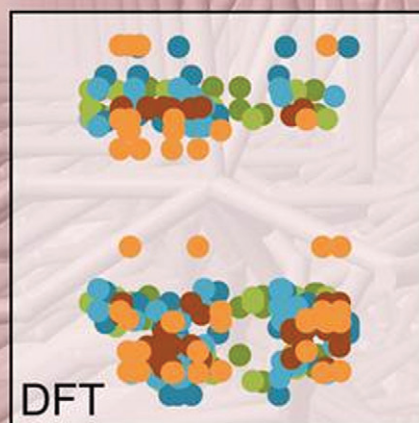
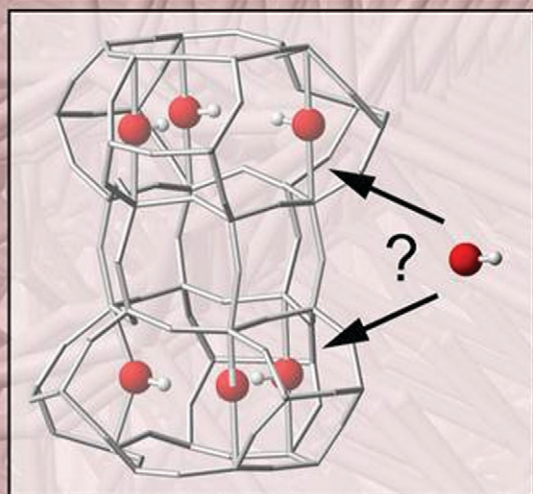
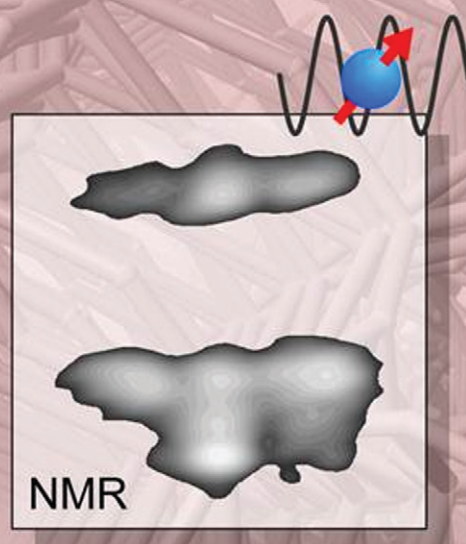
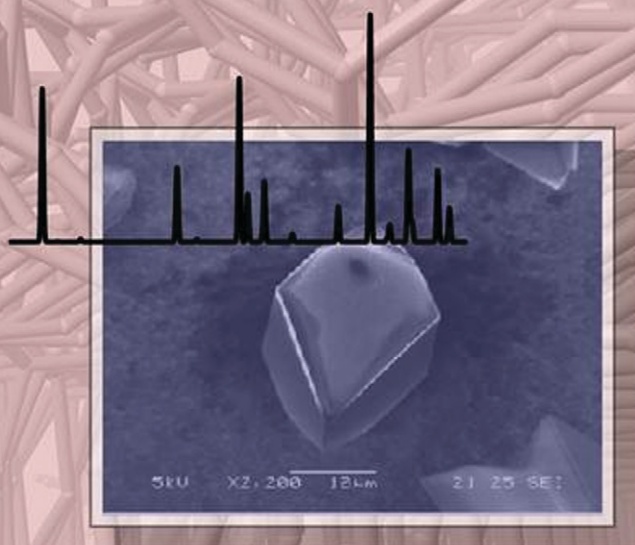


# CrystEngComm

www.rsc.org/crystengcomm

Volume 15 | Number 43 | 21 November 2013 | Pages 8589–8850



# Application of NMR crystallography to the determination of the mechanism of charge-balancing in organocation-templated AIPO STA-2†

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Cite this: *CrystEngComm*, 2013, 15, 8668

The combination of solid-state MAS NMR spectroscopy and first-principles calculations is used to elucidate the structure of an as-prepared microporous AIPO (STA-2), in which the organocation template (bis-diazabicyclooctane-butane) is charge balanced by hydroxyl groups coordinated to framework aluminium species.  $^{27}\text{Al}$  MAS NMR spectra show Al exists in both tetrahedral and five-fold coordination, with the latter directly coordinated to hydroxyl O atoms as well as framework O atoms. The relative intensities of these resonances can be used to determine that the hydroxyls bridge between the two types of Al. Calculation of NMR parameters using a periodic density functional theory approach are able to suggest assignments for the resonances in both  $^{27}\text{Al}$  and  $^{31}\text{P}$  NMR spectra.  $^{31}\text{P}$  chemical shifts are shown to depend not only on the topologically-distinct sites in the SAT framework but also on whether or not the P atoms form part of a six-membered  $\text{P}(\text{OAl})_2\text{OH}$  ring, where OH is a bridging hydroxyl. By diffraction six possible sites for the charge-balancing hydroxyls have been identified, but all are refined with an average occupancy of 0.33. However, predicted peak positions in two-dimensional J-HETCOR NMR spectra indicate that only one hydroxyl is found in each cancrinite cage, and suggest that the most likely arrangements are those that avoid the close approach of bridging hydroxyl groups in adjacent cages. We show that the use of a combination of NMR spectroscopy and calculation to elucidate structure, often referred to as “NMR Crystallography”, offers a promising route for the future study of as-made and substituted microporous materials.

Received 30th May 2013,  
Accepted 9th July 2013

DOI: 10.1039/c3ce40965a

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## Introduction

The wide range of porous structures exhibited by zeotypic materials has resulted in a variety of potential applications, including gas storage and separation, drug delivery and in catalysis. Aluminophosphates (AIPOs), first reported in 1982,<sup>1</sup> are the most extensive group of zeotypes, forming many structures isomorphous with those of the common silica zeolites, but others with unique topologies. In order to avoid the formation of dense phases, synthesis of AIPOs typically requires an organic amine base (usually referred to as a structure-directing agent (SDA) or template), enabling open frameworks to be produced. As the SDA is usually positively

charged, a charge-balancing mechanism is required in the “as-made” materials to maintain the inherently neutral tetrahedral AIPO framework. This can be achieved by the partial substitution of divalent metal cations for Al, or of Si for P, but in many cases the coordination of anions from the synthesis solution, typically  $\text{OH}^-$  or  $\text{F}^-$ , to the aluminium cations of the framework is observed, creating five- and/or six-coordinate Al species. Where hydroxyls are present, different coordination modes can be observed; the hydroxyls may be terminal (*i.e.*, connected to a single Al species), as in STA-15,<sup>2</sup> or they may be bridging (*i.e.*, connected to two Al species simultaneously), as in AIPO-17<sup>3</sup> and AIPO-18.<sup>4</sup> In most cases the SDA, charge-balancing anions and any water within the pores of the AIPO can be removed by calcination to produce a neutral (and purely tetrahedral) framework.

NMR spectroscopy is an ideal tool for the investigation of phosphate frameworks, owing to its sensitivity to the local structural environment, and its ability to provide information in the absence of long-range periodicity. For aluminophosphates, NMR can be used to study the basic components of the framework, as both  $^{27}\text{Al}$  ( $I = 5/2$ ) and  $^{31}\text{P}$  ( $I = 1/2$ ) have 100% natural abundance. More recently, using ionothermal synthesis,

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† Electronic supplementary information (ESI) available: Further information on the synthesis of AIPO STA-2, referencing of the DFT calculations,  $^{27}\text{Al}$  MQMAS spectra of STA-2(BDAB), D-HMQC spectra of STA-2(BDAB), NMR spectra of dehydrated STA-2(BDAB), calculated NMR parameters for structural models of STA-2(BDAB), comparison of DFT calculations and MQMAS spectra for STA-2(BDAB), DFT calculations of J-HETCOR spectra and calculated  $^{13}\text{C}/^{15}\text{N}$  NMR parameters of the template in STA-2(BDAB). The crystallographic information file is also provided. See DOI: 10.1039/c3ce40965a





cost-efficient low-level (*i.e.*, 4–10%)  $^{17}\text{O}$  enrichment of AlPO frameworks has been demonstrated, overcoming the low natural abundance ( $\sim 0.037\%$ ) of this isotope, and offering an interesting new approach to studying host–guest interactions.<sup>5</sup> For as-made AlPOs, further structural information can be obtained as both the charge-balancing anions and SDA molecules contain a range of NMR-active nuclides (*e.g.*,  $^{13}\text{C}$ ,  $^{14/15}\text{N}$ ,  $^1\text{H}$ ,  $^{19}\text{F}$ ). Although the AlPO framework exhibits a periodic, repeating structure, the substitution of other elements into this framework, and the disordered arrangements of many charge-balancing anions and SDAs can pose a challenge for diffraction-based methods, and the combination of multi-dimensional high-resolution NMR alongside more conventional crystallographic methods can provide a more complete and more detailed picture.<sup>2,3,6–14</sup> For example, multinuclear NMR spectroscopy was recently applied to provide information on the distribution of  $\text{OH}^-/\text{F}^-$  anions in AlPO-CJ2,<sup>9</sup> the non-periodic F subnetwork in AlPO cloverite,<sup>10</sup> and the *in situ* reversible transformation of AlPO-53 to JDF-2.<sup>7</sup> NMR spectroscopy is also sensitive to dynamics over a wide range of timescales, and has been applied to investigate the  $\mu\text{s}$ -timescale motion of different SDAs (isopropylamine and piperidine) in AlPO-14.<sup>13</sup>

In recent years, experimental NMR investigation has been complemented by an increasing use of first-principles calculations, facilitating spectral interpretation and assignment in simple systems, but providing insight into local structure, disorder and dynamics in more complex materials. The recent introduction of the GIPAW approach, which exploits the inherent periodicity of the solid state,<sup>15</sup> has led to a resurgence of interest in the calculation of NMR parameters in solids, and applications in areas as wide ranging as biomaterials, minerals, energy materials and ceramics.<sup>16–19</sup> Although applied relatively rarely to AlPO frameworks,<sup>2,7,11,12</sup> this approach has been used to assign spectra and to understand phase transformations and dehydration. Notably, NMR parameters calculated from published crystal structures can be in poor agreement with experiment in some cases, unless optimization of the geometry is performed.<sup>12,18</sup> This suggests that a combination of NMR and first-principles calculations, often termed “NMR crystallography”, will be a useful tool in the structural refinement of microporous materials in the future, particularly for structures that have been derived from powder diffraction experiments.

