \Omega$  and  $R_{OFF} = 386 \Omega$ . The corresponding transmission probabilities are 0.54 and 0.44, whereas the effective numbers of open conductance channels, estimated as  $M = G_N/G_0T$ , are 118 and 62, respectively. This shows that both the ON and OFF states are characterized by rather large transmission values and the 56% change of the conductance between the two states is equally attributed to the variations in  $M$  and  $T$ . In device 2 [Fig. 2(b)] the order of magnitude change in  $G_N$  ( $G_{ON}/G_{OFF} = 11$ ) is accompanied by a large change in  $M$  ( $M_{ON}/M_{OFF} = 6.2$ ) and a minor change in  $T$  ( $T_{ON}/T_{OFF} = 1.8$ ). The ON and OFF state effective contact diameters were estimated to be  $d_{ON} = 2.7$  nm and  $d_{OFF} = 2$  nm in device 1 and  $d_{ON} = 5$  nm and  $d_{OFF} = 2.1$  nm in device 2, demonstrating that resistive switching takes place in truly nanometer-scale junctions.

In order to verify the statistical relevance of these findings,  $T$  and  $d$  were evaluated for various junctions yielding to  $T_{ON} = 0.62 \pm 0.1$  and  $T_{OFF} = 0.42 \pm 0.07$  at effective junction diameters of  $d = 2$ –5 nm. Fig. 3 shows the relative changes in  $T$

and  $M$  upon switching as a function of the resistance ratio of the corresponding ON and OFF states. The two limiting cases of the unchanged transmissions and unchanged channel numbers are indicated in Fig. 3(a) and (b) by the orange and green dashed lines and are also schematically illustrated in Fig. 3(c) and (d), respectively. The numerical accuracy of  $T$  is 20% in the OFF states and better than 5% in the ON states as explained in the ESI† and is indicated by the error bars in Fig. 3(a) and (b). In the studied junctions with ON and OFF state resistances of 50–1000  $\Omega$ ,  $\Gamma$  is 50–15% of  $\Delta$ , respectively. These values are significantly higher than those obtained in the tunneling regime, which agree with previous studies carried out on various diffusive systems.<sup>39</sup> This indicates that in spite of the small junction diameters, electron transport is not entirely ballistic due to the rather short mean free path of 1.8 nm in  $Ag_2S$ .<sup>40</sup>

Our analysis shows that (i) regardless of the relative change in their resistances, all devices exhibit high transmission probabilities in both states corresponding to metallic conductances and (ii) their conductance changes can be attributed to a variation in  $M$  while  $T$  is less significantly changed during the switching. These results give clear evidence that the metallic channels are not completely destroyed when the device is switched off, only their effective cross section is reduced while the highly transmitting nature is preserved. This is in good agreement with the presence of spontaneously metallized  $Ag_2S$  channels<sup>23</sup> and is in contrast with the simplified picture of the formation and complete destruction of pure Ag nanofilaments.