Here we use an NMR crystallography approach to investigate the as-made microporous aluminophosphate framework STA-2. STA-2 was first reported in 1997, using 1,4-bis-*N*-quinuclidiniumbutane (BQNB) as the SDA, in the magnesium aluminophosphate (MgAPO) form (framework composition  $\text{Mg}_{0.15}\text{Al}_{0.85}\text{PO}_4$ ).<sup>20</sup> In subsequent work, a pure AlPO form was prepared using either BQNB or bis-diazabicyclooctane (BDAB) as an SDA, with the latter a potentially cheaper alternative to the former.<sup>21</sup> In both cases, charge balance was achieved through the coordination of  $\text{OH}^-$  species to the framework, and although refinement of synchrotron X-ray powder diffraction suggested locations for some of these hydroxyl groups they were neither fully nor unambiguously located in

the structure. Although NMR spectra of the as-made forms were not assigned in this work, the similarity in the spectra observed for the two SDAs led to the conclusion that (i) BDAB was also divalent when incorporated into the material (this SDA has the potential for its charge to vary from 2+ to 4+) and (ii) that the structures of the two as-made forms were very similar. In this work, we combine multinuclear NMR spectroscopy and first-principles calculations to investigate the distribution of hydroxyls in as-made STA-2(BDAB). A variety of one- and two-dimensional NMR experiments are used, alongside GIPAW calculations performed for a range of simple structural models with a variety of hydroxyl positions, in an attempt to assign and interpret the NMR spectra obtained and to gain insight into the atomic-scale structure of STA-2.

## Experimental details

### Synthesis

AlPO STA-2 materials were prepared according to the method of Castro *et al.*<sup>21</sup> The SDA (BDAB) was prepared in the bromide form by the Menschutkin reaction of diazabicyclooctane with dibromobutane. Before use in the AlPO synthesis, the bromide salt of BDAB was converted to a solution of the hydroxide form either by reaction with excess  $\text{Ag}_2\text{O}$  or by stirring with a large excess of the Amberlite® IRN78 resin (fully  $\text{OH}^-$  exchanged) for 12 h at room temperature. AlPO STA-2 was crystallised by hydrothermal treatment of a gel of composition  $\text{R}(\text{OH})_2:\text{Al}(\text{OH})_3:\text{H}_3\text{PO}_4:\text{H}_2\text{O} = 0.44:1:1:40$  in a Teflon-lined stainless steel autoclave at 190 °C for 48 h. After slow cooling, the product was suspended in water, sonicated to remove uncrystallised material in suspension and the crystalline solid was filtered off (see ESI† for further information). The NMR spectra described below were measured on AlPO samples prepared *via*  $\text{Ag}_2\text{O}$  treatment whereas crystals suitable for single-crystal X-ray diffraction studies (SXRD) were prepared *via* anion exchange over the resin.

The samples were confirmed as STA-2 *via* laboratory PXRD on a Stoe STAD i/p powder diffractometer using  $\text{Cu K}\alpha_1$  radiation. The particle morphology was determined by scanning electron microscopy using a JEOL JSM-5600 at 5 kV acceleration voltage and a working distance of 21 mm. The spot size was set to 25 nm.

### Diffraction

Single crystals of as-prepared AlPO STA-2 of suitable size (10–20  $\mu\text{m}$ , see ESI†) were examined on a Rigaku Saturn92 diffractometer with a CCD detector using  $\text{Cu K}\alpha_1$  radiation (rotating anode). The structure was solved and refined employing SHELX-97,<sup>22</sup> (see Table 1 and crystallographic file in ESI†). The framework was solved with alternating Al and P in tetrahedral cation positions and remaining extra-framework electron density in the large cage was assigned to the template. Electron density found in the smaller cancrinite cage was identified as charge-balancing hydroxide groups in the cages. The occupancy of the hydroxide O atoms (represented as O100 in the refinement of STA-2 in  $R\bar{3}$ ) was allowed to vary, giving an



**Table 1** Crystallographic details for as-prepared AIPO STA-2

Chemical formula unit	(C <sub>8</sub> H <sub>12</sub> N <sub>2</sub> ) <sub>2</sub> Al <sub>12</sub> P <sub>12</sub> O <sub>50</sub>
Z	3
Unit cell weight	5301.69
Calculated density/g cm <sup>-3</sup>	2.029
Temperature/K	173
Space group	R $\bar{3}$
X-ray source	Rotating anode Cu K $\alpha_1$
Wavelength/Å	1.54187
Crystal size/mm	0.01 × 0.01 × 0.02
Unit cell dimensions/Å	<i>a</i> = 13.0310(6) <i>b</i> = 13.0310(6) <i>c</i> = 29.4980(16) $\alpha = 90^\circ$ $\beta = 90^\circ$ $\gamma = 120^\circ$
Unit cell volume/Å <sup>3</sup>	4337.89(4)
<i>R</i> <sub>p</sub> (all data)	0.1121
<i>R</i> <sub>p</sub> ( <i>I</i> > 2 $\sigma$ <i>I</i> )	0.1689
Max and min residual e <sup>-</sup> density/e <sup>-</sup> Å <sup>-3</sup>	-0.77, 1.21

average occupancy of 0.33. Lowering the occupancy gave rise to physically unrealistic thermal parameters for this species.

### NMR spectroscopy

MAS NMR spectra were acquired using Bruker Avance spectrometers at *B*<sub>0</sub> field strengths of either 9.4, 14.1, 18.8 or 20.0 T, at Larmor frequencies of 104.3, 156.4, 208.4 and 221.5 MHz for <sup>27</sup>Al, and 162.0, 242.9, 323.9 and 344.1 MHz for <sup>31</sup>P, at the University of St Andrews (9.4 and 14.1 T), L'Universite de Lille, France (18.8 T) and the UK 850 MHz Solid-State NMR Facility (20.0 T). Powdered samples were packed into conventional 4 or 3.2 mm rotors and rotated at MAS rates between 12.5 and 20 kHz. Typical recycle intervals were 1–3 s for <sup>27</sup>Al and 10–30 s for <sup>31</sup>P. Chemical shifts are reported (in ppm) relative to 85% H<sub>3</sub>PO<sub>4</sub> for <sup>31</sup>P and 1 M Al(NO<sub>3</sub>)<sub>3</sub> (aq) for <sup>27</sup>Al. Triple-quantum MAS NMR experiments were carried out using a phase-modulated split-*t*<sub>1</sub> shifted-echo pulse sequence,<sup>23</sup> or a z-filtered pulse sequence.<sup>24</sup> In each case, the final pulse was chosen to be selective for the central transition. The scale in the indirect dimension is referenced according to the convention in ref. 25. Two-dimensional heteronuclear correlation experiments were performed using transfer through either the *J* coupling (with an INEPT transfer from <sup>27</sup>Al to <sup>31</sup>P)<sup>26,27</sup> or the dipolar coupling (using a D-HMQC<sup>28</sup> experiment). In each case, for <sup>27</sup>Al the pulses applied were selective for the central transition.

### DFT calculations

Calculations of total energies and NMR parameters were carried out using the CASTEP density functional theory (DFT) code (version 5),<sup>29</sup> employing the GIPAW algorithm,<sup>15</sup> to reconstruct the all-electron wave function in the presence of a magnetic field. Calculations were performed using the GGA PBE functional,<sup>30</sup> with core–valence interactions described by ultrasoft pseudopotentials.<sup>31</sup> A planewave energy cutoff of 50 Ry was used, and integrals over the Brillouin zone were performed using a Monkhorst–Pack grid with *k*-point spacing of 0.04 Å<sup>-1</sup>. Calculations were performed on a 198-node (2376 core) Intel Westmere cluster with 2 GB memory per core and QDR Infiniband interconnect at the University of St Andrews.

The initial STA-2 structure was taken from diffraction measurements, with protons added manually to the SDA. Hydroxyl groups were also added manually as discussed in detail in later sections. In a preliminary step the positions of all protons in the structure were optimised (with the coordinates of all other atoms and the unit cell size and shape fixed). Subsequently, geometry optimisation was performed (using a planewave energy cutoff of 50 Ry and a *k*-point spacing of 0.04 Å<sup>-1</sup>) with the positions of all atoms, and the unit cell size and shape, allowed to vary. After 30 iterations of optimisation (5–6 days calculation time) the forces on the atoms had reduced to less than 0.1 eV Å<sup>-1</sup>, although the optimization was not converged (within the criteria specified: 1 × 10<sup>-4</sup> eV per atom, 0.05 eV Å<sup>-1</sup> and 1 × 10<sup>-3</sup> Å for total energy, ionic force and ionic displacement, respectively), at this point. This could be seen to be due to motion of the template within the pores, rather than any change in the position of the framework atoms. Geometry optimization for one structural model was continued until completion (110 steps, 24 days). However, no significant change was observed in the position of the framework atoms. Owing to the lengthy calculation times, for all other models structures optimised for 30 iterations were deemed sufficient.

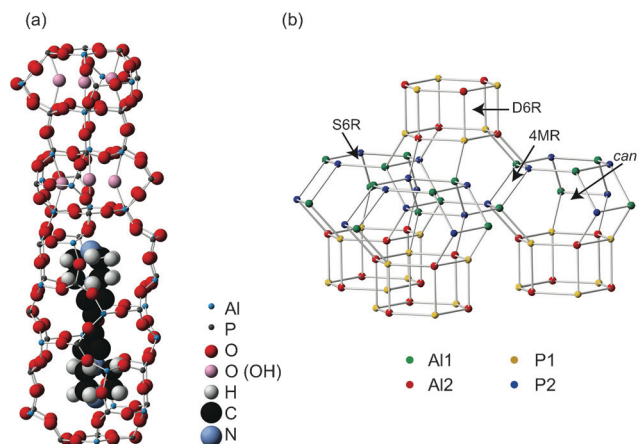
Calculations generate the absolute shielding tensor ( $\sigma$ ) and the electric field gradient (EFG) tensor (*V*) in the crystal frame. In each case, diagonalisation yields the three principal components of the tensor ( $\sigma_{xx}$ ,  $\sigma_{yy}$  and  $\sigma_{zz}$  for the shielding tensor and *V*<sub>xx</sub>, *V*<sub>yy</sub> and *V*<sub>zz</sub> for the EFG tensor). The isotropic shielding,  $\sigma_{iso} = (1/3) \text{Tr}\{\sigma\}$ , and the isotropic chemical shift,  $\delta_{iso}$ , is given by  $-(\sigma_{iso} - \sigma_{ref})$ , where  $\sigma_{ref}$  is a reference shielding. This was determined to be 552.3 ppm for <sup>27</sup>Al and 278.9 ppm for <sup>31</sup>P, from calculations carried out on calcined STA-2 (see ESI† for further details). The magnitude of the quadrupolar coupling constant is given by  $C_Q = eQV_{zz}/h$ , where *Q* is the nuclear quadrupole moment (for which a value of 146.6 mB was used for <sup>27</sup>Al).<sup>32</sup> The asymmetry parameter is given by  $\eta_Q = (V_{xx} - V_{yy})/V_{zz}$ .

## Results and discussion

### NMR spectroscopy

The STA-2 structure (framework type SAT), shown in Fig. 1a, can be described in terms of the stacking of its secondary building units (SBUs), with six-membered rings (6MRs) stacked at different positions in the *xy* plane of a hexagonal unit cell, connected *via* four-membered rings (4MRs).<sup>20,21</sup> As shown in Fig. 1b, the resulting framework contains single and double six-membered rings (S6Rs and D6Rs, respectively), 4MRs and both cancrinite (can) and elongated cages. In the calcined framework there are two distinct Al and two distinct P sites (shown also in Fig. 1b). Al2 and P1 are located in D6R units, and Al1 and P2 are located in S6R units. These join to form cancrinite cages that share a face of a 4MR. Therefore, *via* oxygen, Al1 is bonded to one P1 and three P2, and Al2 is bonded to one P2 and three P1. The <sup>27</sup>Al and <sup>31</sup>P NMR spectra for calcined (dehydrated) STA-2(BDAB) have been shown in previous work,<sup>21</sup> and are also shown in

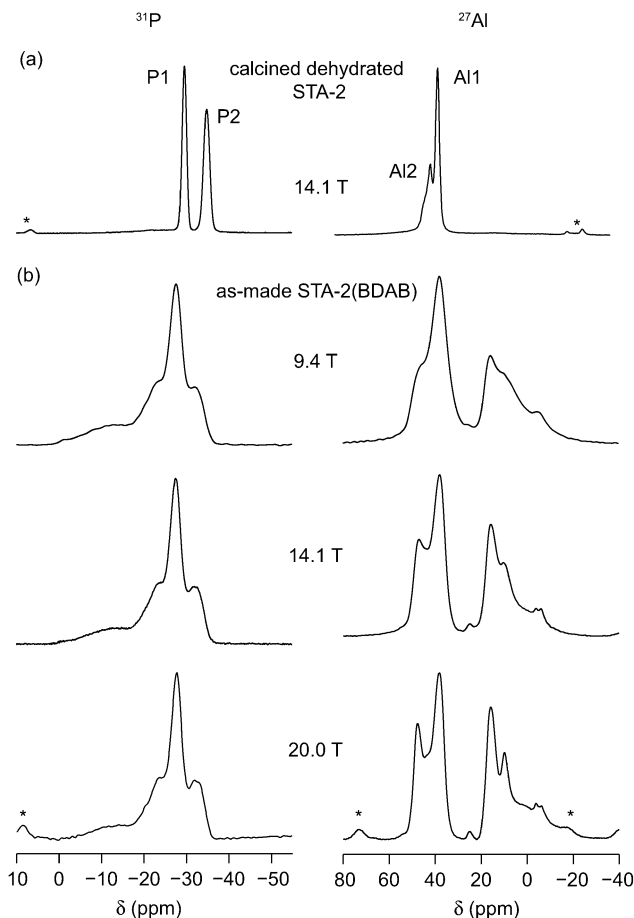




**Fig. 1** (a) Part of the structure of STA-2(BDAB), showing the BDAB template within the pores. The six possible hydroxyl positions suggested by diffraction measurements are also shown (as pink spheres). (b) Layers of cancrinite cages in the STA-2 framework structure, with Al1, Al2, P1 and P2 shown by green, red, yellow and blue spheres, respectively. Single and double six-membered rings (S6Rs and D6Rs, respectively), four membered rings (4MRs) and cancrinite (can) cages are shown.

**Fig. 2a.** The  $^{31}\text{P}$  MAS NMR spectrum shows two distinct resonances at  $-29.6$  and  $-34.8$  ppm, corresponding to P1 (D6R) and P2 (S6R), respectively. The  $^{27}\text{Al}$  spectrum also showed two resonances, albeit overlapping, which could be resolved by MQMAS, with  $\delta_{\text{iso}}$  of  $36.0$  and  $42.0$  ppm, corresponding to Al1 (S6R) and Al2 (D6R), respectively. NMR parameters for calcined STA-2 are summarised in Table 2. The spectral assignment was supported by DFT calculations and two-dimensional heteronuclear correlation experiments.<sup>21</sup>

The as-made AIPO contains the SDA (BDAB), encapsulated within the pores, and charge-balancing hydroxyls coordinated to Al atoms in the framework. The local environment of each Al and P species is now modified, with the presence of both four- and five-coordinated Al sites, while each P species now has a potentially different next-nearest neighbour (NNN) coordination,  $\text{P}(\text{OAl}^{\text{IV}})_{4-n}(\text{OAl}^{\text{V}})_n$ , where the superscript indicates the Al coordination number. Fig. 2b shows  $^{31}\text{P}$  and  $^{27}\text{Al}$  MAS NMR spectra of STA-2(BDAB) at variable external magnetic field strength. (MAS NMR spectra at  $B_0 = 14.1$  T were previously reported by Castro *et al.*,<sup>21</sup> but were unassigned.) For both  $^{27}\text{Al}$  and  $^{31}\text{P}$ , the spectra are more complex than those for the calcined form, with a number of broad and overlapping lineshapes observed. For  $^{31}\text{P}$ , intense resonances are observed between  $-20$  and  $-40$  ppm, with a much broader, low intensity component at higher chemical shift. No difference in the spectrum is observed when using cross polarization, indicating all peaks result from  $\text{Q}^4$  P species and not from species with any directly-bonded hydroxyls. Two distinct sets of resonances are observed in the  $^{27}\text{Al}$  MAS NMR spectrum, corresponding to four-coordinate Al between  $35$  and  $50$  ppm and five-coordinate Al between  $0$  and  $20$  ppm.<sup>33</sup> A broader resonance with low intensity is also observed at lower chemical shift, perhaps resulting from an amorphous or less crystalline impurity phase. As discussed previously,<sup>21</sup> the relative ratio of the signal intensity for  $\text{Al}^{\text{IV}}$  and  $\text{Al}^{\text{V}}$  can be used to provide insight into



**Fig. 2**  $^{31}\text{P}$  (left) and  $^{27}\text{Al}$  (right) MAS NMR spectra of (a) calcined dehydrated STA-2 and (b) as-made STA-2(BDAB) at varying  $B_0$  field strengths. The MAS rate was between  $12.5$  and  $14$  kHz. Spinning sidebands are denoted by \*.

the position of the charge-balancing hydroxyls. As demonstrated previously by comparison to STA-2(BQNB), the charge on the SDA is  $2+$ , and so two hydroxyls are required to charge balance each template molecule.<sup>21</sup> The basic formula therefore becomes  $\text{Al}_{12}\text{P}_{12}\text{O}_{48}(\text{OH})_2\text{BDAB}$ . If the hydroxyls are terminal ligands (*i.e.*, coordinated only to one Al species) then two Al would be five coordinate and the remainder four coordinate, resulting in a  $2:10$  (or  $1:5$ )  $\text{Al}^{\text{V}}:\text{Al}^{\text{IV}}$  ratio. If, however, the hydroxyls bridge between Al species there would be four five-coordinate and eight four-coordinate Al species, and a  $4:8$  (or  $1:2$ )  $\text{Al}^{\text{V}}:\text{Al}^{\text{IV}}$  ratio. The latter case is in much better agreement with ratio observed experimentally (between

**Table 2** NMR parameters (isotropic chemical shift,  $\delta_{\text{iso}}$ , quadrupolar coupling,  $C_Q$ , asymmetry,  $\eta_Q$ , and quadrupolar product,  $P_Q = C_Q(1 + \eta_Q^2/3)^{1/2}$ ) for calcined AIPO STA-2<sup>21</sup>

Species	SBU	$\delta_{\text{iso}}$ (ppm)	$C_Q/\text{MHz}$	$\eta_Q$	$P_Q/\text{MHz}$
Al1	S6R	36.0(5)	2.0(2)	0.7(2)	2.1(2)
Al2	D6R	42.0(5)	3.5(1)	0.9(5)	3.9(2)
P1	D6R	$-29.6$			
P2	S6R	$-34.8$			





**Table 3** NMR parameters (average isotropic chemical shift,  $\langle\delta_{\text{iso}}\rangle$ , FWHM of the distribution of isotropic chemical shift<sup>a</sup>, and peak value of the quadrupolar product,  $\langle P_Q\rangle^b$ ) for as-made STA-2(BDAB)<sup>c</sup>

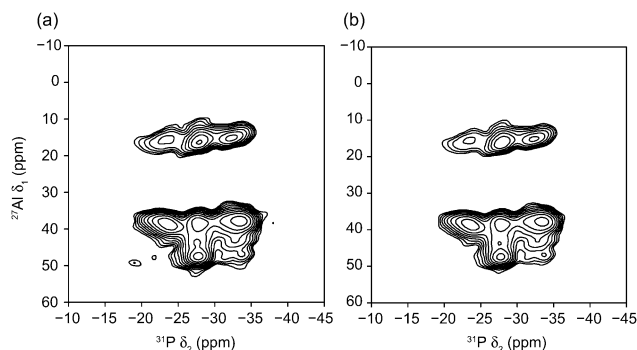
Species	$\langle\delta_{\text{iso}}\rangle$ (ppm)	FWHM (ppm) <sup>a</sup>	$P_Q/\text{MHz}^b$
Al(v)	17(1)	4	3.1(2)
Al(iv)	39(1)	4	2.4(2)
Al(iv)	45(1)	3	3.6(2)
Al(iv)	49(1)	3	3.1(2)

<sup>a</sup> Assuming a Gaussian distribution. <sup>b</sup> Extracted using a Cjzek (GIM,  $d = 5$ ) distribution of the quadrupolar product.<sup>34</sup> <sup>c</sup> Note that most resonances most likely result from the overlap of a number of signals.