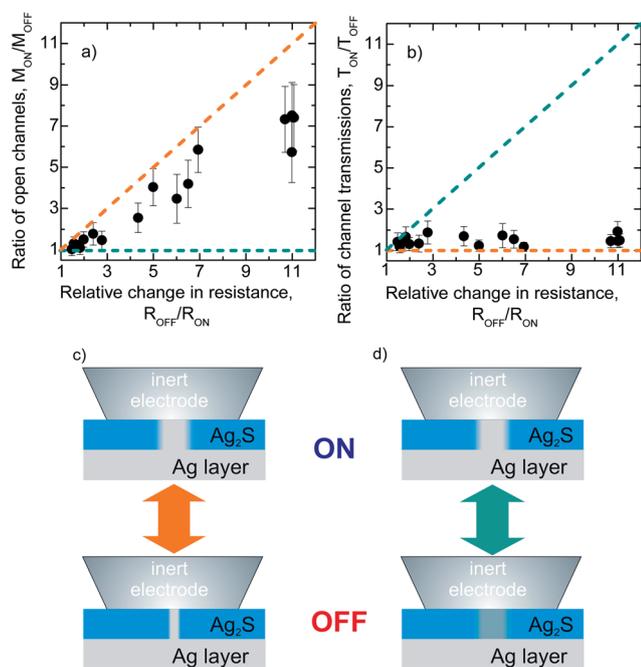


Fig. 3 Analysis of the finite bias differential conductance data obtained in various Ag– $Ag_2S$ –Nb devices. The ratios of the number of conducting channels (a) and the average transmission probabilities (b) in the ON and OFF states of each junction are shown as a function of the corresponding relative change in the normal resistance of the devices. The orange dashed line displays the limiting case where only  $M$  changes as illustrated in (c). The green dashed line indicates the opposite scenario, when the change in the resistance is solely attributed to the variation of  $T$  as illustrated in (d). In the lower panels the width and opacity of the grey region across the  $Ag_2S$  layer (blue) represent the number and transmission of the conducting channels, respectively.

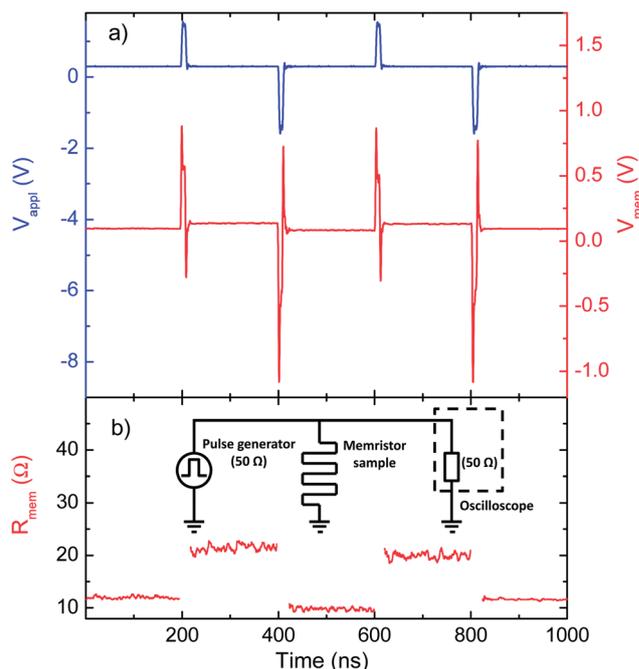


Fig. 4 (a) Alternating 10 ns wide voltage pulses ( $V_{app}$ , blue line) recorded in an open circuit configuration and measured on a Ag– $Ag_2S$ –PtIr junction ( $V_{mem}$ , red line) at room temperature. During the 200 ns time between the pulses a 350 mV dc read-out voltage was applied. (b) The calculated resistance demonstrates reproducible switching behavior. Data acquired within the pulse duration are not shown. The inset illustrates the equivalent circuit of the setup.