1:1.5 and 1:1.7 at the three fields), suggesting the hydroxyls in STA-2(BDAB) are bridging.

<sup>27</sup>Al (14.1 T) multiple-quantum (MQ) MAS NMR spectra (shown in ESI†) exhibit broadened resonances, despite the resolution enhancement expected from the removal of the second-order quadrupolar broadening, as a result of a distribution in both  $\delta_{\text{iso}}$  and  $P_Q$ . Spectra have been recorded using both phase-modulated shifted-echo<sup>23</sup> and  $z$ -filtered<sup>24</sup> experiments to ensure that any broadened resonances are acquired accurately and that no signal is lost as a result of  $T_2$  relaxation during the echo period. Two broadened resonances corresponding to four-coordinate Al are clearly resolved, and are presumed to arise from species that correspond to Al1 and Al2 in the calcined form (*i.e.*, in S6Rs and D6Rs respectively), although these cannot be assigned simply from the spectrum. A small but distinct shoulder is observed between the two. Only one (broadened) resonance is observed in the five-coordinate region of the spectrum. NMR parameters (extracted using dmfit<sup>34</sup> assuming a Cjzek distribution of quadrupolar products from spectra acquired at a range of field strengths) are given in Table 3. (Note that most resonances most likely result from the overlap of a number of signals.) The integrated intensities of the two major Al<sup>IV</sup> resonances is 1:1, leading to the suggestion that a similar number of bridging hydroxyls are attached to both Al1 and Al2.<sup>21</sup> (Although MQMAS is formally a non-quantitative experiment, both species have a similar distribution of  $P_Q$ , meaning that accurate relative intensities should be obtained.) The presence of one major resonance corresponding to Al<sup>V</sup> is, therefore, a little surprising as one might have expected two resonances for Al<sup>V</sup> in S6Rs and D6Rs (*i.e.*, formally Al1 and Al2 species), casting some doubt on the previous spectral interpretation.

Fig. 3 shows two-dimensional <sup>27</sup>Al–<sup>31</sup>P heteronuclear correlation (J-HETCOR) spectra<sup>26,27</sup> of STA-2(BDAB), acquired at  $B_0 = 20.0$  T. Several cross peaks are observed, with varying intensity, although notably the <sup>31</sup>P resonance at –23 ppm is not connected to the Al<sup>IV</sup> species at 49 ppm. The shoulder at  $\delta_1 \approx 45$  ppm (observed in the MQMAS spectra) is clearly visible, and exhibits a connectivity similar to that of the Al species at  $\delta_1 \approx 49$  ppm, suggesting it has a more similar environment to this Al species than that with  $\delta_1 \approx 39$  ppm. No significant difference was observed in the spectra when the evolution interval,  $\tau$ , was increased, suggesting all Al–O–P  $J$  couplings are of similar



**Fig. 3** <sup>27</sup>Al–<sup>31</sup>P (20.0 T) two-dimensional J-HETCOR spectra of STA-2(BDAB), acquired using an INEPT transfer from <sup>27</sup>Al to <sup>31</sup>P, with  $\tau$  of (a) 1.6 ms and (b) 2.4 ms. The MAS rate was 12.5 kHz.

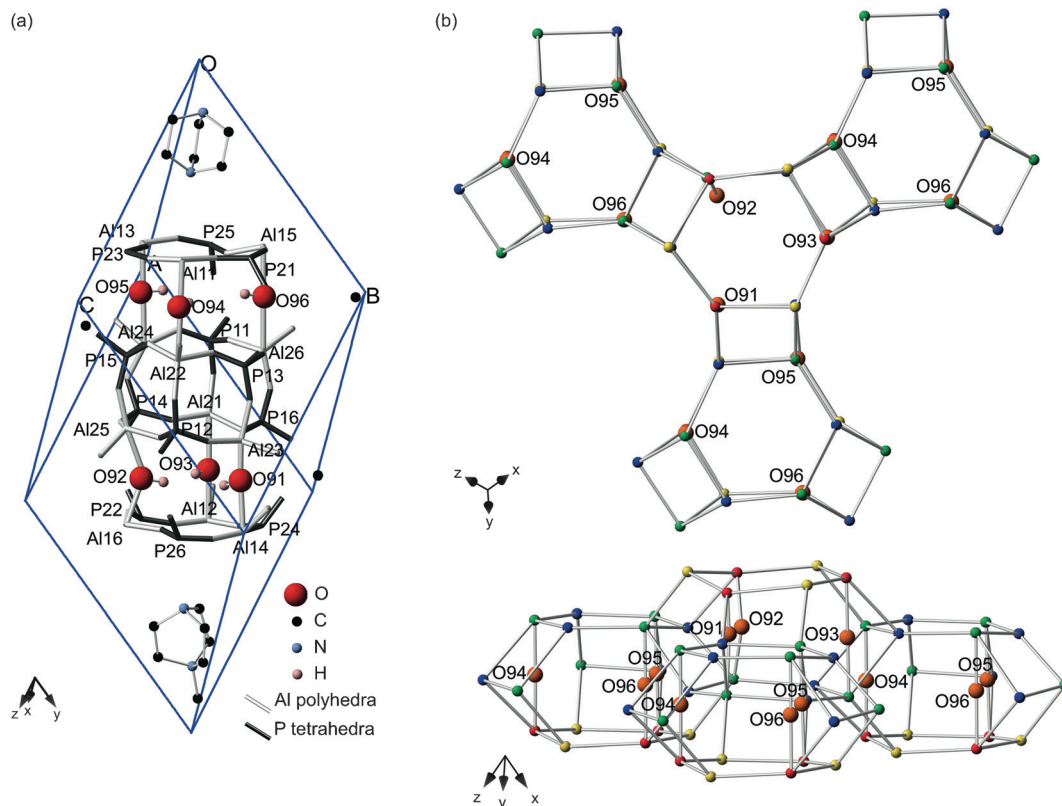
magnitude. MQ-J-HETCOR experiments<sup>27</sup> were also performed, but owing to the disordered nature of the sample there was little increase in resolution, and a significant loss of sensitivity. A D-HMQC spectrum<sup>28</sup> (exploiting transfer *via* the dipolar coupling) was also acquired (at  $B_0 = 18.8$  T), and is shown in the ESI†. In general, similar cross peaks are observed (note the reversal of the nuclear dimensions), with increased intensity of the cross peak between the resonances (now) at –22 ppm (<sup>31</sup>P) and 47 ppm (Al<sup>IV</sup>), which is absent in the experiment probing through-bond connectivity, confirming these species are close in space but not linked through covalent bonds. However, a broad, low intensity cross peak is observed between the broad resonances observed in both <sup>27</sup>Al and <sup>31</sup>P spectra, confirming these are present in the same chemical phase. Notably, no correlations are observed for either of these resonances with any of the sharper, more intense peaks, confirming the suggestion that they result from a less crystalline impurity phase.

STA-2 is known to absorb water in its as-made form.<sup>20,21</sup> However, there were no significant changes in the <sup>27</sup>Al and <sup>31</sup>P spectra of a sample of STA-2(BDAB) dried at 250 °C for 3 hours (shown in ESI†), suggesting that the water is present only within the pores and is not coordinated to the framework Al species. A comparison of the <sup>1</sup>H MAS NMR spectra (also shown in ESI†) reveals a decrease in intensity of the resonance at ~5 ppm upon drying, confirming water was present in the pores initially. (The residual intensity at 5 ppm in the spectrum of the dried material probably results from the hydroxyl protons.) A two-dimensional <sup>27</sup>Al/<sup>31</sup>P J-HETCOR spectrum (see ESI†) of the dried material, acquired at 14.1 T, shows no differences to that of the hydrated as-made material, confirming that the framework structure (and, therefore, the NMR parameters) are unaffected by the presence of water.

## Diffraction

Although there are two crystallographically-distinct Al and P sites in calcined STA-2,<sup>20</sup> the presence of the SDA and the associated hydroxyls must lower the local symmetry from rhombohedral to triclinic. This makes each of the twelve Al and P species in the primitive P1 triclinic cell formally different (six for each of the two topologically-distinct sites).





**Fig. 4** (a) Unit cell corresponding to dehydrated STA-2(BDAB), showing the distinct Al and P species, and the six possible hydroxyl locations. (b) Part of framework structure of dehydrated STA-2(BDAB), shown as a wireframe model, indicating the proximity of the hydroxyl groups. Atoms in the BDAB template are not shown for clarity.

The six Al located in S6Rs (*i.e.*, formally Al1 in the calcined material) are denoted in this work Al11–Al16, whereas those in D6Rs (*i.e.*, formally Al2 in the calcined material) are denoted Al21–Al26. Similarly P11–P16 and P21–P26 refer to P derived from P1 and P2, respectively, in the calcined material. In previous work synchrotron powder X-ray diffraction data for dehydrated STA-2(BDAB) was refined starting from an initial structural model with the position of the template determined by molecular modelling.<sup>21</sup> Difference Fourier maps showed scattering within the cancrinite cages, thought to be due to the hydroxyls. Using information from the <sup>27</sup>Al MAS NMR spectra (which demonstrate the hydroxyls are bridging between two different Al species), six potential positions for hydroxyls within the cancrinite cages (*i.e.*, three within each cage in the unit cell) could be refined, and are shown in Fig. 4a, numbered O91 to O96. More detailed refinement determined one of the two hydroxyls within the structure was definitely located within one of the cancrinite cages (on O91 or O92, with fractional occupancies of 0.25 and 0.75, respectively). The second hydroxyl could not be located unambiguously, but probably also lay within a cancrinite cage.<sup>21</sup>

During the course of the current work it was possible to prepare single crystals suitable for structural analysis in the as-prepared form by SXRD (structural details given in Table 1). The data was indexed in the rhombohedral space group  $R\bar{3}$  and structure solution confirmed the STA framework (average Al–O framework bond lengths (disregarding

coordinated  $\text{OH}^-$ ), Al1–O = 1.72(2) Å; Al2–O = 1.74(1) Å; average P–O bond lengths, P1–O = 1.51(1) Å; P2–O = 1.51(1) Å), and located the intact BDAB template. Furthermore, and consistent with <sup>27</sup>Al MAS NMR, it was possible to locate O atoms in bridging positions between Al1 and Al2 cations in the top and bottom 6MRs of the cancrinite cages, with an average occupancy of 0.33, corresponding to one bridging  $\text{OH}^-$  per cage, or two per BDAB SDA, as expected for complete charge balance of the template. The Al–O distances (Al1–O100 = 2.28(2) Å; Al2–O100 = 2.35(2) Å) are longer than expected, but note that this can be attributed to the fractional occupancy of the hydroxyl group, such that the Al positions are weighted averages of coordinated and uncoordinated environments. It was not possible to determine the positions of the  $\text{OH}^-$  groups relative to one another owing to the statistical disorder in the  $R\bar{3}$  symmetry.

### First-principles calculations

A series of nine possible structural models of STA-2(BDAB) were created, assuming both hydroxyls are located in cancrinite cages, on two of the six hydroxyl sites located by diffraction, and are denoted subsequently as OXOY, where X and Y refer to the position of the hydroxyls as shown in Fig. 4a. In the models chosen, either O91 or O92 were occupied, with the second hydroxyl placed on one of the other (O93 to O96) possible positions. No water was included in



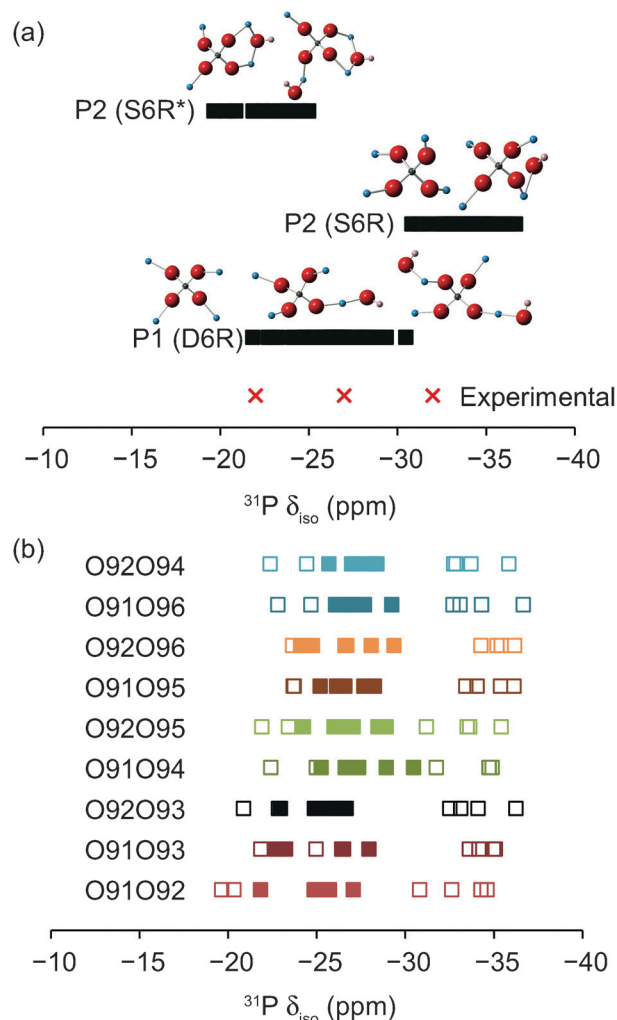
**Table 4** Grouping of STA-2(BDAB) structural models according to the relationship of the two hydroxyl groups

Model	Relationship of hydroxyl groups
O91O92, O91O93, O92O93	Located in same can cage
O91O94, O92O95	Located in different can cages with closest O–O distance of 5.97 Å
O91O95, O92O96	Located in different can cages but share a 4MR
O91O96, O92O94	Located in different can cages with closest O–O distance of 6.19 Å

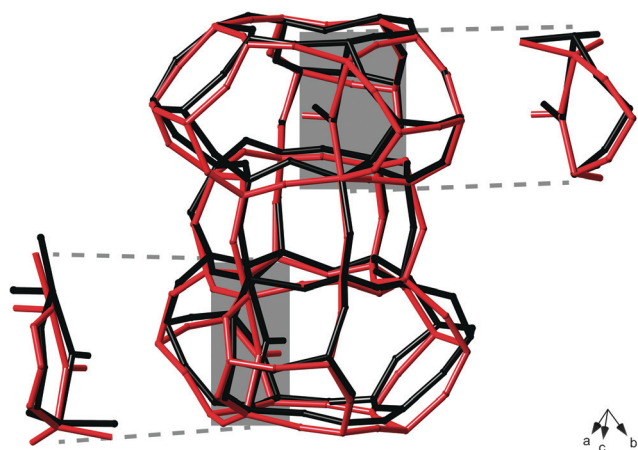
any of the models, as this was not located by diffraction – however, as shown in the ESI,<sup>†</sup> there are no significant changes in the  $^{27}\text{Al}$  and  $^{31}\text{P}$  NMR spectra for hydrated or dried samples. The models can be divided into four separate groups, according to the structural relationship between the two occupied hydroxyl sites, *i.e.*, whether they share a cancrinite cage or a 4MR, and their spatial proximity. This is summarised in Table 4, and shown schematically in Fig. 4b. All structures were optimised as described in the Experimental section. Fig. 5 shows an overlay of part of one of the structural models (O92O94) both prior to and post DFT optimisation. There are some significant changes in the framework geometry after optimisation, as the coordinates determined by diffraction are the weighted averages of atomic positions when the hydroxyl group is present or absent. Furthermore, the Al–OH distances are between 2.15 and 2.3 Å prior to optimisation, but between 1.95 and 1.98 Å post optimisation. The initial distances (too long for typical Al–O bonds) result, as described above, from the weighted average of the Al atomic coordinates.<sup>21</sup> The distances are considerably shorter after optimisation as there is a significant change in the position of the framework Al.

After optimisation,  $^{31}\text{P}$  and  $^{27}\text{Al}$  NMR parameters were calculated for each of the structural models. These are given in full in the ESI.<sup>†</sup> The  $^{31}\text{P}$  calculated isotropic chemical shifts are plotted in Fig. 6a. For P in D6Rs (P11–P16) a range of shifts are observed centred on –26 ppm, reflecting the

variation of the local geometry around P in the disordered structure. P11–16 are found with three different NNN coordination environments in the models, *i.e.*,  $\text{P}(\text{OAl}^{\text{IV}})_4$ ,  $\text{P}(\text{OAl}^{\text{IV}})_3(\text{OAl}^{\text{V}})$  and  $\text{P}(\text{OAl}^{\text{IV}})_2(\text{OAl}^{\text{V}})_2$ , as shown in Fig. 6a. However, there appears to be no significant differences in the calculated chemical shifts for these species, with resulting spectral resonances expected to have considerable overlap in the MAS spectrum. However, there is a significant difference



**Fig. 6** Calculated  $^{31}\text{P}$  isotropic chemical shifts,  $\delta_{\text{iso}}$ , for the structural models of STA-2(BDAB), described in the text. Experimental values are shown for comparison. Also shown are the possible local coordination environments for each P. In (a) all calculated shifts are shown together, while in (b) data for each individual model are shown separately. In (b), filled and empty squares denote P in D6Rs and S6Rs, respectively.