- 7 K. Terabe, T. Hasegawa, T. Nakayama and M. Aono, *Nature*, 2005, **433**, 47–50.
- 8 D. B. Strukov, G. S. Snider, D. R. Stewart and R. S. Williams, *Nature*, 2008, **453**, 80–83.
- 9 R. Waser and M. Aono, *Nat. Mater.*, 2007, **6**, 833–840.
- 10 K. Terabe, T. Hasegawa, C. Liang and M. Aono, *Adv. Mater.*, 2007, **8**, 536–542.
- 11 M. Aono and T. Hasegawa, *IEEE Proc.*, 2010, **98**, 2228–2236.
- 12 E. Linn, R. Rosezin, C. Kugeler and R. Waser, *Nat. Mater.*, 2010, **9**, 403–406.
- 13 A. C. Torrezan, J. P. Strachan, G. Medeiros-Ribeiro and R. S. Williams, *Nanotechnology*, 2011, **22**, 485203.
- 14 R. Waser, R. Dittmann, G. Staikov and K. Szot, *Adv. Mater.*, 2009, **21**, 2632–2663.
- 15 L. Chua, *IEEE Trans. Circuit Theory*, 1971, **18**, 507–519.
- 16 Y. V. Pershin and M. Di Ventra, *Adv. Phys.*, 2011, **60**, 145–227.
- 17 K. Terabe, T. Nakayama, T. Hasegawa and M. Aono, *J. Appl. Phys.*, 2002, **91**, 10110–10114.
- 18 M. Morales-Masis, S. J. van der Molen, W. T. Fu, M. B. Hesselberth and J. M. van Ruitenbeek, *Nanotechnology*, 2009, **20**, 095710.
- 19 M. Morales-Masis, H.-D. Wiemhöfer and J. M. van Ruitenbeek, *Nanoscale*, 2010, **2**, 2275–2280.
- 20 M. Morales-Masis, S. J. van der Molen, T. Hasegawa and J. M. van Ruitenbeek, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 115310.
- 21 A. Geresdi, A. Halbritter, A. Gyenis, P. Makk and G. Mihály, *Nanoscale*, 2011, **3**, 1504–1507.
- 22 Z. Xu, Y. Bando, W. Wang, X. Bai and D. Golberg, *ACS Nano*, 2010, **4**, 2515–2522.
- 23 Z. Wang, T. Kadohira, T. Tada and S. Watanabe, *Nano Lett.*, 2007, **7**, 2688–2692.
- 24 Z. Wang, T. Gu, T. Tada and S. Watanabe, *Appl. Phys. Lett.*, 2008, **93**, 152106.
- 25 J. J. T. Wagenaar, M. Morales-Masis and J. M. van Ruitenbeek, *J. Appl. Phys.*, 2012, **111**, 014302.
- 26 T. Hasegawa, T. Ohno, K. Terabe, T. Tsuruoka, T. Nakayama, J. K. Gimzewski and M. Aono, *Adv. Mater.*, 2010, **22**, 1831–1834.
- 27 T. Ohno, T. Hasegawa, T. Tsuruoka, K. Terabe, J. K. Gimzewski and M. Aono, *Nat. Mater.*, 2011, **10**, 591–595.
- 28 M. D. Pickett, G. Medeiros-Ribeiro and R. S. Williams, *Nat. Mater.*, 2013, **12**, 114–117.
- 29 A. Nayak, T. Tamura, T. Tsuruoka, K. Terabe, S. Hosaka, T. Hasegawa and M. Aono, *J. Phys. Chem. Lett.*, 2010, **1**, 604–608.
- 30 A. Geresdi, A. Halbritter, E. Szilágyi and G. Mihály, *MRS Proceedings*, 2011, 1331.
- 31 A. F. Andreev, *Soviet Physics - JETP*, 1964, **19**, 1228.
- 32 G. E. Blonder, M. Tinkham and T. M. Klapwijk, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1982, **25**, 4515–4532.
- 33 J. C. Cuevas, A. Martín-Rodero and A. Levy Yeyati, *Phys. Rev. B: Condens. Matter*, 1996, **54**, 7366–7379.
- 34 R. J. Soulen, J. M. Biers, M. S. Ofosky, B. Nadgorny, T. Ambrose, S. F. Cheng, P. R. Broussard, C. T. Tanaka, J. Nowak, J. S. Moodera, A. Barry and J. M. D. Coey, *Science*, 1998, **282**, 85–88.
- 35 E. Kótai, *Nucl. Instrum. Methods Phys. Res., Sect. B*, 1994, **85**, 588–596.
- 36 R. Landauer, *Philos. Mag.*, 1970, **21**, 863–867.
- 37 Y. V. Nazarov and Y. M. Blanter, *Quantum Transport: Introduction to Nanoscience*, Cambridge University Press, 2009.
- 38 A. Pleceník, M. Grajcar, Š. Beňačka, P. Seidel and A. Pfuch, *Phys. Rev. B: Condens. Matter*, 1994, **49**, 10016–10019.
- 39 A. Geresdi, A. Halbritter, M. Csontos, S. Csonka, G. Mihály, T. Wojtowicz, X. Liu, B. Jankó and J. K. Furdyna, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **77**, 233304.
- 40 J. Zemek, P. Jiricek, S. Hucek, A. Jablonski and B. Lesiak, *Surf. Sci.*, 2001, **473**, 8–16.
- 41 J. Williams, *Linear Technology Application Note*, 1991, vol. 47.