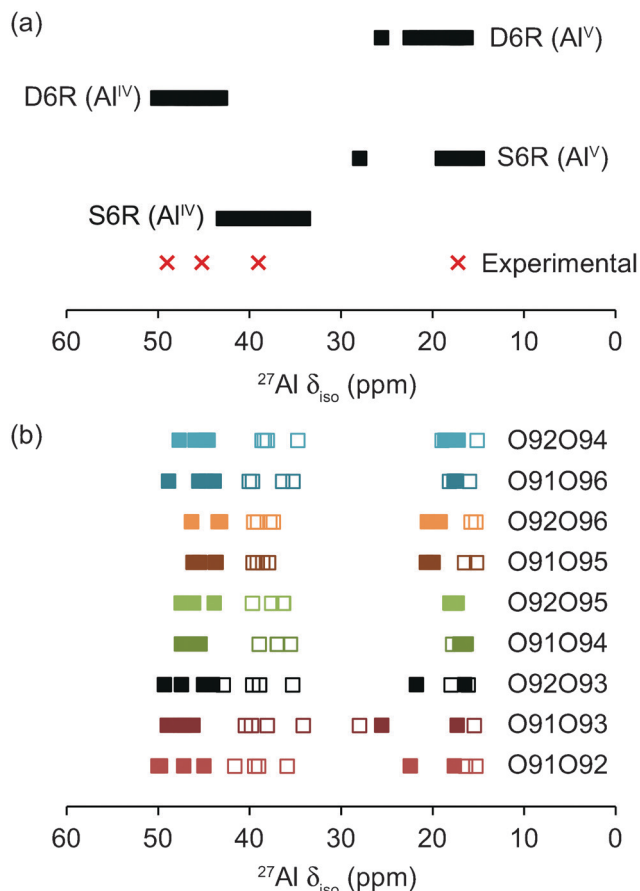


**Fig. 5** Overlay of part of the framework (shown as a wireframe model) in the O92O94 structural model of dehydrated STA-2(BDAB), both pre (black) and post (red) optimisation of the geometry using DFT calculations. Atoms in the BDAB template are not shown for clarity. The significant changes in the position of the  $\text{Al}^{\text{V}}$  species once the neighbouring hydroxyl site is occupied are highlighted.



in the  $^{31}\text{P}$  calculated isotropic chemical shifts for P in S6Rs (P21–P26), with two separate ranges, centred on  $-33$  and  $-22$  ppm, respectively. The P2 sites have a number of different possible environments, as also shown in Fig. 6a, with 0, 1 or 2  $\text{Al}^{\text{V}}$  NNN. However, the large difference in calculated chemical shift results not from the number of  $\text{Al}^{\text{V}}$  NNN, but from whether the  $\text{Al}^{\text{V}}$  are coordinated to the same hydroxyl group, *i.e.*, whether the two Al species the hydroxyl group bridges are attached to the same P atom. These environments are denoted S6R\* in Fig. 6, and result in an increase in the calculated chemical shift of 8–14 ppm. It has been shown previously in the literature<sup>35</sup> that for  $\text{PO}_4$  species there is a relationship between the chemical shift and the average Al–O–P angle, with a decrease in shift corresponding to an increased average Al–O–P angle. Examination of the average P–O–Al bond angles for the structural models used here reveals a decrease from  $150^\circ$ – $154^\circ$  for P21–26 in S6R environments to  $142^\circ$ – $147^\circ$  for S6R\* environments, with the latter much more similar to those for P11–P16 (of  $140^\circ$ – $148^\circ$ ), in agreement with the results of Müller.<sup>35</sup> For each of the different models used the range of calculated shifts (shown in Fig. 6b) is very similar, and it is not possible to distinguish between them purely on this basis, although those where the hydroxyls share a cancrinite cage (O91O92, O91O93 and O92O93) perhaps show slightly poorer agreement with experiment. The three major resonances in the  $^{31}\text{P}$  MAS NMR spectrum may, therefore, be assigned to P in S6R, D6R and S6R\* environments in order of increasing chemical shift.

The  $^{27}\text{Al}$  calculated isotropic chemical shifts are plotted in Fig. 7a, and compared to the average shifts extracted from the MQMAS experiments shown in the ESI† for the resonances in STA-2(BDAB). The shifts predicted for  $\text{Al}^{\text{IV}}$  in D6Rs (*i.e.*, Al21–26) are generally higher than those for S6R  $\text{Al}^{\text{IV}}$  (*i.e.*, Al11–Al16). The distribution of local environments produces a range of calculated  $C_Q$  values. In general, those for  $\text{Al}^{\text{IV}}$  in D6Rs are generally higher (between 3 and 6 MHz) than those for S6R (between 1 and 4 MHz). This prediction agrees with the appearance of the resonances in the  $^{27}\text{Al}$  MQMAS spectra shown in the ESI†, where the resonance at higher chemical shift is broader than that at lower shift, although in both cases the lineshapes suggest a distribution of  $C_Q$ . It should also be noted that the  $C_Q$  values observed for Al1 and Al2 in calcined STA-2 are 2.0 and 3.5 MHz, respectively. For  $\text{Al}^{\text{V}}$  species, those in D6R and S6R have similar ranges of chemical shifts, between 15 and 30 ppm, and so are unlikely to be separated in the experimental MAS spectrum, supporting (and explaining) the observation of a single broad resonance in this region. A broad range of  $C_Q$  values are predicted for these species, from 2 to 9 MHz, though the majority lie between 3.6 and 8.4 MHz. As also observed for  $^{31}\text{P}$ , the shifts calculated for each of the models are very similar, with only those containing hydroxyls in the same cancrinite cage showing some differences to the rest, and these are generally in poorer agreement with experiment. In contrast to  $^{31}\text{P}$ , the position of a resonance for  $^{27}\text{Al}$  depends not just on the isotropic chemical shift, but also on the isotropic second-order quadrupolar shift,  $\delta_Q$  (given by

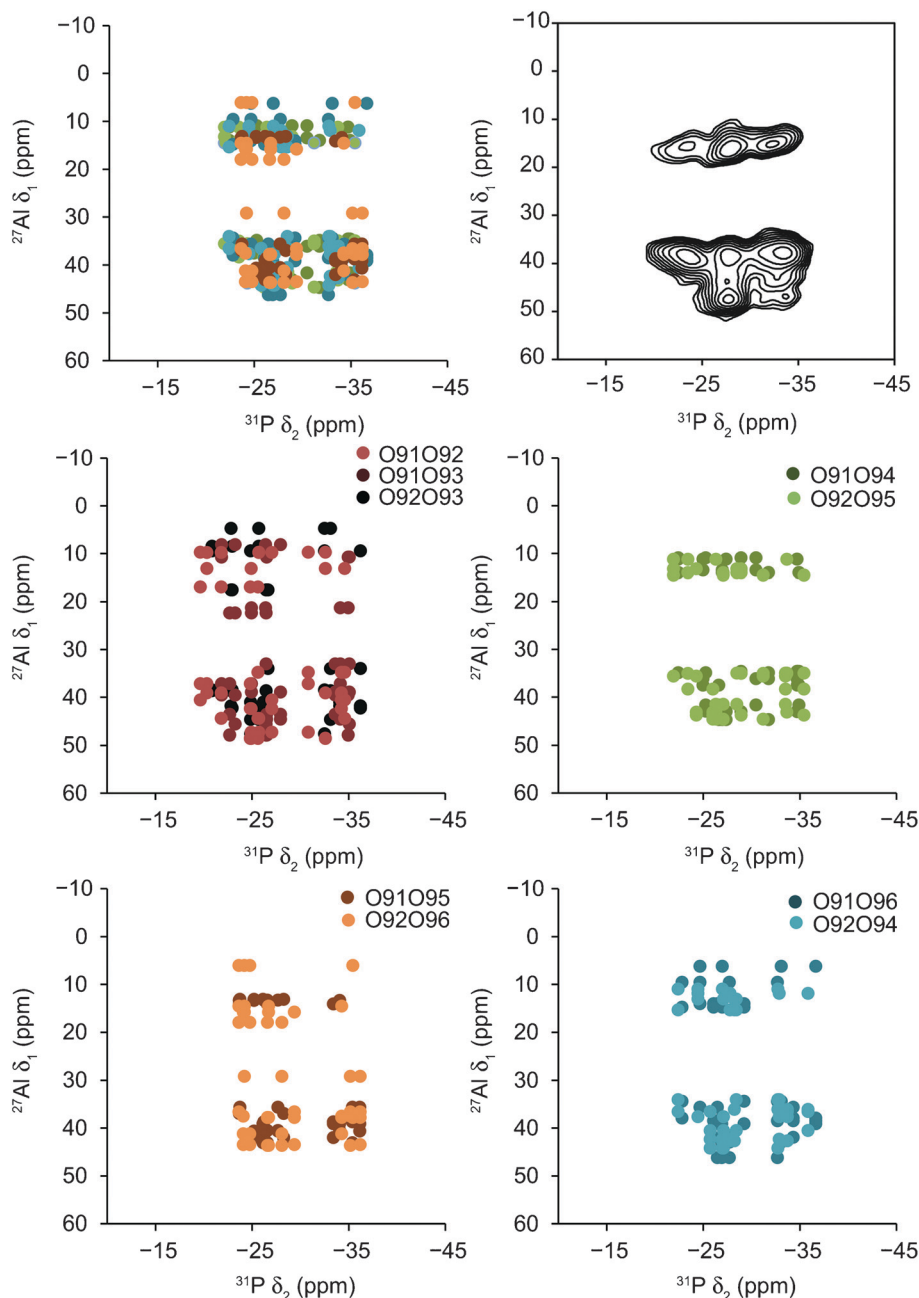


**Fig. 7** Calculated  $^{27}\text{Al}$  isotropic chemical shifts,  $\delta_{\text{iso}}$ , for structural models of STA-2(BDAB), described. Experimental values (extracted from MQMAS NMR spectra) are shown for comparison. In (a) all calculated shifts are shown together, while in (b) data for each individual model are shown separately. In (b), filled and empty squares denote Al in D6Rs and S6Rs, respectively.

$-(16/15) ((\nu_Q^{\text{PAS}})^2/\nu_0) (1 + \eta_Q^2/3)$  for an  $I = 5/2$  nucleus).<sup>36</sup> The presence of the quadrupolar interaction will, therefore, have a small effect upon the resonance position when  $C_Q$  is small (*e.g.*,  $\sim 2.2$  ppm for a spin  $I = 5/2$  nucleus with a  $C_Q$  of 3 MHz at 14.1 T) but an increasingly important effect as the quadrupolar interaction is larger.

It is also possible to consider the agreement between experimental  $^{27}\text{Al}$  MQMAS NMR spectra and the calculated NMR parameters by plotting the predicted centre-of-gravity ( $\delta_1$ ,  $\delta_2$ ) of the resonance associated with each Al species in the calculated models of STA-2(BDAB). The  $\delta_2$  position is simply a sum of the isotropic chemical and second-order quadrupolar shifts (as described for the MAS spectrum above). The position in  $\delta_1$  is given (for a sheared triple-quantum MAS spectrum) by  $\delta_1 = (17/31) \delta_{\text{iso}} + (32/93) \delta_Q$  (for  $I = 5/2$ , using the scaling convention of ref. 25). Fig. S7.1 in the ESI† shows the predicted centre-of-gravity for each of the Al species in the structural models considered. When all models are considered together, there is good agreement between the resonance positions for calculated and experimental data for  $\text{Al}^{\text{IV}}$  species. The agreement for  $\text{Al}^{\text{V}}$  species is less good, perhaps reflecting the disorder and possible





**Fig. 8** Calculated cross peak positions in  $^{27}\text{Al}/^{31}\text{P}$  (20.0 T) J-HETCOR spectra for each of the structural models of STA-2(BDAB), generated by correlating the  $^{31}\text{P}$  isotropic chemical shift with the  $^{27}\text{Al}$  resonance position (a sum of both isotropic chemical shifts and isotropic quadrupolar shifts as described above) for species joined by an Al–O–P linkage, as shown in Table 5. Also shown is the experimental spectrum (20.0 T) of STA-2(BDAB) for comparison.

dynamics of the coordinated hydroxyl groups. The calculated data for hydroxyl groups found in the same cancrinite cage (O91O92/O91O93/O92O93) is in poorest agreement with experiment, with a particularly large spread for the  $\text{Al}^{\text{V}}$  species. The differences between the three other groups of models are much smaller, with all in reasonable agreement with experiment.

Fig. 8a shows the predicted position of cross peaks in a  $^{27}\text{Al}/^{31}\text{P}$  (20.0 T) J-HETCOR spectrum for each of the structural models of STA-2(BDAB). These plots were generated by correlating the isotropic chemical shifts for  $^{31}\text{P}$  with the  $^{27}\text{Al}$

resonance position (a sum of both isotropic chemical shifts and isotropic quadrupolar shifts as described above) for all species joined by a single Al–O–P linkage. The Al and P species linked by two bonds are shown in Table 5. In general, and as noted previously for the MQMAS spectra, the agreement between calculated results and experiment is better for  $\text{Al}^{\text{IV}}$  than for  $\text{Al}^{\text{V}}$ . Particularly good agreement is observed for the two sets of models where the hydroxyl groups are located in different cancrinite cages, with the obvious absence of a cross peak between the  $^{31}\text{P}$  at  $\sim -22$  ppm and the  $^{27}\text{Al}$  at  $\sim 48$  ppm. For the set of models where the hydroxyl groups



**Table 5** Expected through-bond Al–O–P connectivities (denoted by X) in as-made STA-2(BDAB)

	Al/P in templated structure	P1 in calcined dehydrated structure (D6R)						P2 in calcined dehydrated structure (S6R)					
		11	12	13	14	15	16	21	22	23	24	25	26
Al1 in calcined dehydrated structure (S6R)	11				X			X	X	X			
	12			X				X	X		X		
	13						X			X	X	X	
	14					X				X	X		X
	15		X					X				X	X
	16	X							X			X	X
Al2 in calcined dehydrated structure (D6R)	21	X			X		X	X					
	22		X	X		X			X				
	23		X	X			X			X			
	24	X			X	X					X		
	25		X		X	X						X	
	26	X		X			X						X

are in the same cancrinite cage the agreement with experiment is poor, with predicted cross peaks observed where the signal is absent experimentally, and with more generally a wider spread of intensity, particularly for cross peaks involving Al<sup>V</sup>. The models where hydroxyls are in separate cancrinite cages but sharing a 4MR result in correlation plots that do not completely agree with experiment, with only cross peaks observed at higher <sup>31</sup>P shift, *i.e.*, few or no cross peaks observed at <sup>31</sup>P shifts lower than −24 ppm. However, this does not rule out the presence of this type of hydroxyl ordering in STA-2(BDAB), but shows that if it is present it must correspond to a relatively minor component of the structure, co-existing with significantly more local environments where the hydroxyls are more remote in order to generate the correct cross peaks with <sup>31</sup>P at lower chemical shift. The ESI† shows a similar set of plots to those in Fig. 8, but where only the <sup>31</sup>P and <sup>27</sup>Al isotropic chemical shifts are correlated, *i.e.*, the isotropic quadrupolar shift for <sup>27</sup>Al is not included. Whilst, in principle, this should provide less accurate results, the agreement with experiment is perhaps better than that seen in Fig. 8 – a somewhat surprising result. However, while it has been shown that good agreement between experiment and calculation is usually obtained when calculating chemical shifts, the calculation of quadrupolar NMR parameters is often not quite as accurate, and a small but systematic overestimation of *C<sub>Q</sub>* has been observed for a number of nuclei including <sup>25</sup>Mg, <sup>71</sup>Ga, <sup>93</sup>Nb, <sup>17</sup>O and <sup>27</sup>Al.<sup>7,17,37–41</sup> This error ultimately leads to errors in the isotropic quadrupolar shifts, and it appears in this case these are more significant than the small errors associated with the calculation of  $\delta_{\text{iso}}$ . In the latter case, of course, there may be a cancellation of any systematic errors within the referencing process, whereas this is clearly not possible when calculating quadrupolar shifts.

Although attention has focussed on the calculation of <sup>27</sup>Al and <sup>31</sup>P NMR parameters in order to understand the location of the charge-balancing hydroxyls in STA-2(BDAB), DFT calculations also provide information on the NMR parameters for the template molecule (*i.e.*, for <sup>13</sup>C and <sup>15</sup>N in BDAB), and these are shown in full in the ESI.† Very similar results are obtained for all of the sets of models used,

suggesting little interaction between the SDA and the framework itself. For <sup>15</sup>N, the calculations confirm the assignment of the apical and quaternary nitrogens in the previous work,<sup>21</sup> although the difference between the calculated values (~52 ppm) is slightly greater than that observed experimentally (~43 ppm). However, it should also be noted that the previous experimental work was carried out on hydrated samples of STA-2(BDAB), and the presence of water (while having almost no effect upon the framework structure) may have a greater effect on the template molecules with which it shares the pore space. The DFT calculations also confirm the previous assignment of the <sup>13</sup>C CP MAS NMR spectrum of STA-2(BDAB).

It should be noted that significant template dynamics have been observed in AlPO frameworks (often evidenced by changes to the NMR spectra of the framework atoms).<sup>13</sup> Furthermore, significant template motion was also observed upon geometry optimization in the calculations in this work, indicating that dynamics may also play a role in this material. Dynamics may affect the experimental NMR parameters and, ultimately, the agreement between experiment and the calculated results. A further source of disagreement between calculation and experiment is the underestimation of dispersion forces in the DFT methods used (which may be of more significance for these inorganic/organic frameworks than for simple rigid solids). Although not used in this work, recently semi-empirical dispersion correction schemes have become available in periodic codes to attempt to account for this problem, and this approach does offer promise for future work. However, given the complex nature of the systems DFT calculations have been used here to aid spectral assignment and interpretation and to offer overall insight into the local structure, rather than providing precise agreement with experimental values, although both dynamics and dispersion forces may need to be taken into account when considering other materials.

## Conclusions

Insight into the detailed structure of the pure aluminophosphate form of STA-2 was obtained by using a





combination of diffraction, multinuclear solid-state NMR spectroscopy and first-principles calculations. Although previous work had shown that STA-2 could be prepared using bis-diazabicyclooctane-butane (BDAB) as an alternative SDA to the 1,4-bis-*N*-quinuclidiniumbutane (BQNB) used in the initial synthesis, the NMR spectra of the as-made form had not been assigned, and the positions of the charge-balancing hydroxyls not been located. The similarity of NMR spectra for STA-2(BDAB) and STA-2(BQNB) led to the suggestion in the earlier work that BDAB was also divalent when incorporated into the material and that the structures of the two as-made forms were very similar. Hence, the conclusions of the investigation in the current work are expected to be equally applicable to STA-2(BQNB).

From the ratio of  $\text{Al}^{\text{IV}}$ – $\text{Al}^{\text{V}}$  species in  $^{27}\text{Al}$  MAS NMR spectra it is possible to determine that the charge-balancing hydroxyls are bridging rather than terminal in the structure. The similar relative intensities of the two major  $\text{Al}^{\text{IV}}$  resonances suggests that the hydroxyl groups bridge formal “Al1” and “Al2” species (*i.e.*, those in S6R and D6R, respectively), although the presence of only one resonance in the region of the spectrum corresponding to  $\text{Al}^{\text{V}}$  is, therefore, somewhat of a surprise. However, DFT calculations carried out using a series of structural models where hydroxyls were located on two of six possible sites within the cancrinite cages identified by diffraction, predicted distinctly different isotropic chemical shifts for Al11–16 and Al21–26 when four coordinate, but showed that the shifts were very similar for five-coordinate species in both cases, enabling the experimental spectrum to be assigned. A range of  $C_Q$  values were predicted for all types of Al species, but those for  $\text{Al}^{\text{IV}}$  in D6R are generally higher (3–6 MHz), than those for  $\text{Al}^{\text{IV}}$  in S6R (1–4 MHz), in agreement with the shape and width of the resonances observed in two-dimensional  $^{27}\text{Al}$  MQMAS spectra. DFT calculations were also able to assign the three resonances in the  $^{31}\text{P}$  spectrum, with that at –27 ppm resulting from P in D6R (*i.e.*, P11–P16), while the two less intense resonances at –23 and –33 ppm result from P in S6R (*i.e.*, P21–P26). Little difference in the  $^{31}\text{P}$  chemical shift was observed as the number of  $\text{Al}^{\text{V}}$  NNN increased, but a significant change in shift was observed for S6R species when the hydroxyl group bridged between two of the NNN Al species (S6R\*). This was shown to result from a significant change in the Al–O–P bond angles (from 150°–154° for S6R to 142°–147° in S6R\* environments).

From a consideration of simply  $^{27}\text{Al}$  and  $^{31}\text{P}$  calculated isotropic shifts it is difficult to rule out completely any particular type of structural model, as all show reasonable agreement with experiment. However, a comparison of experimental and calculated  $^{27}\text{Al}$  MQMAS spectra, and in particular  $^{27}\text{Al}/^{31}\text{P}$  J-HETCOR spectra (created using the expected Al–O–P connectivities in the STA-2 framework structure), revealed some differences between the model types. In both cases, the Al resonance position was considered (*i.e.*, a sum of isotropic chemical and quadrupolar shifts); however, it was shown that the (systematic) errors associated with the

calculation of the quadrupolar coupling constants (and therefore the quadrupolar shifts) resulted in a greater spread of the calculated results compared to those seen in experiment, particularly for  $\text{Al}^{\text{V}}$  species. In general, models where the hydroxyls were located in the same cancrinite cage were shown to be in less good agreement, ruling out this possibility in the real material. Generally poorer agreement was also shown between experimental and calculated data when the hydroxyl groups were in different cancrinite cages but shared a 4MR. However, this generally resulted in the absence of cross peaks in J-HETCOR spectra, rather than the presence of cross peaks that were not observed experimentally, leading to the conclusion that while most hydroxyls must be located in different cancrinite cages and also separated spatially, the presence of a small number of hydroxyls that shared 4MR could not be ruled out completely. As described in the experimental section, owing to template movement it was difficult to ensure the geometry optimisations were “converged”, making it difficult to compare the relative energies of the structural models. However, in general, for the optimized structures used, most of the models where the hydroxyls shared the same cancrinite cage were slightly higher in energy (0.3 eV per unit cell of STA-2(BDAB), or 28.9 kJ mol<sup>–1</sup> of STA-2(BDAB)), tentatively supporting the conclusions from the NMR calculations.

Although it is difficult to refine a “structural model” for disordered materials using diffraction-based methods, if the information obtained is combined with that from NMR spectroscopy and from DFT calculations of NMR parameters a much more detailed understanding of both long- and short-range structure can be obtained. In many cases, the local structure gives important details on the synthesis mechanisms of these solids, and furthermore plays a very important role in determining the physical and chemical properties of the bulk material. The approach of combining diffraction and NMR spectroscopy and simulation to elucidate structure, often referred to as NMR Crystallography, offers a promising route for the future study of as-made and substituted microporous materials, and an understanding of the structure–property relationships that will ultimately determine their use and applications.

## Acknowledgements

We would like to thank EPSRC (EP/E041825/1, EP/J501542/1, and EP/F018096/1) the European Commission FP6 Marie Curie Research Training Network “INDENS” (MRTN-CT-2004-005503), Johnson Matthey and the Leverhulme Trust (F/00 268/BJ) for support. We thank EaStCHEM for computational support through the EaStCHEM Research Computing Facility. The UK 850 MHz solid-state NMR Facility used in this research was funded by EPSRC and BBSRC, as well as the University of Warwick including *via* part funding through Birmingham Science City Advanced Materials Projects 1 and 2 supported by Advantage West Midlands



(AWM) and the European Regional Development Fund (ERDF). We would also like to thank Dr Julien Trebosc and Dr Dinu Iuga for their help at the facilities in Lille and Warwick, respectively. Professor Alex M. Z. Slawin (University of St. Andrews) is thanked for collection of single crystal diffraction data and helpful discussions.

## References

- 1 S. T. Wilson, B. M. Lok, C. A. Messina, T. R. Cannan and E. M. Flanigen, *J. Am. Chem. Soc.*, 1982, **104**, 1146.
- 2 Z. Han, A. L. Picone, A. M. Z. Slawin, V. R. Seymour, S. E. Ashbrook, W. Zhou, S. P. Thompson, J. E. Parker and P. A. Wright, *Chem. Mater.*, 2010, **22**, 338.
- 3 A. Tuel, C. Lorentz, V. Gramlich and C. Baerlocher, *C. R. Chim.*, 2005, **8**, 531.
- 4 A. Simmen, L. B. McCusker, Ch. Baerlocher and W. M. Meier, *Zeolites*, 1991, **11**, 654.
- 5 J. M. Griffin, L. Clark, V. R. Seymour, D. W. Aldous, D. M. Dawson, D. Iuga, R. E. Morris and S. E. Ashbrook, *Chem. Sci.*, 2012, **3**, 2293.
- 6 S. P. Brown, S. E. Ashbrook and S. Wimperis, *J. Phys. Chem. B*, 1999, **103**, 812.
- 7 S. E. Ashbrook, M. Cutajar, J. M. Griffin, Z. A. D. Lethbridge, R. I. Walton and S. Wimperis, *J. Phys. Chem. C*, 2009, **113**, 10780.
- 8 C. M. Morais, V. Montouillout, M. Deschamps, D. Iuga, F. Fayon, F. A. A. Paz, J. Rocha, C. Fernandez and D. Massiot, *Magn. Reson. Chem.*, 2009, **47**, 942.
- 9 C. Martineau, C. Mellot-Draznieks and F. Taulelle, *Phys. Chem. Chem. Phys.*, 2001, **13**, 18078.
- 10 C. Martineau, B. Bouchevreau, Z. J. Tian, S. J. Lohmeier, P. Behrens and F. Taulelle, *Chem. Mater.*, 2001, **23**, 4799.
- 11 P. J. Byrne, J. E. Warren, R. E. Morris and S. E. Ashbrook, *Solid State Sci.*, 2009, **11**, 1001.
- 12 S. E. Ashbrook, M. Cutajar, C. J. Pickard, R. I. Walton and S. Wimperis, *Phys. Chem. Chem. Phys.*, 2008, **10**, 5754.
- 13 S. Antonijevic, S. E. Ashbrook, S. Biedasek, R. I. Walton, S. Wimperis and H. X. Yang, *J. Am. Chem. Soc.*, 2006, **128**, 8054.
- 14 A. Tuel, S. Calarelli, A. Meden, L. B. McCusker, C. Baerlocher, A. Ristic, N. Rajic, G. Mali and V. Kaucic, *J. Phys. Chem. B*, 2000, **104**, 5697.
- 15 C. J. Pickard and F. Mauri, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2001, **63**, 245101.
- 16 T. Charpentier, *Solid State Nucl. Magn. Reson.*, 2011, **40**, 1.
- 17 C. Bonhomme, C. Gervais, F. Babonneau, C. Coelho, F. Pourpoint, T. Azais, S. E. Ashbrook, J. M. Griffin, J. R. Yates, F. Mauri and C. J. Pickard, *Chem. Rev.*, 2012, **112**, 5733.
- 18 S. E. Ashbrook and D. M. Dawson, *Acc. Chem. Res.*, 2013, DOI: 10.1021/ar300303w.
- 19 J. Cuny, S. Messaoudi, V. Alonzo, E. Furet, J. F. Halet, E. Le Fur, S. E. Ashbrook, C. J. Pickard, R. Gautier and L. Le Polles, *J. Comb. Chem.*, 2008, **13**, 2279.
- 20 G. W. Noble, P. A. Wright and A. Kvik, *J. Chem. Soc., Dalton Trans.*, 1997, 4485.
- 21 M. Castro, V. R. Seymour, D. Carnevale, J. M. Griffin, S. E. Ashbrook, P. A. Wright, D. C. Apperley, J. E. Parker, S. P. Thompson, A. Fecant and N. Bats, *J. Phys. Chem.*, 2010, **114**, 12698.
- 22 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Found. Crystallogr.*, 2008, **A64**, 112.
- 23 S. P. Brown and S. Wimperis, *J. Magn. Reson.*, 1997, **128**, 42.
- 24 J. P. Amoureux, C. Fernandez and S. Steuernagel, *J. Magn. Reson., Ser. A*, 1996, **123**, 116.
- 25 K. J. Pike, R. Malde, S. E. Ashbrook, J. McManus and S. Wimperis, *Solid State Nucl. Magn. Reson.*, 2000, **16**, 203.
- 26 C. A. Fyfe, K. C. Wongmoon, Y. Huang and H. Grondy, *J. Am. Chem. Soc.*, 1995, **117**, 10397.
- 27 J. P. Amoureux, J. Trebosc, J. Wiench and M. Pruski, *J. Magn. Reson.*, 2007, **184**, 1.
- 28 G. Tricot, O. Lafon, J. Trebosc, L. Delevoye, F. Mear, L. Montagne and J. P. Amoureux, *Phys. Chem. Chem. Phys.*, 2011, **13**, 16786.
- 29 M. D. Segall, P. J. D. Lindan, M. J. Probert, C. J. Pickard, P. J. Hasnip, S. J. Clark and M. C. J. Payne, *J. Phys.: Condens. Matter*, 2002, **14**, 2717.
- 30 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- 31 J. R. Yates, C. J. Pickard and F. Mauri, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 024401.
- 32 P. Pyykko, *Mol. Phys.*, 2008, **106**, 1965.
- 33 K. J. D. MacKenzie and M. E. Smith, *Multinuclear Solid-State NMR of Inorganic Materials*, Pergamon Press, Oxford, 2002, ch. 5.
- 34 D. R. Neuville, L. Cormier and D. Massiot, *Geochim. Cosmochim. Acta*, 2004, **68**, 5071.
- 35 D. Muller, E. Jahn, G. Ladwig and U. Haubenreisser, *Chem. Phys. Lett.*, 1984, **109**, 332.
- 36 A. J. Vega, *Quadrupolar Nuclei in Solids*, in *eMagRes*, 2010.
- 37 J. M. Griffin, A. J. Berry and S. E. Ashbrook, *Solid State Nucl. Magn. Reson.*, 2011, **40**, 91.
- 38 P. J. Pallister, I. L. Moudrakovski and J. A. Ripmeester, *Phys. Chem. Chem. Phys.*, 2009, **11**, 11487.
- 39 J. M. Griffin, S. Wimperis, A. J. Berry, C. J. Pickard and S. E. Ashbrook, *J. Phys. Chem. C*, 2009, **113**, 465.
- 40 K. E. Johnston, J. M. Griffin, R. I. Walton, D. M. Dawson, P. Lightfoot and S. E. Ashbrook, *Phys. Chem. Chem. Phys.*, 2010, **13**, 7565.
- 41 F. Blanc, D. S. Middlemiss, L. Buannic, J. L. Palumbo, I. Farnan and C. P. Grey, *Solid State Nucl. Magn. Reson.*, 2012, **42**, 87.

